

Ballistic Tests of Alumina-UHMWPE Composites Submitted to Gamma Radiation

André Ben-Hur da Silva Figueiredo^{a*}, Hélio de Carvalho Vital^b, Ricardo Pondé Weber^a, Édio Pereira Lima Júnior^a, João Gabriel Passos Rodrigues^a, Letícia dos Santos Aguilera^a, Ronaldo Sergio de Biasi^a

^aSeção de Engenharia de Materiais, Instituto Militar de Engenharia, Rio de Janeiro, RJ, Brasil

^bCentro Tecnológico do Exército, Departamento de Irradiações, Rio de Janeiro, RJ, Brasil

Received: March 19, 2019; Revised: May 09, 2019; Accepted: June 10, 2019

The energy absorption in ballistic tests of alumina-ultra high molecular weight polyethylene (UHMWPE) composites with 60, 80 and 90 wt% alumina submitted to gamma radiation doses of 25, 50 and 75 kGy was investigated. The ballistic tests were carried out at subsonic speed using a compressed air system. The results showed that the composite with 80% alumina irradiated with 50 kGy yields the best ballistic results. Gel content, Differential scanning calorimetry (DSC) and X-ray diffraction (XRD) results showed that this composite is the one with the highest concentration of crosslinks and the lowest volume fraction of amorphous UHMWPE. Scanning electron microscopy (SEM) images of the same composite showed a high pullout, suggesting that gamma irradiation increases the adhesion between alumina and UHMWPE.

Keywords: Ballistic shielding, ballistic impact, alumina-UHMWPE composite, gamma irradiation.

1. Introduction

In the beginning of the 21st century, despite the absence of international conflicts on the scale of the two World Wars, local and regional conflicts involving different tribes, ethnic groups, militias, gangs and heavily armed drug dealers have become a serious threat in several parts of the world.

Standard bulletproof personal protection vests use aramid fabric as a single layer of shielding. This protection is limited to relatively low impacts, up to 9 mm ammunition. Protection against high impact projectiles requires a multi-agent system (MAS) ¹. A conventional MAS has, beside the aramid fabric, a ceramic front layer which absorbs most of the impact energy, eroding the projectile tip. However, such protection increases the cost and compromises the soldier mobility due a significant increase of vest weight. In addition, the front layer may be fragmented by the first impact, compromising its resistance to subsequent shots.

The main ceramic materials used for ballistic protection are alumina (Al₂O₃), silicon carbide (SiC) and boron carbide (B₄C). Alumina has been suggested for ballistic protection due to good physical and chemical properties. However, the low flexural strength and low fracture toughness mean that the use of pure alumina for ballistic protection may lead to catastrophic failure. Moreover, the high density, about 4 g/cm³, limits its use in applications where weight is crucial, such as bulletproof vests ^{2,3}.

According to Figueiredo et al. ⁴, alumina-UHMWPE composites may yield a good compromise between high energy absorption and low density, the only problem being low adhesion between the alumina particles and the polymer matrix ⁵. The adhesion may be increased by exposure of the composites to gamma radiation ^{6,7}, that has other favorable effects, such as increased stiffness and stability, as reported by Shafiq. et al. ⁸.

On the other hand, according to Hobbs et al. ⁹, high radiation doses generate microcracks that weaken the alumina component.

The purpose of this work was to investigate the properties of gamma irradiated alumina-UHMWPE in order to determine the best combination of alumina concentration and radiation dose for ballistic protection applications. The UHMWPE is used to decrease the density and increase the flexural strength, making the shield more suitable for personal protection and avoiding fracture after the first shot ^{10,11}. There is also an economic factor involved, since the composite is prepared at a relatively low temperature, 230 °C, while pure alumina must be prepared sintering alumina powder at high temperatures, of the order of 1400 °C, a more expensive procedure ¹²⁻¹⁵.

2. Materials and methods

2.1 Materials

The materials used were 60 Mesh Alundum powder with 9.25 Mohs hardness (Fisher Scientific) and UHMWPE Mipelon PM-200 powder 10 μm to 30 μm in diameter (Mitsui Chemicals).

2.2 Sample preparation

Composites with different alumina-UHMWPE mass ratios were prepared by mechanical mixing for 10 min and labeled A00/00, A00/25, A00/50, A00/75, A60/00, A60/25, A60/50, A60/75, A80/00, A80/25, A80/50, A80/75, A90/00, A90/25, A90/50 and A90/75, where the first number is the alumina (A) mass concentration in percent (00 is a pure UHMWPE sample used for comparison). The second number is the radiation dose in kGy (00 is non-irradiated sample).

*E-mail: abenhur@ime.cb.br

The samples were produced in the shape of discs 5 mm thick and 51 mm in diameter. The discs were pressed at 230 °C for 10 min under a force of 90 kN and kept in cast aluminum forms.

Gamma Irradiation of the composites was performed using a Gammacell 220 Excel irradiator with a Co-60 source.

2.3 Ballistic tests

For the ballistic tests, an air rifle Gunpower SSS was used with a noise suppressor Padrão Armas. The projectile was a 22 gauge lead shot with an estimated mass of 3.3 g. An Air Chrony ballistic chronograph model MK3, with a precision of 0.15 m/s, was used to measure the impact speed, and a ProChrono ballistic chronograph model Pal, with a precision of 0.31 m/s, was used to measure the residual speed.

The air rifle was positioned 5 m away from the target, consisting of the composite disk attached in an aluminum frame, secured by a bench vise and aligned perpendicularly to the rifle. One ballistic chronograph was positioned 10 cm from the nozzle of the noise suppressor and another was placed 10 cm behind the target.

The energy absorbed by the target was calculated using the equation

$$E_{abs} = m_p (v_i^2 - v_r^2) / 2 \quad (1)$$

where m_p is the projectile mass, v_i is the impact speed and v_r is the residual speed¹⁶.

Taking into account the fact that in the case of personal protection weight may be an important factor, one may define a figure of merit given by the following equation:

$$FM = E_{abs} / m_c \quad (2)$$

where m_c is the composite mass.

2.4 Gel content measurements

The gel content shows the amount of crosslinking in the samples. It was determined according to ASTM D2765 using a Soxhlet device¹⁵. The samples were extracted using boiling xylene for 6 h and were subsequently washed with acetone and dried at 140 °C. The percent gel content was determined using the following expression:

$$gel\ content = (W_1 / W_0) 100\% \quad (3)$$

where W_0 and W_1 are, respectively, the weight of sample before and after extraction.

2.5 Differential scanning calorimetry

Differential scanning calorimetry (DSC) analysis was performed with a Netzch DSC 404 F1 Pegasus calorimeter. The measurements were carried out with a heating and cooling rate of 10 °C min⁻¹ under nitrogen (50 mL min⁻¹).

The samples were heated from room temperature to 180 °C and cooled to 50 °C and then reheated to 180 °C. The melting temperature and the percent crystallinity was determined only for the first run. The percent crystallinity X_c was calculated using the following expression:

$$X_c = (\Delta H_o / \Delta H_m) 100\% \quad (4)$$

where ΔH_o and ΔH_m are, respectively, the enthalpy of melting of crystalline polyethylene (290 Jg⁻¹) and the enthalpy of melting of the sample.

2.6 X-ray diffraction

The sintered samples were characterized by X-ray Diffraction (XRD) in an X'PERT PRO PANalytical diffractometer, with monochromatic radiation (Cu K α , $\lambda = 1.5406$ Å), step size of 0.05° s⁻¹, time per step 150 s and 2 θ between 10° and 90°. The XRD patterns were refined using the Rietveld method, with the help of the TOPAS Academic version 4.1 software.

2.7 SEM images

After the ballistic tests, images of composites with 80% alumina were taken in a FEI Quanta FEG 250 SEM.

3. Results and discussion

3.1 Ballistic tests

All shots completely penetrated the disks. Two shots were made in each experiment and eight experiments were performed for each composition. Representative samples are shown in Figure 1 after the first shot.

No significant decrease in energy absorption was observed after the first shot.

Since the test was performed with the samples in the cast aluminum holders, lateral movement was suppressed, reducing the damage to the composite^{17,18}.

Table 1 shows the average values of the composite mass (m_c), projectile mass (m_p), average impact speed (v_i), average residual speed (v_r), absorption energy (E_{abs}) and merit factor (FM) for each composition. As expected, m_c increases with increasing alumina concentration.

Figure 2 shows the FM of composites with different alumina concentrations and radiation doses. One can see that the best compromise between high energy absorption and low weight was displayed by the composite with 80% alumina irradiated with 50 kGy.

3.2 Gel content measurements

Table 2 shows that the gel content of UHMWPE, non-irradiated and gamma irradiated with doses of 25 kGy, 50 kGy and 75 kGy. The highest gel content was for the samples irradiated with a dose of 50 kGy, suggesting the concentration of crosslinks is maximum for this radiation dose.

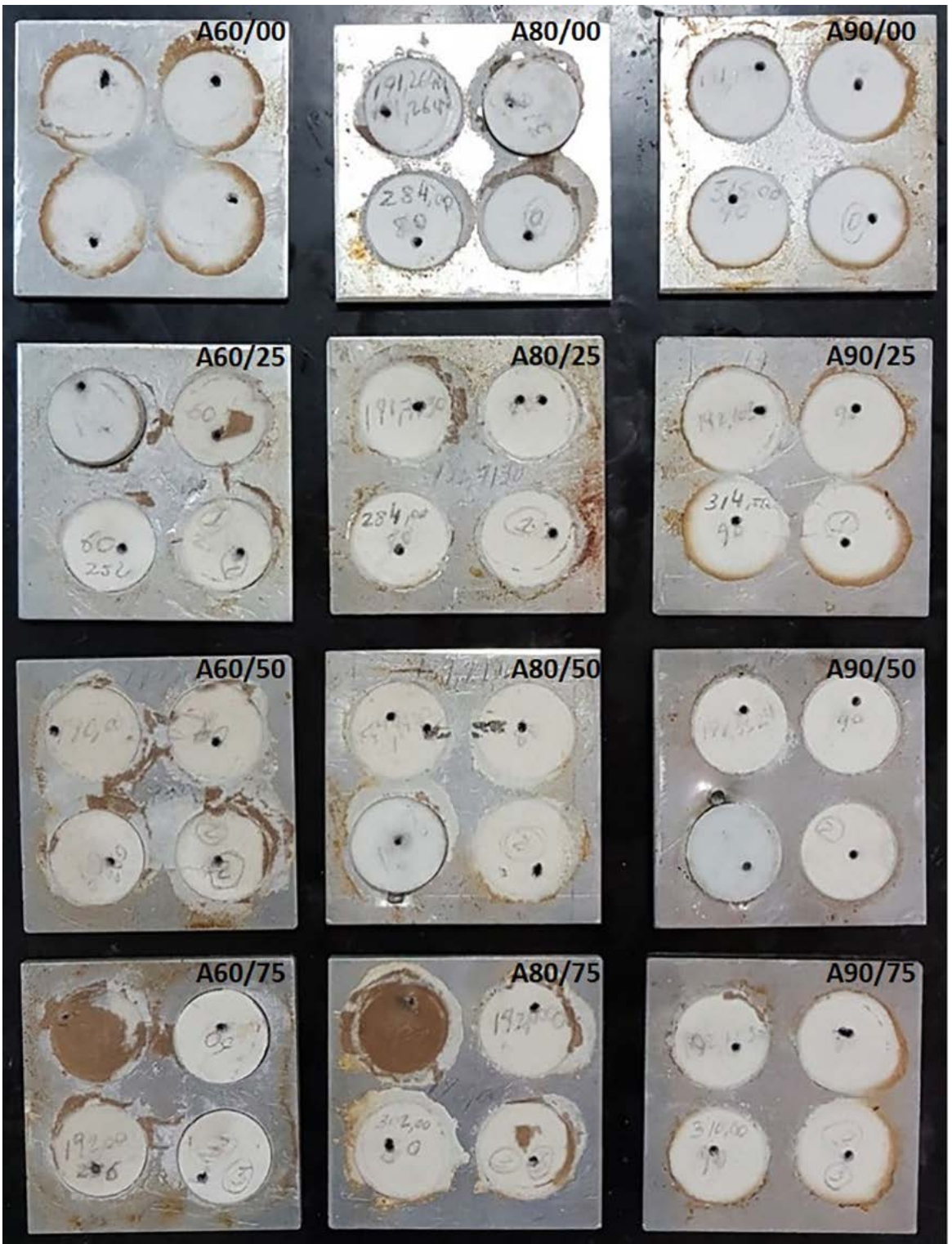
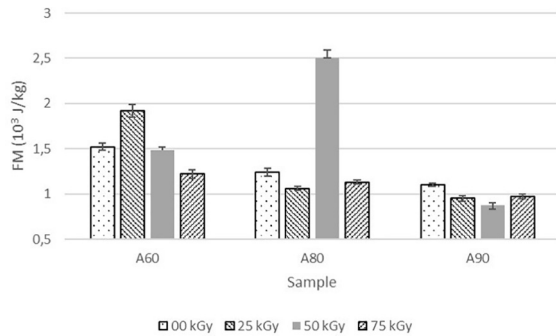


Figure 1. A60/00, A60/25, A60/50, A60/75, A80/00, A80/25, A80/50, A80/75, A90/00, A90/25, A90/50 and A90/75 samples after the first shot, arranged in top-down rows.

Table 1. Average results of the ballistic tests.

COMPOSITE	m_c (g)	m_p (g)	V_i (m/s)	V_r (m/s)	E_{abs} (J)	FM (10^3 J/kg)
A60/00	16.50 ± 0.10	3.29 ± 0.23	251.51 ± 2.59	219.15 ± 2.91	25.01 ± 0.86	1.52 ± 0,04
A60/25	15.50 ± 0.09	3.27 ± 0.24	250.91 ± 2.68	211.53 ± 1.45	29.73 ± 1.30	1.92 ± 0,07
A60/50	17.50 ± 0.12	3.25 ± 0.12	250.98 ± 2.31	216.79 ± 2.74	25.89 ± 0.86	1.48 ± 0,04
A60/75	16.00 ± 0.08	3.27 ± 0.14	251.24 ± 2.57	226.24 ± 2.66	19.44 ± 0.83	1.22 ± 0,05
A80/00	23,18 ± 0.11	3.25 ± 0.13	250.13 ± 2.22	211.91 ± 2.26	28.67 ± 1.04	1.24 ± 0,04
A80/25	25.07 ± 0.13	3.30 ± 0.25	247.66 ± 2.59	212.52 ± 2.63	26.60 ± 0.68	1.06 ± 0,02
A80/50	26.02 ± 0.08	3.27 ± 0.20	248.87 ± 2.66	112.93 ± 8.12	64.96 ± 2.51	2.50 ± 0,09
A80/75	27.50 ± 0.11	3.24 ± 0.21	253.71 ± 2.57	212.52 ± 2.93	31.01 ± 0.70	1.13 ± 0,02
A90/00	31.20 ± 0.08	3.29 ± 0.18	248.39 ± 2.44	202.01 ± 2.75	34.28 ± 0.72	1.10 ± 0,02
A90/25	30.47 ± 0.15	3.29 ± 0.28	248.03 ± 2.93	209.70 ± 2.29	28.92 ± 0.95	0.95 ± 0,03
A90/50	30.57 ± 0.13	3.27 ± 0.18	251.44 ± 2.58	216.71 ± 3.08	26.47 ± 0.89	0.87 ± 0,03
A90/75	29.33 ± 0.09	3.31 ± 0.24	246.53 ± 2.06	208.71 ± 2.74	28.44 ± 0.83	0.97 ± 0,03

**Figure 2.** Merit factor of composites with 60%, 80% and 90% alumina concentration, non-irradiated and irradiated with 25 kGy, 50 kGy and 75 kGy.**Table 2.** Gel content of UHMWPE, non-irradiated and gamma irradiated with doses of 25 kGy, 50 kGy and 75 kGy.

Sample	Gel content (%)
A00/00	91.20
A00/25	97.37
A00/50	100.00
A00/75	99.69

The decrease of gel content for a higher radiation dose is attributed to degradation of UHMWPE, forming free radicals.

3.3 DSC

The DSC analyses were performed during the first fusion and crystallization. Table 3 shows that the samples irradiated with 50 kGy had the highest values of crystallinity, and among them the largest value was that of the sample 80/50.

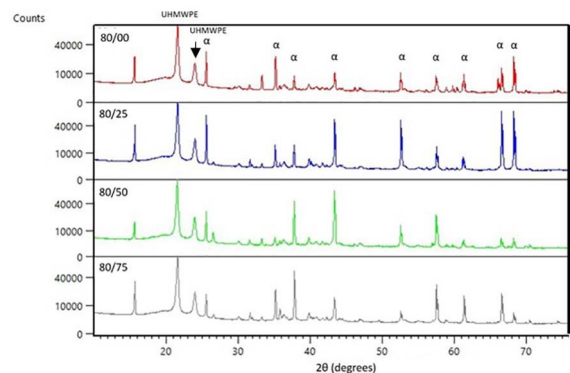
3.4 X-ray diffraction

The samples with 80% alumina were investigated further due to the good ballistic performance displayed by the sample with 80% alumina irradiated with 50 kGy. The purpose was to determine why the ballistic performance increased with radiation dose up to a radiation dose of 50 kGy, but decreased when the radiation dose was increased to 75 kGy.

Table 3. DSC average results.

Sample	T_{onset} (°C)	T_{max} (°C)	T_{end} (°C)	ΔH_{fusion} (J/g)	Crystallinity (%)
A60/00	128.8	139.4	145.7	53.02	45
A60/25	127.9	139.7	146.3	68.97	59
A60/50	128.6	142.9	152.0	103.10	88
A60/75	128.7	141.4	148.7	77.91	66
A80/00	128.5	140.4	147.5	37.97	65
A80/25	130.3	141.7	148.6	44.36	76
A80/50	129.2	142.1	148.1	56.21	96
A80/75	129.6	140.9	147.4	45.54	78
A90/00	126.8	136.4	143.2	15.17	52
A90/25	130.0	138.5	144.5	21.30	73
A90/50	129.4	139.1	145.6	24.86	85
A90/75	130.9	140.7	147.2	23.38	80

Figure 3 shows the XRD patterns of samples A80/00, A80/25, A80/50 and A80/75. Except for a broad line at small angles attributed to amorphous UHMWPE, only the diffraction lines of α -alumina and UHMWPE are seen, showing that there was no phase transformation due to gamma irradiation.

**Figure 3.** XRD patterns of A80/00, A80/25, A80/50, A80/75 samples.

In Figure 4, one can see that the amplitude of the broad line goes through a minimum for a radiation dose of 50 kGy. This suggests that the crystallinity of UHMWPE increases with increasing radiation dose up to a radiation dose of 50 kGy, but decreases for larger radiation doses.

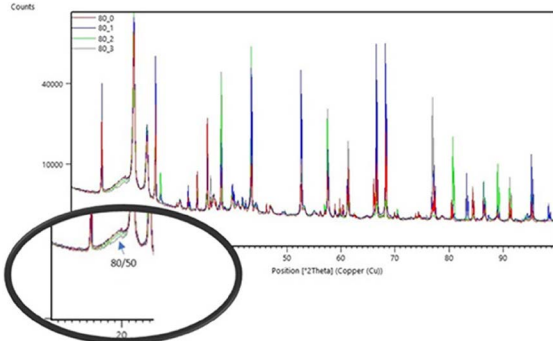


Figure 4. XRD Superimposed patterns of A80/00, A80/25, A80/50, A80/75 samples.

Figure 5 shows amplified views of parts of the diffraction patterns displayed in Figure 4. In the case of the UHMWPE line, there is a significant displacement to the left for the sample submitted to a radiation dose of 50 kGy. This is attributed to a relaxation due to increasing crystallinity and is consistent with the decrease in the amplitude of the broad line in Figure 4. In the case of the alumina lines, there is a shift to the right in the samples irradiated with 25 and 75 kGy. In the first case, the compression may be due to loss of water and in the second to radiation-induced microcracks.

3.5 Images of alumina powder, sample A80/00 and sample A80/50

Figure 6 is an SEM Image of the alumina powder, showing the irregular shape of the grains, which improves the stiffness of the composites.

When impact occurs, compressive shock waves propagate along the thickness of the sample and are responsible for

several cracks whose interaction creates subsequently a conical damage zone with ductile and fragile failure, similar to that observed in ceramic tiles¹⁹. Figure 7 shows the conical damage zone in the distal face of an A80/50 sample.

Figure 8 shows microcracking on the distal faces of A80 and A80/50 samples after the first impact. The main difference between Figures 8a and 8b is that there is evidence for pullout in the irradiated samples as a consequence of increased adhesion between alumina and UHMWPE, as also shown in Figure 9.

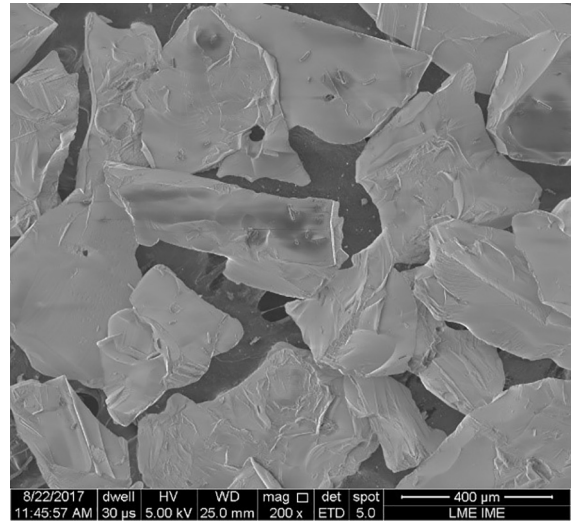


Figure 6. SEM image of the alumina powder.

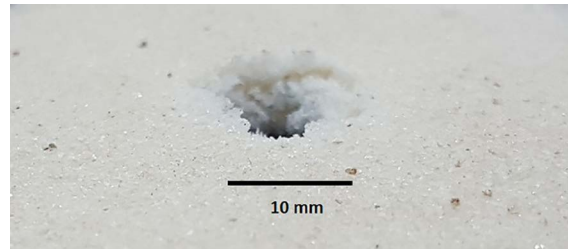


Figure 7. Optical image showing the conical damage zone on the distal face of an A80/50 sample.

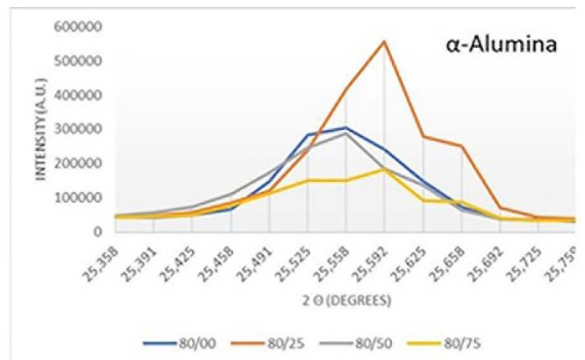
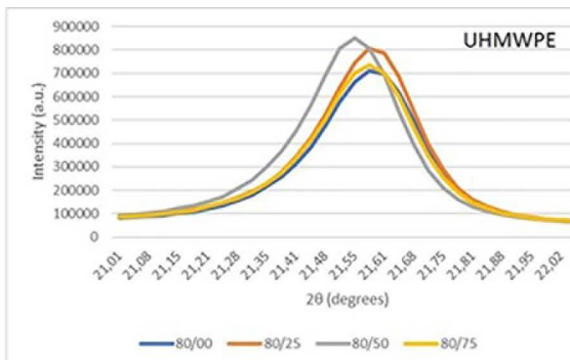


Figure 5. Selected XRD lines of UHMWPE and α -alumina.

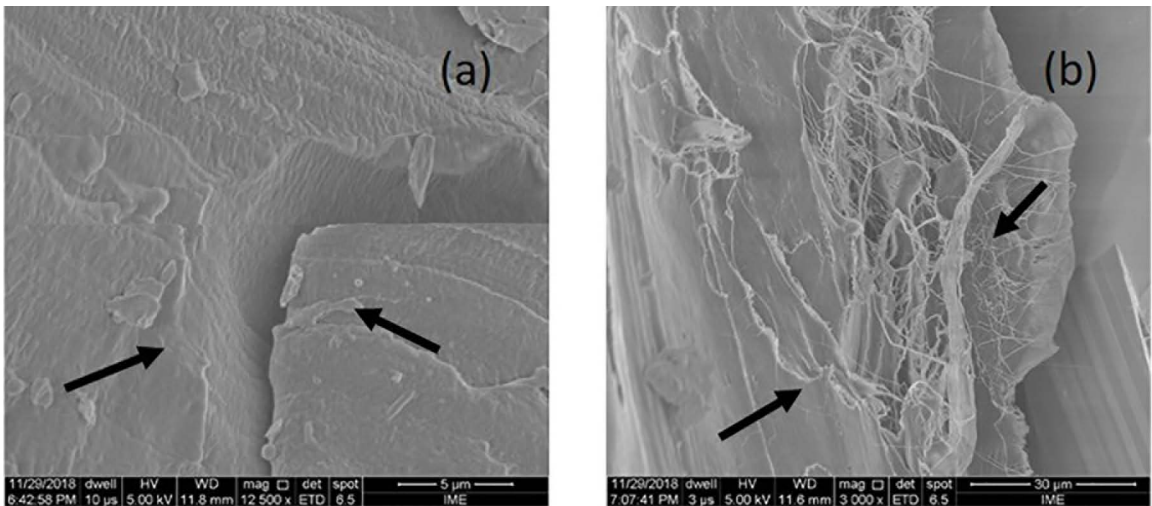


Figure 8. SEM images, after the first impact, of the distal face (a) of an A80/00 sample showing the microcracking without pullout and (b) of an A80/50 sample showing microcracking with pullout.

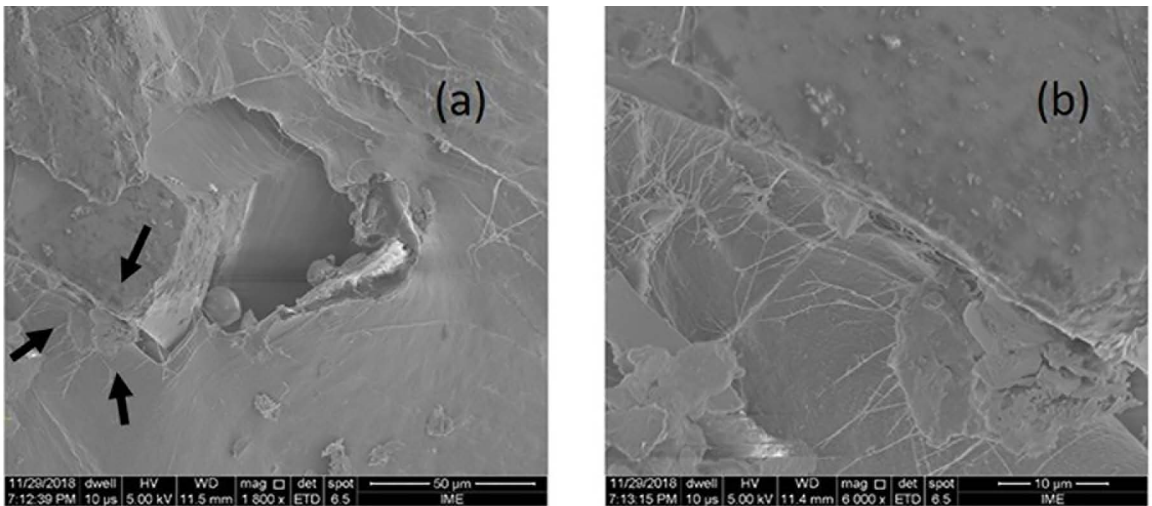


Figure 9. SEM images, after the first impact, (a) of the distal face of an A80/50 sample, showing the region of contact between alumina and UHMWPE, and (b) enlargement of the same region, showing UHMWPE pullout.

4. Conclusions

Ballistic tests were performed on gamma-irradiated alumina-UHMWPE composites. The composite with 80% alumina submitted to a radiation dose of 50 kGy was the one that had the best ballistic properties. Gel content, DSC, XRD and SEM results suggest that this is due to three effects of gamma radiation: an increased extent of crosslinking, a decrease of the volume fraction of amorphous UHMWPE and an increased adhesion between alumina and UHMWPE. The results also showed that increasing the radiation dose above 50 kGy has adverse effects, probably due to the production of a large number of microcracks that leads to a weakening of the alumina particles and to UHMWPE degradation, generating free radicals.

5. References

1. da Luz FS, Lima Junior EP, Louro LHL, Monteiro SN. Ballistic Test of Multilayered Armor with Intermediate Epoxy Composite Reinforced with Jute Fabric. *Materials Research*. 2015;18(Suppl 2):170-177.
2. Cavallaro PV. *NUWC-NPT Technical Report 12,057 - Soft Body Armor: An Overview of Materials, Manufacturing, Testing, and Ballistic Impact Dynamic*. Newport: Naval Undersea Warfare Center Division; 2011.
3. Carlucci DE, Jacobson SS. *Ballistics: Theory and Design of Guns and Ammunition*. Boca Raton: CRC Press; 2008. 496 p.
4. Figueiredo ABS, Lima Júnior EP, Gomes AV, Melo GBM, Monteiro SN, de Biasi RS. Response to Ballistic Impact of Alumina-UHMWPE Composites. *Materials Research*. 2018;21(5):e20170959.

5. Madhu V, Ramanjaneyulu K, Bhat TB, Gupta NK. An experimental study of penetration resistance of ceramic armour subjected to projectile impact. *International Journal of Impact Engineering*. 2005;32(1-4):337-350.
6. Neves JC, Silva GG, Mendes MWD, Bressiani AH, Bressiani JC, Garcia FG. Efeito da irradiação gama nas propriedades mecânicas e térmicas de redes DGEBA/amina cicloalifática com potencial para aplicações médicas. *Polímeros*. 2013;23(6):814-822.
7. Lima IS, Araújo ES. Efeitos da radiação gama na estrutura e nas propriedades do poliestireno. In: *4th Meeting on Nuclear Applications*; 1997 Aug 18-22; Poços de Caldas, MG, Brazil. Rio de Janeiro: Associação Brasileira de Polímeros (ABPol); 1997. p. 137-139.
8. Shafiq M, Mehmood MS, Yasin T. On the structural and physicochemical properties of gamma irradiated UHMWPE/silane hybrid. *Materials Chemistry and Physics*. 2013;143(1):425-433.
9. Hobbs LW, Clinard FW Jr., Zinkle SJ, Ewing RC. Radiation effects in ceramics. *Journal of Nuclear Materials*. 1994;216:291-321.
10. Coutinho FMB, Mello IL, de Santa Maria LC. Polietileno: Principais Tipos, Propriedades e Aplicações. *Polímeros*. 2003;13(1):1-13.
11. Senatov FS, Gorshenkov MV, Tcherdyntsev VV, Kaloshkin SD, Sudarchikov VA. Fractographic analysis of composites based on ultra high molecular weight polyethylene. *Composites Part B: Engineering*. 2014;56:869-875.
12. Medvedovski E. Ballistic performance of armor ceramics: Influence of design and structure. Part I. *Ceramics International*. 2013;36(7):2103-2115.
13. Medvedovski E. Ballistic performance of armor ceramics: Influence of design and structure. Part II. *Ceramics International*. 2013;36(7):2117-2127.
14. Medvedovski E. Lightweight ceramic composite armor system. *Advances in Applied Ceramics*. 2006;105(5):241-245.
15. Spiegelberg SH. Characterization of Physical, Chemical, and Mechanical Properties of UHMWPE. In: Kurtz SM, ed. *UHMWPE Biomaterials Handbook*. 2nd ed. San Diego: Academic Press; 2009. p. 355-368.
16. Azevedo G, Aragão JCT. *Apontamentos sobre balística*. Rio de Janeiro; 2010.
17. Sherman D. Impact failure mechanisms in alumina tiles on finite thickness support and the effect of confinement. *International Journal of Impact Engineering*. 2000;24(3):313-328.
18. Bittencourt BA, Ellwanger MV, Nascimento WA, Belchior LF, Araújo EM, Melo TJA. Moldagem por compressão a frio do polietileno de ultra alto peso molecular. Parte 1: Influência do tamanho, distribuição e morfologia da partícula na densidade a verde. *Polímeros*. 2009;19(3):224-230.
19. Bresciani LM, Manes A, Giglio M. An analytical model for ballistic impacts against ceramic tiles. *Ceramics International*. 2018;44(17):21249-21261.