

## The Oxidation Mechanism of CdTe (110) Surface

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Using a first-principles pseudopotential technique within a generalized gradient approximation of the density functional theory, we have investigated the mechanism of adsorption of molecular oxygen on the CdTe(110) surface. The determination of the more favorable structures and their formation through the activation barrier analysis indicates that the formation can occur in the regimes of low and high temperatures. According to our calculations it was verified that in the regime of low temperatures the oxygen molecule binds exclusively to Cd at the topmost site through Cd-O-O or Cd-O<sub>2</sub> bonds. The electronic band structure presents a characteristic state like a double acceptor defect. In the high temperature regime the molecule adsorbs between Cd of the surface and Te of the subsurface, breaking Cd-Te bond and causing an upward dislocation of Cd from the subsurface. In this regime of high temperatures the complex presents Cd-O, Cd-O<sub>2</sub>, Te-O and O-O bonds and the electronic band structure is almost free of gap of states.

Keywords: Oxidation mechanism; First-principles pseudopotential technique; CdTe(110)

### I. INTRODUCTION

It is well known that surface cleaning, oxidation and any other chemical or physical process brought about on the surface, determine the character and density of surface states. Thus, such a process influences the electronic properties of the surface and of the structure grown on it. Optoelectronic devices based on II-VI semiconductors attracted recently a considerable interest to the investigations of the surface electronic structure of these materials as well controlled preparation of insulating oxides, essential for the development of device technology. CdTe, as a prototype of zinc-blend II-VI crystals, is one of the most intensively studied material, its surface properties were also thoroughly investigated and a quite complete scheme of electronic structure was determined for clean surfaces of the crystal. However, the processes occurring on various surfaces of CdTe during oxidation, passivation, chemical and plasma etching are still investigated by means of many experimental techniques. For most device applications, the surface of CdTe must be properly passivated. If we consider solid state radiation detectors, for example, among the factors that affect the performance at room temperature of such devices, surface leakage current is one of the most detrimental. This limiting factor can be significantly reduced by chemical oxidation of CdTe surface, enhancing their performance [1]. Although the calculated ternary phase diagram indicates that CdTeO<sub>3</sub> and not TeO<sub>2</sub> is an equilibrium oxide on CdTe, some early experimental studies showed that oxides formed on the CdTe surface exposed to air at room temperature were essentially TeO<sub>2</sub> with little or undetectable formation of CdO [2, 3]. Other works [4, 5] indicate a small amount or even a major formation of CdTeO<sub>3</sub> and suggest that TeO<sub>2</sub> is a metastable phase on CdTe surfaces exposed to air at room temperature [5]. However, in a recent experimental work by Fritsche *et al* [6] it is observed in their X-ray photoelectron spectroscopy (XPS) spectra that after exposed to air, the CdTe surface is severely oxidized with the formation of TeO<sub>2</sub> and CdO. On the other hand, because of the known peculiarities of the phase diagram, CdTe bulk crystals grown by different

techniques have excess Te than the stoichiometric composition which aggregates as Te precipitates. Here we report selective mechanisms for the O<sub>2</sub> molecules adsorption on the CdTe(110) surface based on first-principles total-energy calculations with a temperature-dependent scheme.

### II. THEORETICAL CONSIDERATIONS

The calculations were performed in the framework of the density-functional theory within pseudopotential scheme. The electron-electron exchange correlation was considered by using the generalized-gradient approximation (GGA) [7]. The pseudopotentials for Cd, Te, O and H were derived by using the scheme of Troullier and Martins [8]. For the Cd specie, the 4d electrons were treated as core state but its influence on valence electrons was taken into account by applying the non linear core correction (NLCC) [9]. The surface was modelled in a super-cell geometry, with an atomic slab of six Cd-Te layers and a vacuum region equivalent to six atomic layers. The theoretical lattice constant of 6.47 Å obtained for the bulk CdTe is used in surface calculations. On the top side of the slab we placed the oxygen molecules in different configurations, and the back surface was passivated by H atoms arranged in a dihydride structure. We used four Monkhorst-Pack special **k** points for the Brillouin-some summation. The single particle orbitals were expressed in a plane-wave basis set up to the kinetic-energy cutoff of 35 Ry. The electronic and ionic degrees of freedom were relaxed by adopting the scheme described by Bockstedte *et al* [10]. The atoms were assumed to be in their fully relaxed positions when the forces acting on the ions were smaller than 0.005 eV/Å. The adsorption energies for the oxygen molecules was calculated as  $E_{conf}^{ads} = -[E_{conf} - E_{ref}]$ , where  $E_{conf}$  is the total energy for the considered configuration,  $E_{ref}$  is the total energy for the free CdTe(110) surface plus the total energy for a free oxygen molecule in a box of side 20 a.u.. The pathways from the gas-phase molecule to the adsorbed states were obtained by using a constraint dynamics scheme in which atomic move-

ment along the pathway is a response to forces in a local region around it. The constraint is set only for the oxygen molecule during the adsorption process. Considering the phenomenological approach in the form of the Arrhenius equation [11, 12] we have estimated the adsorption barriers.

### III. RESULTS

For bulk CdTe our first-principles calculations give 6.47 Å for the equilibrium lattice constant ( $a_o$ ), 0.40 Mbar for the bulk modulus ( $BM$ ), and 1.32 eV for the energy band gap. Our structural parameters are in good agreement with the experimental values of  $a_o = 6.48$  Å, and  $BM = 0.59$  Mbar [13]. They are also in agreement with the substitution profile observed for the energy band gap typical of supercell-DFT calculations, as the experimental value of  $E_g$  is 1.48 eV. The clean CdTe(110) surface show a well-established relaxation pattern: at the surface layer the anion moves away from the bulk in favor of an  $s^2p^3$  bonding with three neighbouring cations, resulting in a pyramidal geometry. The cation on the surface moves into the bulk in favour of an  $sp^2$  bonding with three neighbouring cations, resulting in a planar geometry. The tilt angle for our calculated CdTe(110) surface is found to be  $30.0^\circ$ , while the vertical buckling of the top layer is 0.11 Å, in good agreement with the experimental values of  $30.5^\circ$  and 0.12 Å obtained via LEED analysis. The electronic band structure for the free CdTe(110) surface compare well with the available angle-resolved photoemission results. We have also investigated the structural parameter for the  $O_2$  gas molecule. We calculated the O-O bond length of 1.20 Å and an enthalpy of formation of 218 kcal/mol, also in good agreement with the experimental values of 1.21 Å and 205 kcal/mol, respectively [13]. In a previous work we verified that  $O_2$  molecules could be formed over the surface with atomic adsorption [14].

We have considered many possibilities for the adsorption of  $O_2$  molecules on the CdTe(110) surface but only a few structures could be found has the more favorables considering the calculated adsorption energies. These structures with the adsorbed molecule present bonds between Cd and the molecule and, in some cases, also bonding to Te. In order to analyse the possibility of  $O_2$  molecule to bond only to Te, we verified that the molecule on that site was desorbed from the surface or its relaxed structure was energetically unfavorable. Our calculations indicate that the oxygen molecule adsorbs preferentially over Cd surface-sites forming Cd- $O_2$  and Cd-O-O bonds and between Cd from the first layer and Te of the second layer forming an oxygen complex composed of Cd-O, Cd- $O_2$ , Te-O and O-O bonds. Both types of structures preserve the Cd-Te surface-bond but in the second case, the bond between Cd from the surface and Te of the second layer is broken, in order to fix the molecule, with a dislocation upward of Cd from the second layer.

For the structures that adsorbs oxygen molecules over the surface, Cd-O bond length is around 1.90 Å, 14% smaller than the sum of their covalent radii (1.21 Å) and O-O bond length is of 1.23 Å, 2% bigger than in free molecule. At the surface, the Cd-Te bond length is of 2.79 Å, only 0.3% smaller than

this crystal distance and 3% bigger than that of free surface. The Cd-Te tilt angle at surface is now reduced to less than  $6^\circ$ . From these parameters we can verify that the adsorption of  $O_2$  molecule at these structures restore the surface to crystal configuration. The adsorption energies calculated for these structures are between 7.1 and 8.4 kcal/mol with a difference of 1.3 kcal/mol. As the difference is of the some order of the precision in this kind of theoretical calculations, we find three possible structures with the same characteristic above described. Fig.1 shows the surface electronic band structure for these configurations. The expected passivation effect of the oxygen adsorbed system does not occur but there is a double acceptor like state localized around 10 meV above the top of the valence band.

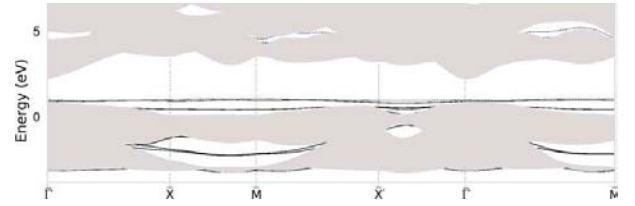


FIG. 1. The electronic band structure for the adsorption of  $O_2$  molecules on CdTe(110) at low temperatures regime. The projected bulk spectrum corresponds to the hatched region. The calculated localized surface electronic states are shown by dashed lines.

For the second kind of adsorption it was found only one structure that favour the  $O_2$  adsorption between the surface and the second layer. On that configuration both oxygen atoms bond to the surface Cd with a bond length of 1.90 Å and one of the oxygen atoms bond to the subsurface Te with a bond length of 2.03 Å. The distance between the molecule atoms is 1.32 Å, 10% bigger than the free molecule distance. Te atom from the subsurface moves 0.60 Å downward and Cd of this layer moves 1.23 Å, in order to bond one oxygen. Cd-Te bond between the surface and subsurface is broken but the surface atoms now have almost the same positions as the free surface (tilt angle of  $31.1^\circ$ ) but with a upward Cd-Te displacement of 0.60 Å. The calculated electronic band structure for this structure, shown in Fig.2, reveal a band gap almost free of surface states and a good passivation around the top of the valence band. The adsorption energy for that structure is found as 25.3 kcal/mol indicating that it is around 17 kcal/mol more favorable than the first kind of structures with the  $O_2$  adsorption at the surface. Comparing our results for both kind of structures we can infer that the complexes CdTe $O_3$  and Te $O_2$  verified on the experimental results are only related to the Te excess at the surface.

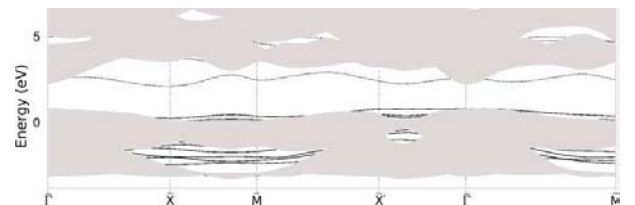


FIG. 2. The electronic band structure for the adsorption of  $O_2$  molecules on CdTe(110) at high temperatures regime. The projected bulk spectrum corresponds to the hatched region.

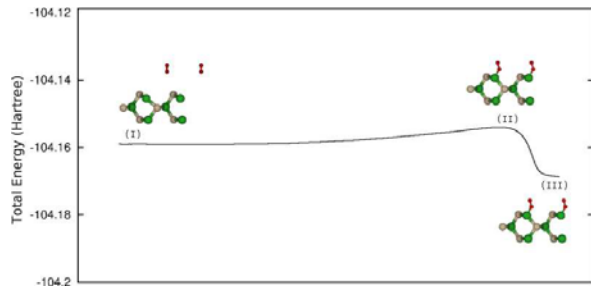


FIG. 3. Calculated energy barrier diagram for the  $O_2$  adsorption on the CdTe(110) surface at low temperatures regime. The structures formed during the adsorption process are shown.

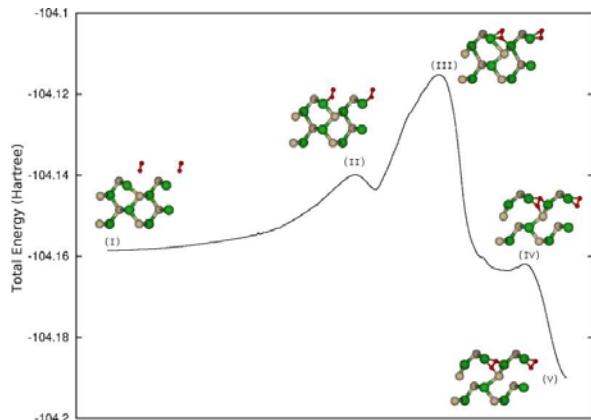


FIG. 4. Calculated energy barrier diagram for the  $O_2$  adsorption on the CdTe(110) surface at high temperatures regime. The structures formed during the adsorption process are shown.

Using a constraint dynamics scheme with pathways from the gas-phase molecule to the adsorbed structure and considering the phenomenological approach in the form of the Arrhenius equation, we have stimulated, for complete adsorption and with no zero-point energy correction, the activation barriers. It is worthy emphasizing that is well known that the use of

pseudopotentials in non-local density functional theory calculations is responsible for an underestimation of the calculated activation energies.

The energetics related to the process of  $O_2$  adsorption at the surface is schematically represented in Fig.3. The activation barriers are between 2.4 and 7.8 kcal/mol corresponding to a thermal activation energy between 40 and 130 K, suggesting a low temperatures regime for this kind of adsorption. Fig.4 represents the process to the molecule adsorption between the surface and subsurface layers. The activation barrier for this mechanism is of 25.5 kcal/mol with a thermal activation energy of 430 K, suggesting a high temperatures regime.

#### IV. SUMMARY

First-principles pseudopotential calculations for the formation of  $O_2$  complexes on CdTe(110) surfaces are presented. Our findings could identify two possible kind of structures associated to a two temperature regimes. At low temperatures the adsorption occur at the surface layer with Cd-O-O and Cd- $O_2$  bonds, the surface becomes flat and the associated electronic band structure present a state like double acceptor defect. At more high temperatures there is a formation of a complex, between the surface and subsurface, with Cd-O, Cd- $O_2$ , Te-O and O-O bonds. The related electronic band structure associated to this complex is almost free of surface states.

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