

Thermal performance of nanoencapsulated phase change material in high molecular weight polystyrene[†]

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Abstract

A novel nanoencapsulation of n-hexadecane in high molecular weight polystyrene nanoparticles for thermal energy storage was carried out by miniemulsion polymerization using an iron-containing imidazolium-based ionic liquid (IL) as catalyst. The particle size, morphology, molecular weight, and thermal performance of nanoparticles containing the phase change material (PCM) were measured by dynamic light scattering, transmission electron microscopy, gel permeation chromatography, and differential scanning calorimetry, respectively. The nanoparticles were regular spherical, with narrow size distribution and particle size ranged from 138 nm to 158 nm. The enthalpy of melting for the nanoencapsulated PCM increased from 19 J/g to 72 J/g, as the content of n-hexadecane used increased from 20 wt% to 50 wt%. In addition, the nanoparticles showed thermal reversibility after 100 thermal cycles. The high molecular weights of the polymer, up to 1800 kDa, that could be reached with this IL may have contributed positively to this thermal behavior.

Keywords: cationic miniemulsion polymerization, ionic liquid, nanoencapsulation, phase change material.

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1. Introduction

The tireless search for new technologies to capture and store energy in more efficient, ecologically friendly, and low-cost ways has been boosting the development of phase change materials. Latent heat storage materials or simply phase change materials (PCMs) are capable of storing and releasing large amounts of energy during melting and solidification at specific temperatures^[1]. Due to its useful properties, such a higher energy storage densities, and phase change behaviors at almost constant temperatures^[2] they are widely developed for several applications especially for thermal comfort building, solar heating system, thermal protection, air-conditioning, thermal regulated textiles, electronic devices, among others.^[3]

Phase change materials are classified into two main categories, organic and inorganic. Most of the organic PCMs have low thermal conductivity and poor thermal response as well as being flammable, however, these limitations may be overcome by employing encapsulation technologies^[1]. The encapsulation provides protection of PCM from external

environmental influences, increases the surface of heat transfer, and the capsule shell helps to retain the changes in the volume that result from the phase change^[4].

Several techniques have been used to encapsulate PCMs such as in-situ polymerization which includes miniemulsion polymerization, complex coacervation, sol-gel methods, and solvent extraction/evaporation method, among others.^[1] Miniemulsion polymerization is a versatile technique for the in-situ encapsulation of different compounds into polymeric nanoparticles. In general, miniemulsions consist of small (50 – 500 nm)^[5], stable, and narrowly distributed monomer droplets in a continuous phase^[6]. The miniemulsion design involves the manipulation of the monomer-water interface stability and the dispersion mechanism (e.g., ultrasonic homogenizers^[7-9], rotor-stator systems^[7,9], high-pressure homogenizers^[8,10], static mixers^[7,11], or membranes^[9]). The high stability of the droplets is ensured by the combination of the surfactant, and the costabilizer, which retards the droplet diffusional degradation (Ostwald ripening). Typically,

highly hydrophobic and low molecular weight compounds, such as hexadecane, are used as costabilizer. The monomer nanodroplets become the polymerization locus, and in an idealized concept each of these nanodroplets can be regarded as a nanoreactor, without major secondary nucleation or monomer mass transport through the aqueous phase^[5]. Then, different compounds such as PCMs can be added into the monomer droplets before the miniemulsification step in the water phase, and followed by the polymerization leading to high encapsulation efficiencies.

Luo and Zhou^[12] employed the miniemulsion polymerization process to nanoencapsulate a paraffin ($T_m = 52\text{ °C}$) using a styrene-methacrylic acid copolymer. They noticed that the amount and type of surfactant, the concentration of comonomer (methacrylic acid), and the monomer/paraffin ratio, as well as the amount of the chain-transfer agent, and the nucleation mechanism did have a significant influence on the process.

de Cortazar and Rodríguez^[13] obtained stable nanocapsules containing up to 60 wt% of paraffin ($T_m = 60\text{ °C}$) by miniemulsion polymerization. The droplet nucleation was the main nucleation mechanism for the systems containing under 20 wt% of paraffin, for higher paraffin contents (40–60 wt%) secondary unwanted nucleation mechanism took place. Whereas systems with even higher paraffin concentration (80 wt%) showed phase separation.

Fang et al. published a series of works in which the encapsulation of n-octadecane^[14,15] ($\Delta H_m = 232.3\text{ J/g}$), n-tetradecane^[2] ($\Delta H_m = 220.6\text{ J/g}$) or n-dotriacontane^[16] ($\Delta H_m = 286\text{ J/g}$) in polystyrene nanoparticles (100–170 nm) via miniemulsion polymerization was evaluated. They used a monomer mixture of styrene and up to 2 wt% of acrylate (butyl, acrylic, or ethyl) to prepare the polystyrene shells. The latent heat of melting of nanoencapsulated PCM was 124.4 J/g for n-octadecane, 98.71 J/g for n-tetradecane, and 174.8 J/g for n-dotriacontane.

Fuentsanta et al.^[17] encapsulated a paraffin type phase change material (5–20 wt%), RT80 ($T_m = 85\text{ °C}$, $\Delta H_m = 161.1\text{ J/g}$) in a styrene-butyl acrylate copolymer shell. The particle sizes ranged from 52 nm to 112 nm with encapsulation efficiencies close to 80%, and good thermal reversibility after 200 thermal cycles. The thermal energy storage capacity of the nanoparticles was measured by DSC, obtaining melting and crystallization enthalpies in the range of 10–20 J/g.

Chen et al. produced nanocapsules (90–200 nm) containing n-dodecanol ($T_m = 27\text{ °C}$, $\Delta H_m = 222–248\text{ J/g}$) with poly(methyl methacrylate) (PMMA)^[18] or styrene-butyl acrylate copolymer^[19] as the shell materials by miniemulsion polymerization. The latent heat of nanocapsules was 98.8 J/g at 18.2 °C for the n-dodecanol/PMMA system and 109.2 J/g at 18.4 °C for the n-dodecanol/styrene-butyl acrylate system.

Many types of polymerization reaction mechanisms can be carried out in miniemulsion. Radical polymerization is the most common mechanism, but non-radical polymerization such as ionic polymerization, anionic^[20–22] and cationic^[23–26], can also be adapted to be carried out in miniemulsion.

Recently, Alves et al.^[26] showed that the iron-containing imidazolium-based ionic liquid 1-*n*-butyl-3-

methylimidazolium heptachlorodiferrate (BMI. Fe_2Cl_7), proposed by Rodrigues et al.^[27] as a new class of cationic catalysts for the styrene polymerization, is water tolerant for the miniemulsion polymerization. BMI. Fe_2Cl_7 not only proved to be an efficient initiator in aqueous medium as well as provided the synthesis of polystyrene nanoparticles (150–175 nm) of high molecular weights (M_v up to 2220 kDa), higher than those obtained with the commonly used initiators.

Combining the high molecular weights obtained in the cationic miniemulsion polymerization of styrene catalyzed by the ionic liquid BMI. Fe_2Cl_7 and the encapsulation feature that miniemulsion polymerization technique offers, this work aims to study the encapsulation of n-hexadecane as phase change material in high molecular weight polystyrene nanoparticles. It is expected that the n-hexadecane combines its properties as hydrophobic agent, retarding the droplet diffusional degradation (Ostwald ripening), and acts simultaneously as a phase change material storing and releasing energy during the melting and solidification process.

2. Materials and Methods

2.1 Materials

Technical grade monomer styrene (Sty, Innova) used to prepare the polymeric matrix was distilled under reduced pressure before use and stored at -4 °C. n-Hexadecane (HD, Sigma Aldrich) was used as costabilizer and as phase change material. Hexadecyltrimethylammonium bromide (CTAB, Sigma Aldrich) was used as surfactant. HD and CTAB were used as received. The ionic liquid 1-*n*-butyl-3-methylimidazolium heptachlorodiferrate (BMI. Fe_2Cl_7) used as catalyst was synthesized as reported by Rodrigues et al.^[27]. Deionized water was used in all reactions.

2.2 Synthesis of nPCM by miniemulsion polymerization

The nanoparticles containing the phase change material (nPCM) were synthesized by cationic miniemulsion polymerization using the ionic liquid BMI. Fe_2Cl_7 as catalyst according to the following procedures. The surfactant CTAB (60 mg) was dissolved in deionized water (11 mL). Different HD contents were evaluated, and the organic phase compositions (HD/Sty wt%/wt%) were 5/95, 20/80, 30/70, 35/65, 40/60, and 50/50. Regardless of composition, 3.3 g of organic phase was used in all reactions. The aqueous phase and the organic phase were magnetically stirred for 20 min (150 rpm) before being mixed and pre-emulsified under magnetic stirring (1500 rpm) for further 20 min. The coarse emulsion formed was sonicated for 1 min (10 s on/5 s off) at 70 % amplitude (Sonic Dismembrator Model 500 – Fisher Scientific, tip probe size of ½ in) using an ice-bath to avoid thermal polymerization. Then, an aliquot (1 mL) of a BMI. Fe_2Cl_7 aqueous solution, used at 1:1000 molar ratio of IL:Sty, was added to the miniemulsion. All the polymerizations were carried out for 8 h under nitrogen atmosphere at 85 °C. After polymerization, the latex was cooled down to room temperature. Right after the latex was demulsified with ethanol, and filtered to obtain the nanoparticles containing the PCM. The nanoparticles were washed three times with ethanol to remove the unencapsulated n-hexadecane and then were dried at room temperature.

Reactions were performed in duplicate. The monomer conversion was determined by gravimetry.

2.3 Characterization and performance of nPCM

2.3.1 Dynamic Light Scattering (DLS)

Intensity average sizes of monomer droplets and polymer nanoparticles were determined by Dynamic Light Scattering – DLS (Zetasizer Nano S, Malvern). Miniemulsion samples were diluted before measurement with a water solution saturated with styrene containing an amount of CTAB below the critical micelle concentration. Latex samples were diluted with deionized water.

2.3.2 Transmission Electron Microscopy (TEM)

Nanoparticles morphology was observed by Transmission Electron Microscopy with a JEM-1011 TEM (Jeol) electron microscope operating at an accelerating voltage of 100 kV. The diluted latex (0.01 wt%) was mounted on a carbon-coated copper grid and was left to dry at room temperature before analysis. Different regions were analyzed to assure representative images.

2.3.3 Differential Scanning Calorimetry (DSC)

Nanoparticles energy storage capacity was measured by Differential Scanning Calorimetry (Jade DSC, Perkin Elmer) under nitrogen atmosphere at a heating and cooling rate of 10 °C/min. The heating scans were from -10 °C to 30 °C, and the cooling scans were performed in the same range. The phase change latent heat (ΔH_m) and the melting temperature (T_m) were recorded from the second heating ramp.

2.3.4 PCM content in the nanoparticles

The PCM (n-hexadecane) content in the nanoparticles was calculated based on the enthalpy value of pure n-hexadecane by using the following Equation 1^[4,13,18]:

$$PCM \text{ content in nanoparticles (wt \%)} = \frac{\Delta H_{nPCM}}{\Delta H_{PCM}} \cdot 100 \quad (1)$$

where ΔH_{nPCM} is the enthalpy of melting for the nanoencapsulated PCM (J/g) and ΔH_{PCM} is the enthalpy of melting for the pure n-hexadecane (J/g).

2.3.5 Gel permeation chromatography (GPC)

Molecular weight distributions were determined by gel permeation chromatography. The analyses were conducted in a high-performance liquid chromatography instrument (HPLC, model LC 20-A, Shimadzu) equipped with a PLgel MiniMIX-C (PL1510-1500) guard column, and two in series PLgel MiniMIX-C (PL1510-5500) HPLC columns, from

Agilent. Tetrahydrofuran (THF) was used as eluent (flow rate of 0.3 mL/min) at 40 °C. The calibration was performed using polystyrene standards with molecular weights ranging from 580 g/mol to 9.225x10⁶ g/mol.

3. Results and Discussions

Polystyrene nanoparticles with different n-hexadecane contents were prepared by miniemulsion polymerization using the ionic liquid BMI₂Fe₂Cl₇ as catalyst. In all cases, highly stable latexes containing PCM nanoparticles have been obtained, and no phase separation was observed. Monomer conversion was higher than 80 % in all cases studied.

Dynamic light scattering (DLS) was used to analyze droplet (dd) and particle (dp) sizes of the miniemulsion and the latex, respectively. Table 1 summarizes the droplet (dd) and particle (dp) sizes before and after polymerization, respectively, and their dispersities (PdI).

PCM nanoparticles synthesized with different amounts of n-hexadecane ranged from 138 nm to 158 nm. According to the sizes obtained from droplets (miniemulsion, before polymerization) and polymeric particles (latex, after polymerization), droplet nucleation was the main nucleation mechanism and droplets/particles remained stable during polymerization. The miniemulsions produced with a higher amount of n-hexadecane had a slightly bigger droplet size. Increasing the amount of HD from 5 wt% to 50 wt%, droplet size increased from 131 nm (nPCM-5) to 168 nm (nPCM-50). This behavior was also observed by Shirin-Abadi et al.^[28] in the synthesis of poly(methyl methacrylate) nanocapsules containing n-hexadecane as PCM. They suggested that this behavior may be due to the decreasing of the contribution of the surfactant molecules per particle when the HD amount is increased using the same amount of surfactant. However, added to the contribution of the surfactant molecules per particle there is the viscosity effect. An increase in the amount of HD increases the viscosity of the organic phase, and since the ultrasonication conditions were kept constant for all HD contents used, i.e. the same amount of energy was supplied to the system to break the organic phase into smaller droplets, the higher viscosity may have led to the increase in the droplet sizes. In the systems containing amounts of HD greater than 5 wt% it was observed a decrease in the difference between droplet and particle size compared to the system containing 5 wt% (nPCM-5). The narrow particle size distributions, Figure 1, also suggest that the main nucleation mechanism was the droplet nucleation and that secondary nucleation (homogeneous or micellar nucleation) did not take place.

Table 1. Effect of the n-hexadecane (HD) content on the droplet size (dd) and particle size (dp) and their dispersities (PdI). (Average values refer to duplicate reactions).

Entry	HD/Sty	dd (nm)	PdI (-)	dp (nm)	PdI (-)
nPCM-5	5/95	131.2 ± 8.8	0.10 ± 0.01	158.4 ± 2.4	0.02 ± 0.01
nPCM-20	20/80	138.4 ± 4.8	0.10 ± 0.01	137.5 ± 2.4	0.04 ± 0.02
nPCM-30	30/70	141.4 ± 4.8	0.10 ± 0.01	142.3 ± 3.9	0.06 ± 0.02
nPCM-35	35/65	148.9 ± 1.3	0.11 ± 0.01	140.6 ± 2.1	0.06 ± 0.02
nPCM-40	40/60	154.4 ± 1.1	0.09 ± 0.01	139.7 ± 4.4	0.07 ± 0.01
nPCM-50	50/50	168.0 ± 9.2	0.16 ± 0.06	150.6 ± 1.5	0.08 ± 0.02

The morphology of nanoparticles containing the phase change material (nPCM) was observed by Transmission Electron Microscopy (TEM). The micrographs of the nPCM with 50 wt% of HD in relation to the organic phase (sample nPCM-50), Figure 2, show that the nanoparticles obtained have a spherical shape with a fairly uniform size distribution.

From the TEM images it is not possible to identify a capsule morphology (lighter core of HD surrounded by a darker PS shell), but according to Tiarks et al.^[29] for the same system HD/Sty due to the similarity of the particle sizes obtained by DLS (150.6 nm) and TEM, it is deduced that the HD is located within the particles, presumably in a spongelike morphology.

Molecular weight distributions were obtained using gel permeation chromatography. The weight-average molecular weight (Mw) and the molecular-weight dispersity (*D*) of the polystyrene nanoparticles are summarized in Table 2.

GPC data shows that the cationic miniemulsion polymerization of styrene catalyzed by the ionic liquid BMI₂Fe₂Cl₇ provided the synthesis of high molecular weight polystyrene nanoparticles. Under similar experimental conditions of the nPCM-5 system, 5 wt% of HD in relation to the organic phase, 1:1000 molar ratio of IL:Sty and at 85 °C, Alves et al.^[26] obtained average viscosity molar mass (M_v) up to 2150 kDa, at conversions slightly higher than 80 %. It is worth mentioning that the high amount of HD in the system (nPCM-20 to nPCM-50) did not affect the high molecular weights obtained which were similar to those achieved when the HD was used only as costabilizer (nPCM-5).

The molecular-weight dispersities (*D*) denote broad molecular weight distributions. Rodrigues et al.^[27] by on-line direct infusion electrospray mass spectrometry (ESI-MS(/MS)) detected three active catalytic species in the cationic polymerization of styrene catalyzed by the ionic liquid BMI₂Fe₂Cl₇. These three active catalytic species: (i) the chloronium cation (derived from styrene), (ii) the chloronium cation associated with [FeCl₄]⁻ and (iii) the chloronium cation associated with Cl⁻ have different reactivities. The presence of active catalytic species with different reactivities elucidates the broad molecular weight distributions obtained.

Thermal properties of the nPCM were measured by DSC. The DSC curves of the n-hexadecane nanoencapsulated in high molecular weight polystyrene polymeric matrix synthesized using different HD weight ratios in relation to the organic phase are presented in Figure 3 and summarized in Table 3. Also measured by DSC, the heat storage capacity of pure n-hexadecane, defined as the heat storage of melting (ΔH_m), and its melting temperature (T_m) were 290.35 J/g and 16.4 °C, respectively.

The enthalpy of melting for the nanoencapsulated PCM ranged from 19 J/g to 72 J/g, increasing as the amount of HD in the system also increased. By increasing the HD content in the organic phase, the values of melting enthalpy also increased due to the higher contribution of HD during the encapsulation process. Similar behavior was described by Shirin-Abadi et al.^[28] in the in situ encapsulation of n-hexadecane with poly(methyl methacrylate) shell by

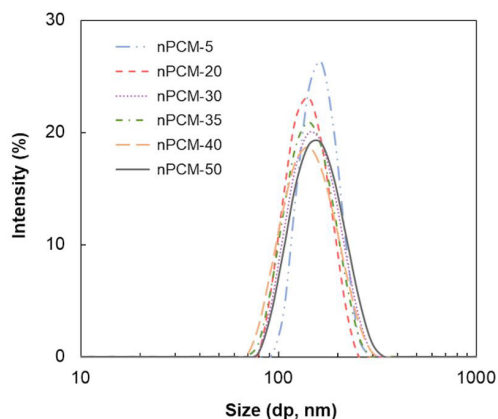


Figure 1. Particle size distribution of polymeric latexes synthesized with different HD/Sty weight ratios.

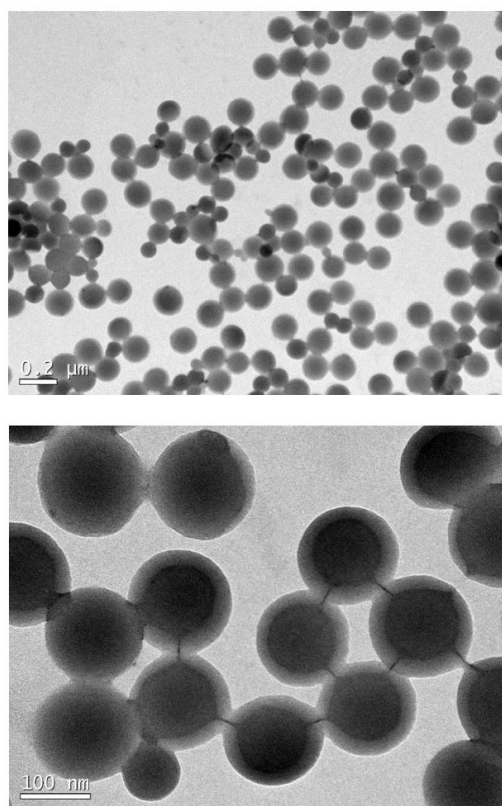


Figure 2. TEM micrographs of the polystyrene nanoparticles obtained in miniemulsion polymerization at 85 °C using an IL:Sty molar ratio of 1:1000 and 50 wt% of HD in relation to the organic phase, sample nPCM-50 (Table 1).

Table 2. Molecular weight of nPCM synthesized with different HD weight ratios in relation to the organic phase. (Average values refer to duplicate reactions).

Entry	HD/Sty	Mw (kDa)	<i>D</i> (-)
nPCM-5	5/95	1771 ± 24	2.1 ± 0.1
nPCM-20	20/80	1572 ± 43	2.7 ± 0.1
nPCM-30	30/70	1370 ± 19	2.9 ± 0.1
nPCM-35	35/65	1238 ± 20	3.0 ± 0.1
nPCM-40	40/60	1456 ± 26	2.7 ± 0.1
nPCM-50	50/50	1693 ± 228	2.2 ± 0.1

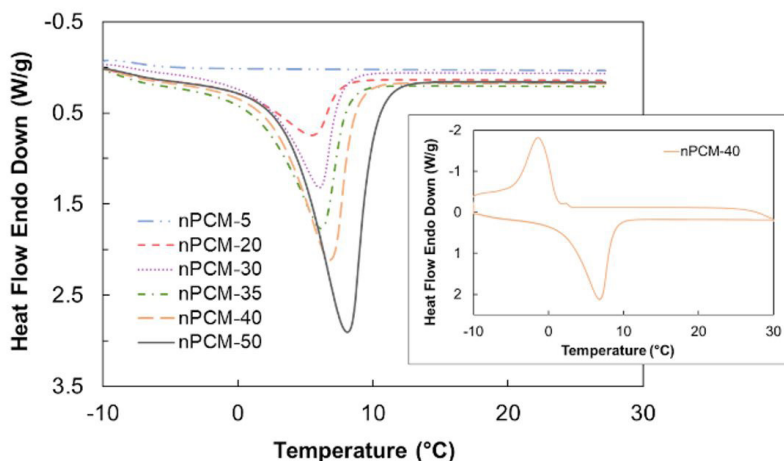


Figure 3. DSC curves of nPCM synthesized with different HD/Sty weight ratios.

Table 3. Thermal properties of nPCM synthesized with different HD/Sty weight ratios. (Average values refer to duplicate reactions).

Sample	HD/Sty	ΔH_m (J/g)	T_m (°C)	PCM content (wt%)
nPCM-5	5/95	-	-	-
nPCM-20	20/80	18.77 ± 0.78	5.67 ± 0.28	6.46
nPCM-30	30/70	27.30 ± 7.44	6.01 ± 0.01	9.40
nPCM-35	35/65	43.58 ± 0.07	6.60 ± 0.67	15.01
nPCM-40	40/60	50.85 ± 1.30	6.83 ± 0.12	17.51
nPCM-50	50/50	71.77 ± 5.01	7.92 ± 0.33	24.72

mini-emulsion polymerization. They obtained enthalpy of melting for the nanoencapsulated PCM ranging from 67 J/g to 86 J/g for HD:MMA weight ratios ranging from 1:3 to 3:1. Final PCM content in the nanoparticles ranged from 6.46 wt% to 24.72 wt% for an increase in HD content from 20 wt% (nPCM-20) to 50 wt% (nPCM-50) in relation to the organic phase.

The nPCM-5 that was synthesized with 5 wt% of HD did not exhibit enthalpy of melting. This absence of enthalpy denotes an unwanted plasticizing effect of the n-hexadecane on nanoparticles. By acting as plasticizer part of the HD is incorporated into the polymer matrix, reducing the amount of free HD that can act as PCM. The plasticizing effect significantly decreases the glass transition temperature of the polymer. To evaluate the plasticizing effect, the glass transition temperature of samples nPCM-5 and nPCM-50 was determined by DSC under nitrogen atmosphere at a heat rate of 10 °C/min in a temperature range from -10 °C to 140 °C. The glass transition temperature (T_g) was recorded from the second heating ramp. The T_g of sample nPCM-5 was 77.3 °C, while the T_g of sample nPCM-50 dropped to 54.5 °C.

The phase change temperature of the n-hexadecane contained in the nanoparticles was below the melting temperature of pure n-hexadecane (16.4 °C) and ranged from 5.67 °C to 7.92 °C when the HD content was increased from 20 wt% (nPCM-5) to 50 wt% (nPCM-50). Chen et al.^[18,19] also observed a reduction in the phase change temperature of the n-dodecanol contained in the nanocapsules in relation to the pure n-dodecanol ($T_m = 27$ °C) used as PCM. The nanocapsules containing n-dodecanol as the core and poly(methyl methacrylate) (PMMA)^[18] or styrene-butyl acrylate copolymer^[19] as the shell showed a

Table 4. Thermal properties of the nPCM synthesized with 40 wt% of HD in relation to the organic phase (nPCM-40) after different numbers of thermal cycles.

Thermal cycles	ΔH_m (J/g)	T_m (°C)
0	49.66	6.91
20	49.84	6.92
40	50.34	6.92
60	50.11	6.92
80	50.15	6.92
100	50.16	6.91

phase change temperature of 18.2 °C and 18.4 °C, respectively. Fang et al.^[2] reported a slight difference in the melting points between the core material and the nanocapsules. The melting temperature for pure n-tetradecane (Tet) used as PCM was 6.94 °C, whereas the change phase temperature for the Tet/PS nanocapsules was 4.04 °C. They suggested that this difference may be related to the slower heat conduction rate of the polymer shell that delays the phase change process of the nanocapsules.

The thermal reversibility of the nPCM was investigated until 100 thermal cycles (heat/cooling) carried out by using DSC. The thermal properties of nPCM-40 after different numbers of thermal cycles are summarized in Table 4.

The phase change latent heat of sample nPCM-40 remained virtually constant after 100 cycles, ranging from 49 to 50 J/g. Thus, the nanoencapsulated n-hexadecane nanoparticles show thermal reversibility after 100 thermal cycles. The melting temperature also remained constant at 6.9 °C.

4. Conclusions

The cationic miniemulsion polymerization of styrene catalyzed by the ionic liquid $\text{BMI.Fe}_2\text{Cl}_7$ has proved to be a convenient one-step encapsulation technique to encapsulate a phase change material into high molecular weight polymeric nanoparticles. Particle size analyses (DLS) and TEM micrographs indicated that polymeric particles are spherical, with narrow size distribution and particle size ranging from 138 nm to 158 nm. In addition, these analyses also denoted that the HD is located within the particles, presumably in a spongelike morphology.

The presence of HD in the nanoparticles was confirmed by the DSC analyses that have shown the enthalpy of melting for the nanoencapsulated n-hexadecane. The enthalpy of melting increased from 19 J/g to 72 J/g, as the content of n-hexadecane used increased from 20 wt% to 50 wt% in relation to the organic phase. DSC analyses also supported the thermal reversibility of nanoparticles after 100 thermal cycles. DLS, TEM, and DSC results also indicated that the HD acted simultaneously as hydrophobic agent and as phase change material.

As expected, the low ionic liquid concentration (1:1000 molar ratio of IL:Sty) used as catalyst led to high molecular weights, up to 1800 kDa, even in the presence of high amounts of HD. These high molecular weights may have contributed positively to the thermal behavior of the nanoparticles.

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