

Study on Effect of EAFD Particulate Reinforcement in AA7075 Aluminum Matrix Composites

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Electric arc furnace dust is dangerous solid waste that is generated in steel manufacturing processes. The utilization of the dust from these industries avoids disposing it into industrial waste landfills and saves costs. In this study, an attempt has been made to fabricate aluminum matrix composites reinforced with EAFD particulate, through the conventional powder metallurgy technique. Al-5%EAFD proposed was then compared to the AA7075 alloy, manufactured under the same conditions. According to the results obtained, the addition of EAFD into aluminum matrix resulted in increased microhardness values and had significant contribution to improve Storage modulus compared with the base metal. The reinforcement particle was uniformly dispersed into the matrix and intermetallic phases were not observed in the content of waste blended into the composite.

Keywords: EAFD, aluminum composite, powder metallurgy.

1. Introduction

Steel mills utilize primarily five different inputs groups in their manufacturing processes: raw materials, air, water, fuel and power; however, these supplies are responsible for generating large quantities of several types of solid waste. It is estimated that during steel production between 2 and 4 tonnes of waste, as sludge, dust and slag are generated¹.

Electric arc furnace dust (EAFD) is one of by-products of steelmaking industry resulting from the melting of scrap and from steel refining. It is formed through the conversion of vapours into dust, by means of agglomeration, and physical and chemical transformations. The steel dust formed is collected by a system that captures gases and particulate matter. The type of scrap metal melted inside the electric arc furnace (EAF) determines the chemical composition of the dust, which is defined as a rich material in ferrous or nonferrous metal. EAFD is composed of iron and zinc oxide, which is volatilized, and also calcium, chromium, magnesium, silica and, other toxic compounds as lead and cadmium^{2,3,4,5}.

The formation process of EAFD occurs in two stages: first, the emission of dust "precursors", second, the conversion of these precursors into dust by agglomeration and physico-chemical transformations. The different types of particles are formed by five emission mechanisms of dust precursors, mainly projection of fine droplets by bursting of CO bubbles. Studies confirm the existence of two fine particle populations: submicronic particles of zincite coming from the vaporization of the liquid steel, and film drops;

homogeneous and heterogeneous spherical particles, whose composition corresponds either to the slag or to the steel bath with an enrichment in zinc. The size of large particles vary between 20 and 500 μm , with irregular shapes, and come from the direct fly-off of solid particles during the introduction of powder materials into the EAF^{6,7}.

Zinc is generally found as franklinite, ferrite particles rich in zinc (spinel - ZnFe_2O_4), and as calcium ferrite (spinel - CaFe_2O_4), or ferrites with isomorphously replaced metals ($\text{Zn}_x\text{Me}_y\text{Fe}_2\text{O}_4$, where $\text{Me} = \text{Mn}, \text{Co}, \text{Ni}, \text{Cr}, \text{etc.}$, and zincite (ZnO). With the exception of franklinite, iron is mostly found in the magnetite phase (Fe_3O_4). In addition to the oxides mentioned, the EAFD includes in its composition: carbon (coke), calcite (CaCO_3), silica (SiO_2), and aluminum silicate. Depending on its composition and combination of elements, EAFD becomes a byproduct that is difficult to recycle, impeding the development of recovery process of the metal of interest^{6,7,8,9}.

In the steelmaking process in electric arc furnaces, dust formation has been estimated to be between 15-25 kg of dust per tonne of steel produced. Among the suggested methods for recycling this solid waste, pyrometallurgical processes, as well as its application in the ceramic materials field, have been widely used in industry. However, about 60% of dust produced all over the world is still being stored in yards^{4,6,10,11,12,13}.

The composition of the EAFD is extremely variable and influenced by type of steel manufactured, nature and quantity of scrap and alloy additions, which make up part of burden. In addition to factors related to raw materials, operational factors also affect its composition, such as the type of furnace, operating cycle, and gas cleaning system efficiency¹⁴.

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Powder Metallurgy (PM) is an appropriate method for metal matrix composite processing due to its simple processing and ample applications in different fields, such as transportation, the aerospace and infrastructure^{15,16}. PM allows for recycling of solid waste in its solid state, transforming it into useful products through the reduction of metal oxides present in the steelmaking dusts and in its composition during sintering¹⁷.

Many solutions have been proposed for the technical utilization of the EAFD in several fields. The main approaches for recycling the waste are related to pyrometallurgical process, by agglomeration of dust and briquette manufacture to recovery of iron contained in the EAFD⁴. Havlik et al.⁷ study efficiency of recovery of zinc with sulfuric acid as the leaching agent. Among the approaches, the addition of EAFD into ceramic materials are most commonly seen^{10,18,19}. Stathopoulos et al.¹¹ investigated the stabilization of Zn, Pb and Cd into structural ceramics containing 2.5% and 5% wt. of EAFD.

The application of EAFD as reinforcement in aluminum matrix composites (AMC) has been a technological and economic alternative when compared to other reinforcements traditionally applied. In this study, in order to assess the influence of EAFD on the mechanical properties of the AMC, 5% of waste was embedded into the AA7075 alloy, and later hardness and morphological behaviour were examined.

2. Materials and Methods

This research project was carried out in three stages: EAFD characterization, composite preparation, testing and characterization of composite manufactured. The AA7075 was blended to content of 5% in EAFD weight (wt.%), which had particle size smaller than 53 μm , and 2 wt.% stearic acid ($\text{C}_{18}\text{H}_{36}\text{O}_2$) of the total weight of the sample. Percentage of waste applied was based in blends that have been mentioned in previous studies^{20,21,22}. The AA7075 alloy used was produced by ALCOA and EAFD was supplied by a steel mini-mill located in northeast Brazil, originating from smelting of carbon steel scrap.

2.1 Powders characterization

Initially EAFD was separated into 4 size ranges, and only the size range smaller than 53 μm was used for this research. The EAFD and the AA7075 alloy were submitted to chemical analysis through X-ray fluorescence (XRF) in a Rigaku ZSX Primus spectrometer. A Bruker D8 advance X-ray diffractometer (XRD), using a $\text{CuK}\alpha$ ($\lambda = 1.5405 \text{ \AA}$) was operated to establish the phases present, the measuring range being from 10 to 80° (2 θ) using a step size of 0.02° and a step time of 1s per step. For scanning electron microscopy (SEM), the samples were prepared on a graphite support, and micrographs were carried out on the HITACHI TM3000, and a JOEL JSM5900 scanning electron microscope equipped with

an EDS was also used for detailed study of the microstructural features and qualitative elemental composition of EAFD, and the particle size distribution by laser diffraction was measured by Malvern Mastersizer.

2.2 Composite preparation

The AA7075 aluminum alloy powder was used as starting material. EAFD, aluminum alloy, powders (5 wt.% and average particulate size of 20 μm) and stearic acid (2 wt.%) were mixed, with a ball to powder weight ratio of 10:1, in isopropyl alcohol in order to prevent oxidation risk and provide effective mixing, reducing cold welding of powders²³. Other alcohols are commonly used in ball milling to as control agent^{15,21}. After mixing, the blend was dried at 150°C to remove isopropyl alcohol. Afterwards, they were compacted by uniaxial pressing in a circular (16 mm diameter) metal die at 1500 MPa, and the sintering was carried out at a temperature of 500°C during 5 hours under nitrogen atmosphere.

2.3 Characterization and Testing

The sintered composite morphology was observed by HITACHI TM3000 scanning electron microscope. Vickers microhardness values were determined using a Emcotest Durascan G5 microhardness tester under a load of 200 g for 15 s, following the ASTM E-384. Young's modulus (E) measurements were performed using a Shimadzu Dynamic UltraMicro Hardness tester type DUH-W201S with a Vicker indenter. For each sample at least five indentations on the sample surface were made, with a maximum load of 100 mN, and a hold time at load-unload of 10 s.

Specimens, originally cylindrical, were prepared and cut for DMA tests with a low speed diamond saw into rectangular section 16x4x2 mm (length x width x thickness). These tests were performed in a dynamic mechanical analyzer (DMA 242E Artemis, Netzsch) using a three-point bending. Dynamic mechanical analysis (DMA) tests were performed for frequency of 1Hz, and using a maximum amplitude of 20 μm , from room temperature to 390°C, during the heating phase with 10 °C/min rate^{24,25}.

3. Results and Discussion

Results from chemical analyses of the EAFD are presented in Table 1. The table shows that iron and zinc are the main components of the dust, nearly 32.69 and 26.06 wt.%. These values are consistent with those observed in wastes generated in the carbon steel manufacturing processes, however, despite of results obtained being within the expected values in the literature, high iron content indicates a great loss of interest element in the process. The residue researched also has small amounts of heavy metals, including cadmium and lead. Levels of lead and cadmium remained within the

Table 1. Chemical analysis of the main elements present in the EAFD and AA7075.

Element	Zn	Fe	Al	Si	Mg	Cu	Ca	Cd	Pb
AA7075	2.06	0.05	94.78	0.21	2.08	0.62	0.02	nd	nd
EAFD	26.06	32.69	2.49	7.54	3.98	0.17	6.80	0.02	0.66

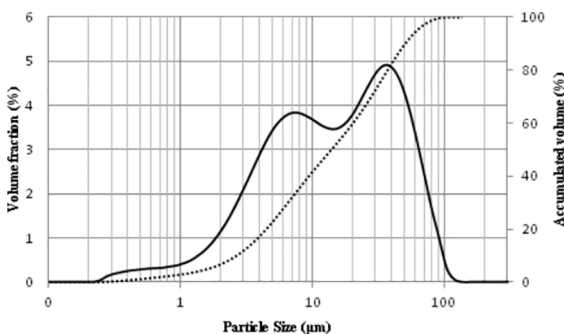
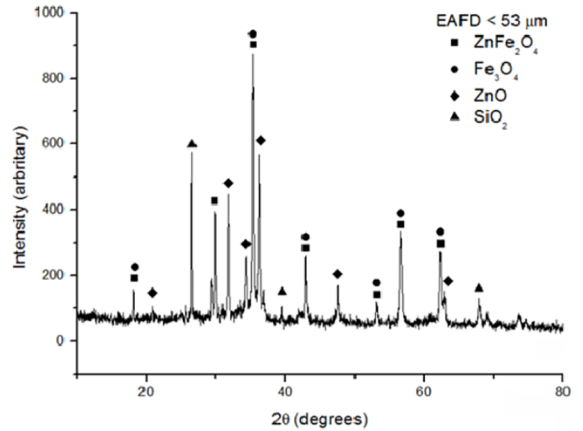
minimum levels observed in the literature^{7,8}, suggesting a good selection of the scrap used in the production process.

Particle size distribution of the examined EAFD is shown in Figure 1. It presents a heterogeneous distribution of particle size and contains two major size fractions: a very fine grained portion (10 nm to 7 μm) and a coarser part (1-120 μm)²⁶. Most of the EAFD have a volume average diameter of 20.22 μm . The study found that the median particle size (d_{50}) is around 13.87 μm , and the majority (d_{90}) of the particles were below 50.44 μm . This prevailing fine size distribution favour their application in metallic powders.

Dutra et al.¹³ analyzed a EAFD of a steel plant located in the southeast region of Brazil and found that 15% of the particles were coarser than 10 μm , the median particle size being around 0.5 μm . Sofilić et al.⁸ characterized the dust samples of a Croatian steelmaker consisting of 100-125 μm particles. On the other hand, Omran & Fabritius² examined the electric arc furnace dust of carbon steel produced in steel plants from Finland, showing that 90% of the particles were smaller than 5.42 μm , the median being 1.63 μm .

The X-ray diffraction pattern of the as-received EAFD is illustrated in Figure 2. It is a waste of complex mineralogy, in which four major phases were identified: Franklinite (ZnFe_2O_4), Magnetite (Fe_3O_4), Zincite (ZnO) and Quartz (SiO_2), which form the phases for the main elements of its composition. The results reveal that EAFD has other peaks for minority elements, but in insufficient amounts for appropriate detection. Other phases typically found in the literature^{2,4,7,8} and the peaks overlap can interfere in the identification of phases belonging to the spinel group²⁷.

A typical morphology of the EAFD fraction obtained by SEM with EDS analysis of selected particles are presented in Figure 3a. EAFD are composed by irregularly-shaped particles (see arrows in Figure 3a) and smooth spherical (Figure 3b). Similar forms can be observed in literature^{2,6,11,13}. EAFD is

**Figure 1.** Particle size distribution of the EAFD.**Figure 2.** X-ray diffraction diffractogram of EAFD.

composed predominantly of spherical particles in the form of agglomerates with varied dimensions (Figure 3b). The predominantly fine particles can differ from each other because of their mineralogy. They come from the projection of liquid droplets, and during the condensation of the vapors of zinc contained in the EAF fumes⁶. EDS analysis (Figure 3c) indicates that most of the zinc is associated with the finer dust particles, probably as zincite and ferrites.

SEM micrographs of the Al-5%EAFD sintered composite are shown in Figure 4. An image with a magnification of 200 \times (Figure 4a) was used to characterize the matrix of composite and an overview of the dispersion of larger and irregular particles of EAFD with a magnification of 500 \times (Figure 4b). Spherical shape particle composed of aluminum and zinc agglomerated (Figure 4c). The effect of variation of particles size has not been investigated.

The addition of 5 wt.% of EAFD resulted in a significant increase in Vicker microhardness (HV) from 85.09 ± 5.71 to 124.6 ± 16.92 , in relation to the unreinforced alloy under experimental conditions implemented. The increase in hardness value is caused by the presence of iron and zinc which are in the form of oxides, which have high hardness. Adeosun et al.²⁰ studied the effect of EAFD addition in 2-20 wt.% content, on the mechanical properties of 6063 aluminum alloy, by casting in a sand mould, which was subsequently cold rolled and heat treated. Flores-Vélez et al.²⁸ produced a AMC based on the conventional powder metallurgy technique and obtained the best results of compressive strength and hardness for samples made with 10 wt.% EAFD to about 70 HV. Selvam et al.²⁹ applied fly ash in AA6061 aluminum alloy composites obtaining maximum value of 116 HV to 12 wt.% of waste. Other wastes have been applied as reinforcement

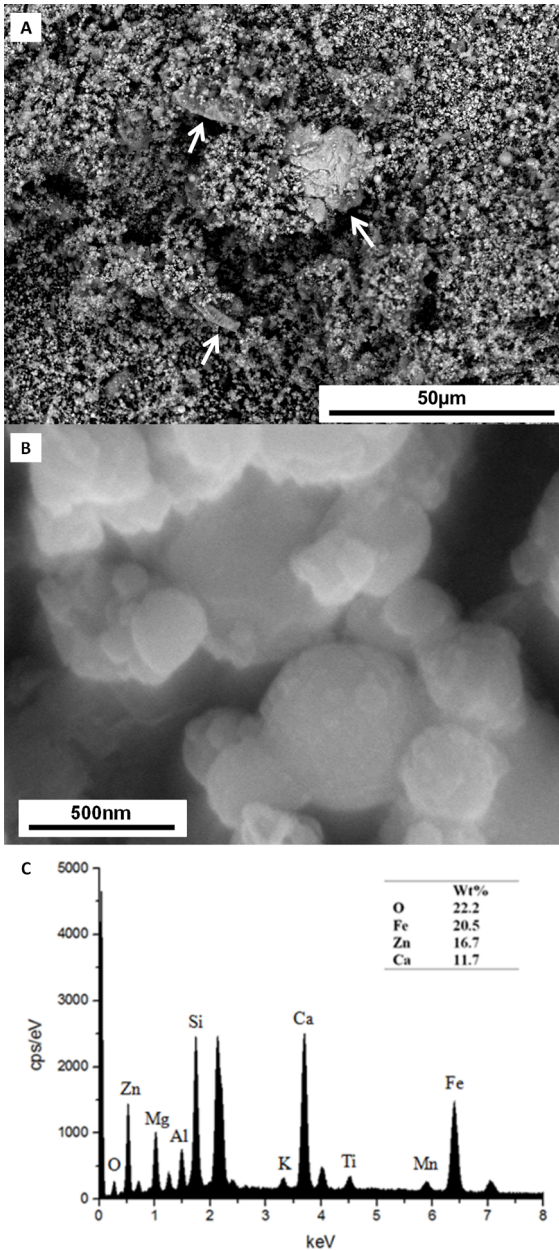


Figure 3. SEM images of EAFD (A), high-magnification SEM images of EAFD (B) and EDS spectra of EAFD particles (C).

aluminum composite, such as granulated slag²⁸, red mud³⁰, rice husk ash and graphite³¹. In addition, in the present work it was revealed that EAFD levels higher than 5% may cause fracture of the composite during the sintering.

Despite the low residue content, the mechanical behavior of the composite differs from that observed in the unreinforced aluminum alloy. The addition of waste in the aluminum matrix has resulted in the increase of mean instrumented Young's modulus (GPa) values, which are distributed widely. For the AA7075 alloy, the Young Modulus (E) are 35.51 ± 8.50 and 70.48 ± 8.36 for Al-5%EAFD. These

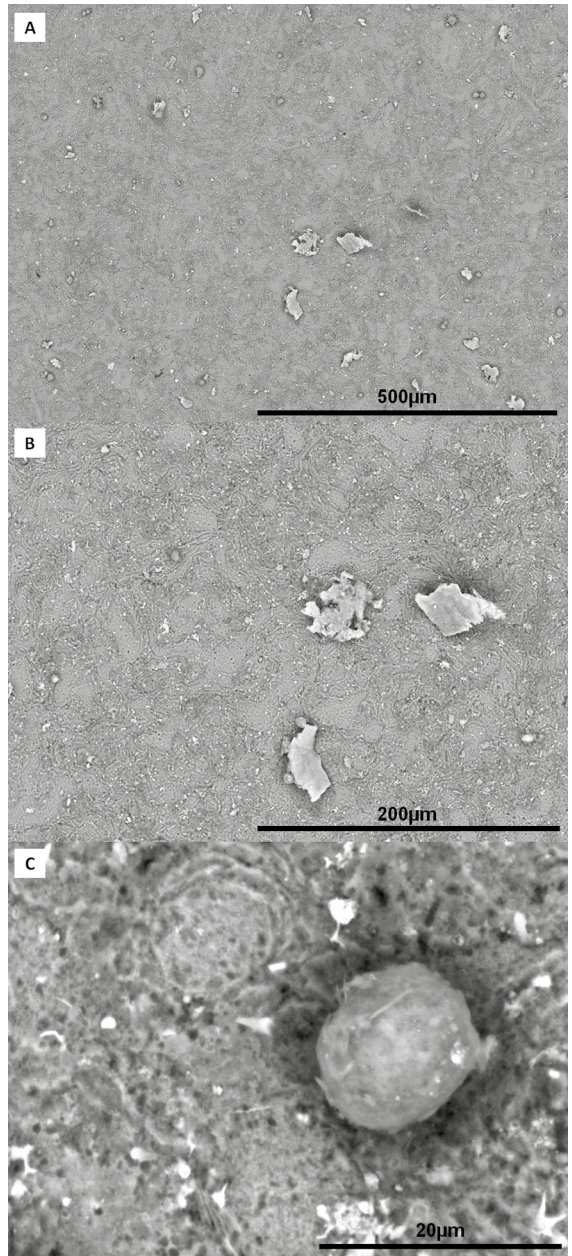


Figure 4. SEM images of Al-5%EAFD sintered composite with magnification of 200x (A); irregular particles of EAFD with magnification of 500x (B) and single agglomerated particle in metal matrix (C).

results corroborate with others found in literature^{32,33}. Figure 5 shows the Storage Modulus (GPa) data of AA7075 alloy and Al-5%EAFD composite as a function of temperature at 1Hz (quasi-static condition). The addition of EAFD into Al matrix has significant contribution to improve Young's modulus since the embedded reinforcement particles are stiffer than the matrix³⁴. Temperature increase (150-270°C) leads in a increase in Young's modulus values because foster precipitation hardening³⁴. The results obtained are similar with results of Young's modulus determined by UltraMicro

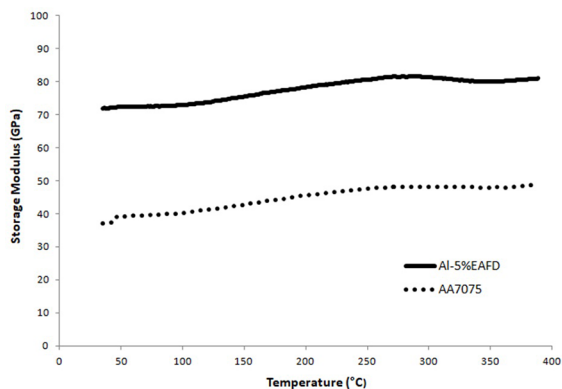


Figure 5. Storage modulus vs. temperature from DMA at frequency of 1 Hz, for Al-5%EAFD and AA7075 alloy.

Hardness tester at room temperature. The effect of temperature and other reinforcements are reported in literature^{25,34,35}.

The XRD patterns for Al-5%EAFD composite after 1-hour high-energy ball milling and sintered, and for AA7075 alloy as received, are shown in Figure 6. All the diffraction peaks from the Al matrix are clearly seen. An interesting feature viewed in the X-ray diffractograms is the decrease in the intensity of peaks after composite milling and sintering. This might be due to the reduction in the particle size. Fecht³⁶ reported that the initially sharp diffraction lines are considerably broadened after ball milling due to the refinement of the crystallite size and increase in atomic level strains. Due to the short time of high-energy ball milling assuming that effect don't occur. Formation Al-Fe or Al-Fe-Si intermetallic phases was not observed in the content of the waste blended in the composite.

4. Conclusion

EAFD particles smaller size than 53 μm were successfully added to aluminum matrix composites of up to 5%, fabricated through the conventional powder metallurgy technique. Morphology of the EAFD reveal the presence of smooth and irregularly-shaped particles, which can be observed in sintered composite. Addition of EAFD resulted in a significant increase (46%) in the hardness value and an increase in instrumented Young's modulus value in relation to the unreinforced alloy, under the experimental conditions implemented. The improvement observed can be due to the oxide particles introduced and the reinforcement particle distribution in the aluminum matrix. The addition of EAFD to Al matrix contributed to the improvement in storage modulus, as well, the temperature increase because foster precipitation hardening between 150-270°C. The DMA results obtained are similar with results of Young's modulus determined by UltraMicro Hardness tester at room temperature.

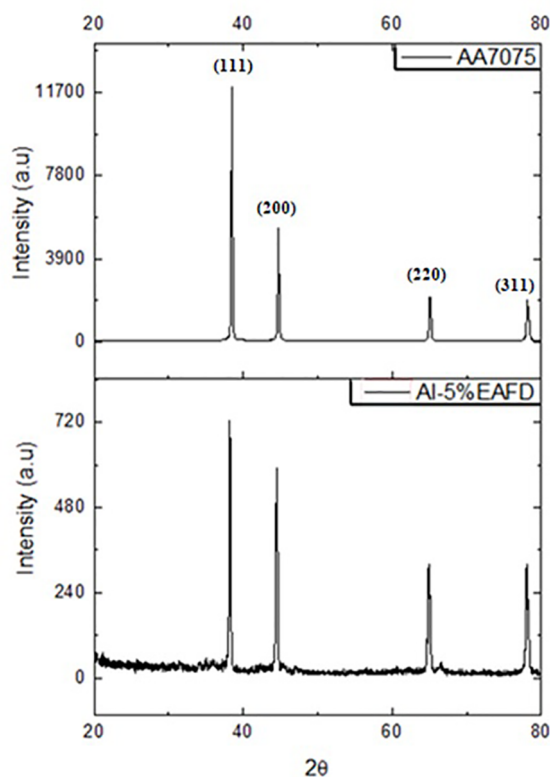


Figure 6. X-ray diffraction diffractogram of Al-5%EAFD sintered composite and AA7075 alloy as received.

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