

Molybdenum carbide doped with nanostructured nickel for application in degradation of reactive dyes

(Carbeto de molibdênio dopado com níquel nanoestruturado para aplicação na degradação de corantes reativos)

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Abstract

Carbide-type ceramic materials such as nickel-doped molybdenum carbide have promising photocatalytic degradation activity. The addition of nickel to the molybdenum carbide aims to increase its reaction yield and also its characteristics. The objective of this paper was to study the photocatalytic activity of Mo₂C and 5% and 10% nickel-doped Mo₂C materials used for the degradation of the Maxilon Blue GRL 300 textile dye. The structural properties were characterized by X-ray diffraction and scanning electron microscopy. For the determination of band gap energy, diffuse reflectance spectroscopy was used. The samples of the photocatalysis tests were analyzed by UV-visible spectroscopy. The best reaction yield was observed for the 5% nickel-doped molybdenum carbide sample in photocatalytic tests, reaching a yield of around 92% in the final concentration of the dye solution.

Keywords: molybdenum carbide, nickel, band gap, degradation.

Resumo

Materiais cerâmicos do tipo carbeto, como carbeto de molibdênio dopado com níquel, têm promissora atividade de degradação fotocatalítica. A adição de níquel ao carbeto de molibdênio visa aumentar seu rendimento de reação e também suas características. Este trabalho teve por objetivo estudar a atividade fotocatalítica dos materiais Mo₂C e Mo₂C dopado com 5% e 10% de níquel para a degradação do corante têxtil Maxilon Blue GRL 300. As propriedades estruturais foram caracterizadas por difração de raios X e microscopia eletrônica de varredura. Para a determinação da energia de band gap, foi utilizada espectroscopia de refletância difusa. As amostras dos testes de fotocatalise foram analisadas por espectroscopia UV-visível. O melhor rendimento de reação foi observado para a amostra de carboneto de molibdênio dopado com 5% de níquel em testes fotocatalíticos, atingindo um rendimento em torno de 92% na concentração final da solução de corante.

Palavras-chaves: carbeto de molibdênio, níquel, band gap, degradação.


INTRODUCTION

In recent years, transition metal carbides have received considerable attention due to their high melting point, mechanical resistance, and thermal stability [1]. Furthermore, considering the electronic structure presented by these materials, transition metal carbides demonstrate catalytic activity in hydrogenolysis, hydrodeoxygenation, isomerization, and hydrogenation, helping to break H-H bonds [1, 2]. In this sense, some studies suggest the addition of Ni in molybdenum carbide (Mo₂C) to modify the activity, selectivity, and formation of the carbide phase [3].

There are several methods for synthesizing transition metal carbides [4], for instance: carbothermal reduction of metal oxides with graphite in an inert environment

[5], electrochemical synthesis, graphite melting, thermal decomposition of oxometallic diethylenetriamine compound, oxide reduction by H₂, and a mixture of H₂ and hydrocarbons as cementing gas or other carbon sources [6]. One aspect of these processes is that the activation of carbon in the environment is thermodynamically favorable for the formation of carbide phases [7]. The production of molybdenum and nickel carbide from nickel and molybdenum oxides was studied using the programmed carburization method using a CH₄/H₂ gas mixture as a carbon source [8]. The results suggest that the addition of nickel decreased the carburization temperature with increasing surface area and significantly increased the catalytic activity in dibenzothiophene hydrodesulfurization [8]. In a study [9] with molybdenum and potassium-doped nickel carbide in the steam reforming of ethanol to produce hydrogen, the carbide was produced at a controlled temperature of 996 °C in a reactor with a flow of 20% CH₄ and 80% H₂ gas mixture

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for the carburization of the precursors. The composition of $\text{Ni}_{24}\text{Mo}_{74}\text{K}_2$ carbide carburized at 1071 °C was the most active in the production of hydrogen. The addition of potassium to the catalyst was effective for the cleavage of the O-H bond [9].

Over the past decades, nanometric materials have been widely used for high-efficiency photocatalytic applications. Heterogeneous photocatalysis is a process that involves the degradation of organic compounds through the formation of hydroxyl radicals on the surface of semiconductor materials by irradiation of sunlight or artificial light [10, 11]. The photocatalytic behavior of TiO_2 supported by Mo_2C in vapor phase decomposition of formic acid was studied, where Mo_2C prepared with TiO_2 increased the extent of photocatalytic decomposition exhibited by pure TiO_2 , with dehydrogenation and dehydration reactions occurring at the same time [12, 13]. The highly efficient photocatalytic production of H_2 with TiO_2 using $\text{Ni}(\text{NO}_3)_2$ as an additive in aqueous methanol solution has been reported [14]. In this system, the Ni^{2+}/Ni reduction potential ($E^0 = -0.23$ V) is higher than the TiO_2 conduction band (-0.26 V) but lower than the H^+/H_2 reduction potential ($E^0 = 0.00$ V). This energy level structure favors the transfer of photogenerated electrons from the TiO_2 to Ni^{2+} conduction band and the reduction of Ni^{2+} to Ni^0 , which could assist in the charge separation and serve as a co-catalyst for H_2 generation. As a result, Ni-charged TiO_2 obtained under optimal conditions could achieve an H_2 generation rate as high as 2547 $\mu\text{mol/h}$ [14, 15].

In this paper, the potential of photocatalytic degradation of Mo_2C and $\text{Mo}_2\text{C-Ni}$ through the degradation of the Maxilon Blue GRL 300 dye from the textile industry was studied. The addition of Ni to the Mo_2C catalyst improved its photocatalytic performance.

EXPERIMENTAL

Pure Mo_2C and Ni-doped Mo_2C with concentrations of 5% and 10% by weight of nickel were produced. The preparation of the precursor material and the production of carbides occurred in two stages: i) preparation of the precursor powder via wet synthesis; and ii) reduction/carburization by gas/solid reaction of the material in a fixed bed reactor with a gas flow of H_2 and CH_4 . *Synthesis of precursors:* ammonium molybdate (CRQ, 99%) and nickel nitrate hexahydrate (Vetec, 99%) were used as precursors. The nickel nitrate reagent was weighed to the proportions of 5% and 10% by mass of atomic nickel. The reactants were mixed in a beaker with 42 mL of deionized water at 80 °C with magnetic stirring [16]. After mixing of the reactants and complete evaporation of water, the resulting mixture was dried in a muffle furnace at 70 °C for 7 h [17]. *Synthesis of the carbide material:* the precursor material was synthesized in the fixed bed horizontal reactor at 700 °C during 2 h under a flow of 12.5 L/h of CH_4 and 237.5 L/h of H_2 at atmospheric pressure. Before reaction, the reactor was outgassed with argon gas with a flow of 10 mL/min in

order to remove contaminants. In the next step, the material was submitted to a heating rate of 5 °C/min [17, 18]. The gas space velocity (GSV) for the methane gas was 337.0 h^{-1} , and hydrogen gas was 6401.6 h^{-1} .

Material characterization: the structural properties were determined with an X-ray diffractometer (D8 Advance Davinci, Bruker) with $\text{CuK}\alpha$ source. From the X-ray diffraction (XRD) patterns, it was possible to infer the size of the crystals of the powder. To study the morphology of the samples, a powder metallization was performed (K550x, Emitec), and the micrographs were obtained by scanning electron microscopy (SEM, LEO1430, Zeiss). The study of band gap energy of the carbide was performed using a UV-vis-NIR spectrophotometer (5G, Varian Cary) with an accessory for diffuse reflectance measurement. Reflectance spectra were obtained in the range of 200 to 800 nm at room temperature, using barium sulfate (BaSO_4) as the reference sample for reflectance measurements. *Preparation of the dye solution:* Maxilon Blue GRL 300 dye solution was prepared by weighing and dissolving the adequate mass in distilled water in order to achieve a final solution concentration of 50 ppm. The pH of the solution was adjusted from 9.0 to 2.0 by the addition of concentrated HCl. *Photocatalysis reaction:* this reaction was performed in a homemade device consisting of a wooden box isolated from light. The reactor was a 25 mL volume transparent tube of borosilicate glass having a jacket for temperature control interconnected with a thermostatic bath. In a typical procedure, the 20 mL mixture of the dye solution with a mass of approximately 5 mg of carbide was irradiated by three 150 W tungsten lamps separated at 120° to each other and a distance of 12 cm between the lamp and the reactor. The reaction was kept at 45 °C for 60 min. At the end of the reaction, the resulting solution was centrifuged for 12 min at 10000 rpm in order to separate the catalytic solid from the dye solution; then the liquid was analyzed with the UV-vis-NIR spectrophotometer [18] to determine the absorbance value of the sample [19].

RESULTS AND DISCUSSION

Structural characterization: carbide-like materials were produced, replicated and analyzed by X-ray diffraction (Fig. 1) using similar parameters employed in a previous study [16]. It was possible to identify the characteristic peaks of molybdenum carbide with orthorhombic structure (ICSD 18205) [2] on both Mo_2C and Ni-doped Mo_2C (5% and 10%) samples, featuring only Mo_2C peaks. The peak at $2\theta \sim 43.4^\circ$ was associated with the (111) plane of free carbon (PDF 00-006-0675). The absence of the Ni peaks in the XRD patterns can be associated with the low dopant concentration and high grain boundaries [20-22]. In particular, it was reported that the Co oxide peaks emerge in the XRD pattern for systems in which the Co oxide concentration is higher than 40% in the ZnO matrix [22, 23]. For Mo_2C , according to the literature [3, 24-26], this catalyst has several phases with different conductivities; therefore, it can have an uneven distribution of surface charge. From the results of XRD

analysis, the crystallite size (D_{hkl}) was estimated using the Scherrer equation (Eq. A) and the Halder-Wagner-Langford (HWL) method [27], considering normal and integral methods for the determination of peak width.

$$D_{hkl} = \frac{k \cdot \lambda}{\beta \cdot \cos \theta} \quad (A)$$

where k is the shape coefficient of the reciprocal network point (0.9-1.0), λ is the wavelength of the radiation, β is the full width at half maximum (FWHM), and θ is the angle of diffraction. Table I shows the crystallite size for the studied samples. The Scherrer model does not consider the micro-tension in the sample structure. On the other hand, the HWL model [27] is more comprehensive in terms of defects existing in the crystalline structures and approximates the true value of crystallite size more reliably. Comparing the values of crystallite size by the HWL and Scherrer methods for all studied samples, it was observed that there was an increase from the sample without doping in comparison with the Ni-doped samples, by using both methods. In particular, for the $\text{Mo}_2\text{C-Ni5\%}$ sample, there was a variation in the crystallite size value calculated by the normal HWL method;

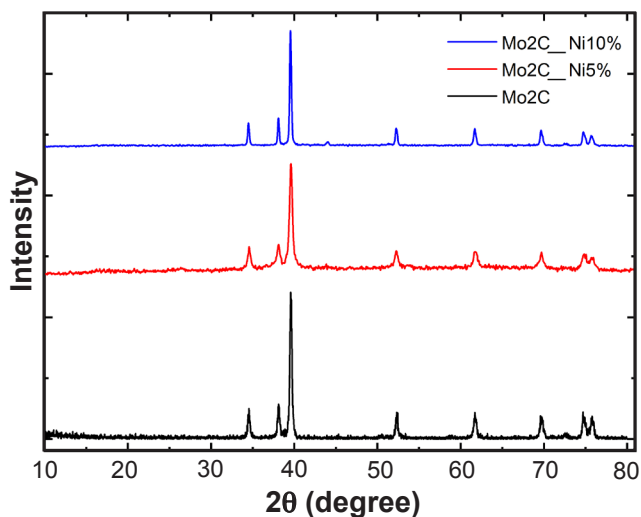


Figure 1: XRD patterns for undoped Mo_2C and Ni-doped Mo_2C samples.

[Figura 1: Padrões de DRX das amostras de Mo_2C e Mo_2C dopadas com Ni.]

Table I - Crystallite size calculated from the Scherrer equation and HWL method.

[Tabela I - Tamanho de cristalito calculado a partir da equação de Scherrer e método HWL.]

Sample	Normal		Integral	
	HWL (nm)	Sch. (nm)	HWL (nm)	Sch. (nm)
Mo_2C	34.7	28.9	23.5	19.1
$\text{Mo}_2\text{C-Ni5\%}$	43.7	32.4	27.3	24.1
$\text{Mo}_2\text{C-Ni10\%}$	37.5	36.6	38.0	37.2

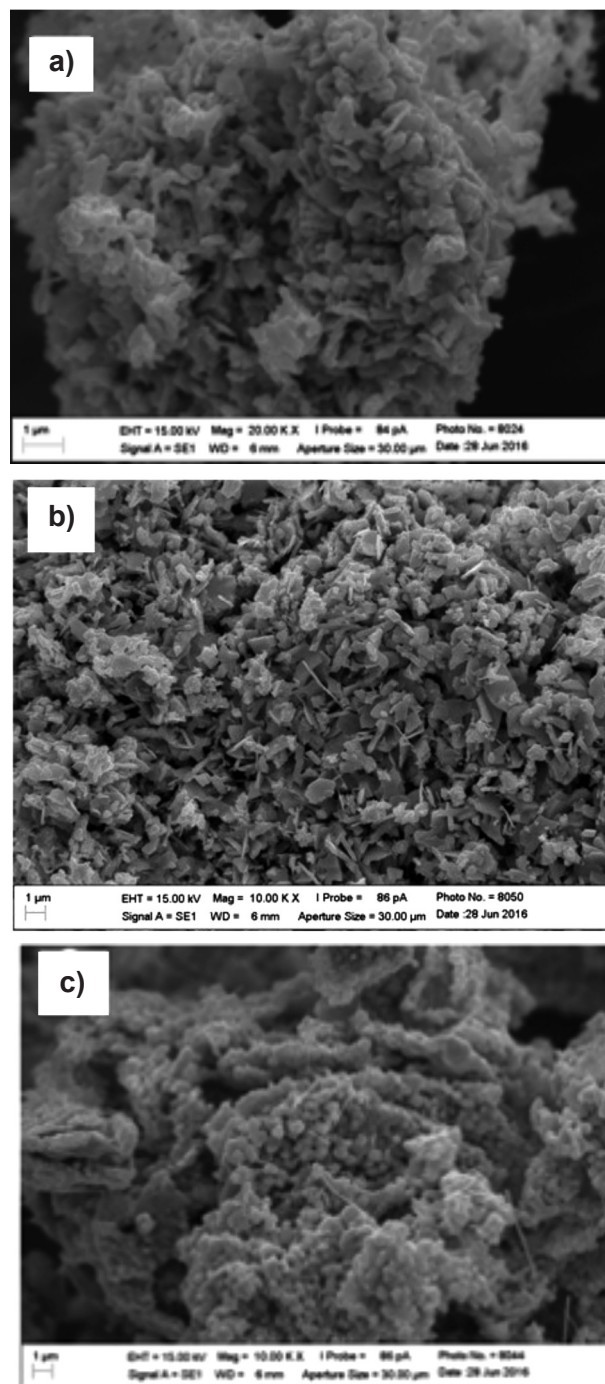


Figure 2: SEM micrographs for the samples: a) Mo_2C ; b) $\text{Mo}_2\text{C-Ni5\%}$; and c) $\text{Mo}_2\text{C-Ni10\%}$.

[Figura 2: Micrografias de MEV das amostras: a) Mo_2C ; b) $\text{Mo}_2\text{C-Ni5\%}$; e c) $\text{Mo}_2\text{C-Ni10\%}$.]

this behavior was related to a micro-deformation in the structure of this sample. Moreover, we considered that the Mo_2C and $\text{Mo}_2\text{C-Ni10\%}$ samples had similar values. At the same time, little or no micro-deformation and/or anisotropy in the samples were found in terms of the crystalline structure of the materials. This feature may be related to the efficiency of the doping method and the gas-solid reaction synthesis technique, reinforcing the concept that nickel was

incorporated in the molybdenum carbide structure.

Scanning electron microscopy: SEM micrographs for the studied samples are presented in Fig. 2. In Fig. 2a, the molybdenum carbide morphology is presented, and it is possible to observe the agglomerates of the material. The morphology of Mo₂C-Ni5% is presented in Fig. 2b; it is possible to observe the formation of particle agglomerates, mostly of platelets with heterogeneous sizes. In addition, the presence of rod-shaped particles was observed, related to the aggregation of the metallic nickel to molybdenum, causing grain growth. When comparing the micrographs of Mo₂C-Ni5% (Fig. 2b) and Mo₂C-Ni10% (Fig. 2c), it was more evident the decrease in the particle size with the addition of Ni ions into the molybdenum carbide; the sample with 10% Ni showed spheroidal shaped particles in the great majority, attributed to the molybdenum carbide, and also the presence of some rod-shaped particles. The increase in Ni content on the samples led to a decrease in the mean particle diameter, indicating that the Ni played a catalytic role in the formation of Mo₂C by gas-solid reaction [24]. The Ni-doping on Mo₂C leads to a decrease in the average grain agglomerate diameter; a similar result was shown in a recent study reported by our group [23]. Here, Ni controlled the rate of the grain formation resulting in the formation of Mo₂C cores, which coalesced at approximately the same rate. This mechanism led us to conclude that the rate of Mo₂C formation was slower than the grain coalescence in the absence of Ni. These results were in agreement with the reaction conditions utilized (700 °C and atmospheric pressure). Through energy-dispersive X-ray spectroscopy (EDX) coupled in SEM, it was possible to observe the homogeneous distribution of Ni in the entire sample and not only on its surface, indicating the absence of a biphasic composite. The results showed Ni content values close to those calculated in the experiments, reaching 5.88% for the sample with 5% Ni and 11.06% for the sample with 10% Ni, making these values acceptable according to the calculations performed.

Analysis of band gap: the calculation of optical band gap energies (E_{gap}) of molybdenum carbides with and without Ni were evaluated according to the results obtained by the analysis of diffuse reflectance spectroscopy using the equation proposed by Wood and Tauc [24]. According to this equation, the E_{gap} relates the absorbance and energy of the photons by:

$$\alpha \cdot h \cdot \nu = A(h \cdot \nu - E_{\text{gap}})^n \quad (\text{B})$$

where α is the absorbance, A is a proportionality constant, h is the Planck constant, ν is the frequency, and n is a variable that depends on the type of the electronic transition, given by $n = 1/2, 2, 3/2, \text{ or } 3$ for direct permitted, indirect permitted, direct prohibited, or indirect prohibited, respectively. Since molybdenum carbide presents an indirect electronic transition [28], we considered the value $n=2$ in Eq. B. Graphs of $(\alpha \cdot h \cdot \nu)^{1/2}$, $\alpha = F(R)$, as a function of the photon energy ($h \cdot \nu$) for the materials Mo₂C, Mo₂C-Ni5%, and Mo₂C-Ni10% are shown in Fig. 3. The results

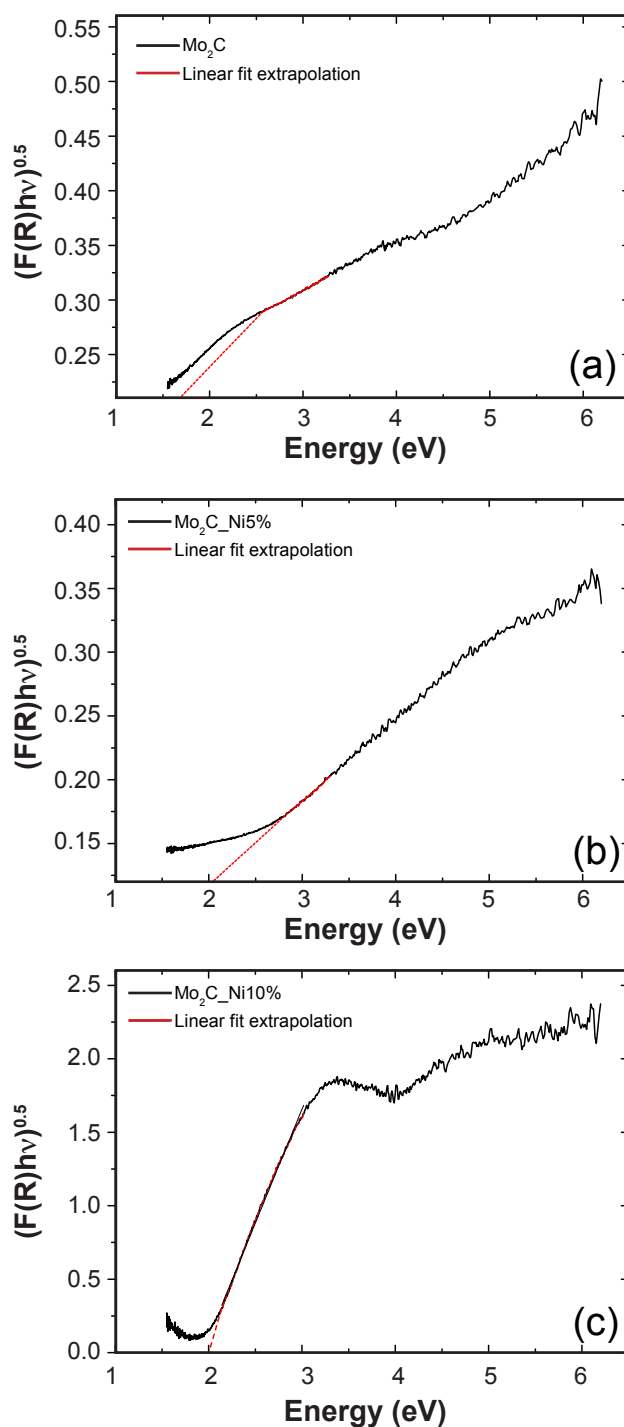


Figure 3: Tauc plots based on diffuse reflectance spectroscopy measurements for the samples: a) Mo₂C; b) Mo₂C-Ni5%; and c) Mo₂C-Ni10%.

[Figura 3: Gráficos de Tauc baseados em medidas de espectroscopia de refletância difusa para as amostras: a) Mo₂C; b) Mo₂C-Ni5%; e c) Mo₂C-Ni10%.]

of diffuse reflectance spectroscopy measurements of the samples showed the E_{gap} values of 1.71, 2.02, and 2.00 eV for Mo₂C, Mo₂C-Ni5%, and Mo₂C-Ni10%, respectively. As reported in [28], the E_{gap} of the Mo-Mo₂C sample is 1.49

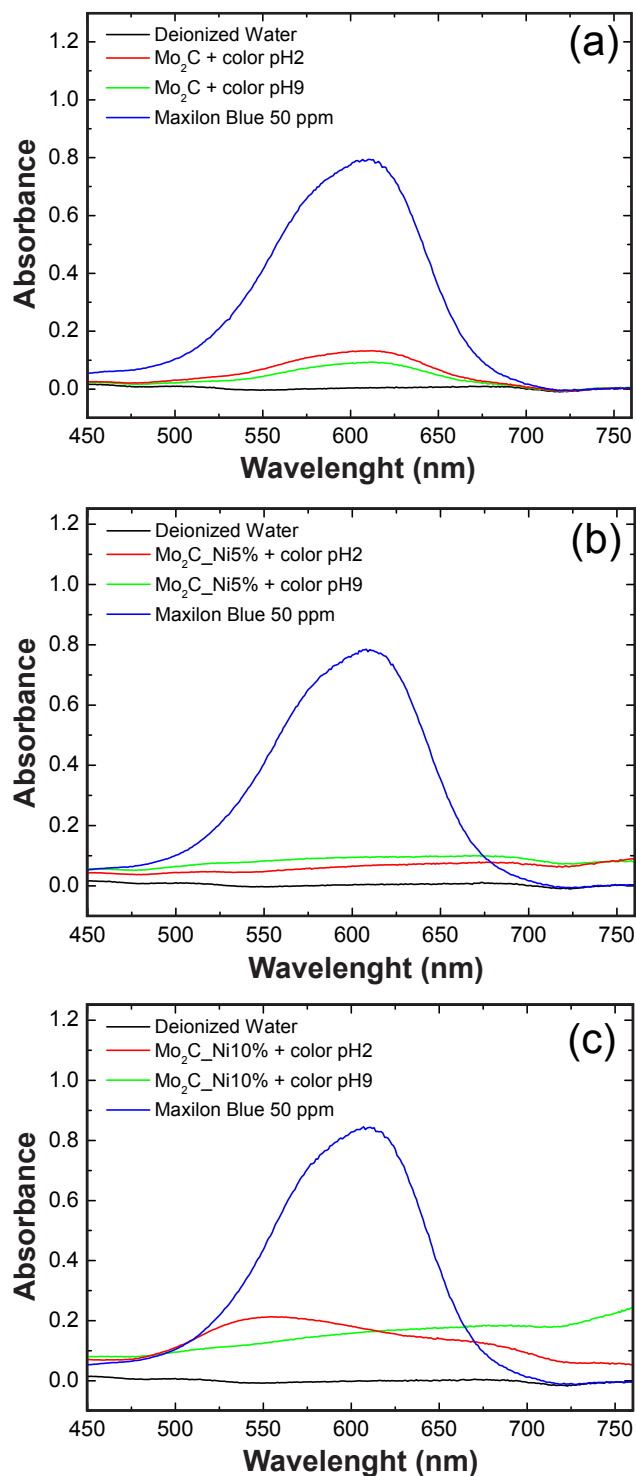


Figure 4: UV-visible spectra after photocatalytic test with Maxilon Blue GRL 300 for: a) Mo_2C ; b) $\text{Mo}_2\text{C-Ni5\%}$; and c) $\text{Mo}_2\text{C-Ni10\%}$.

[Figura 4: Espectros no UV-visível após teste fotocatalítico com Maxilon Blue GRL 300 para: a) Mo_2C ; b) $\text{Mo}_2\text{C-Ni5\%}$; e c) $\text{Mo}_2\text{C-Ni10\%}$.]

eV, stressing that the lower the intensity of this energy, the greater the charge separation efficiency of electron-hole pairs of the photocatalyst. The variation of the band gap energy is directly related to the content and type of dopant

Table II - Concentration values of dye solutions and yields in photocatalytic reaction determined from UV-vis spectra. [Tabela II - Valores de concentração das soluções do corante e rendimentos na reação fotocatalítica determinados a partir dos espectros no UV-vis.]

Sample	pH 9.0		pH 2.0	
	Conc. (ppm)	Yield (%)	Conc. (ppm)	Yield (%)
Mo_2C	6.64	86.7	4.74	90.5
$\text{Mo}_2\text{C-Ni5\%}$	4.77	90.4	3.56	92.8
$\text{Mo}_2\text{C-Ni10\%}$	8.59	82.8	8.42	83.2
Error (%)	1.10	2.19	1.46	2.89

added to the matrix. From the XRD analysis, there was a decrease in the intensity of the main peak with the addition of the dopant, resulting in changes in the E_{gap} [29, 30]. Also, the occurrence of molybdenum carbide structure change from orthorhombic to hexagonal leads to a change in the E_{gap} value. Our results were in accordance with that report in [31], in which the hexagonal structure of the molybdenum carbide shows an E_{gap} value of around 1.05 eV.

The E_{gap} is associated with the presence of intermediate energy levels within the material gap. These intermediate energy levels depend on the order-disorder degree of the crystalline structure of the material, indicating that the increase of the structural organization within the grid leads to a reduction of the intermediate energy levels and, consequently, E_{gap} increases. In this regard, we can explain the increase in the E_{gap} value of Mo_2C in relation to that found in the literature. This organization's degree is associated with the synthesis method in the production of molybdenum carbide [29, 30]. It should be noted that the presence of nickel in the molybdenum carbide structure causes changes in the band gap values and may be related to the change in the morphology of the doped materials [24]. The E_{gap} variation can be explained by distinct factors such as: particle morphology, synthesis method, shape (thin film or powder), annealing temperature, and processing time [29]. These factors led to different structural defects in the crystalline network of the sample, such as oxygen vacancies and network distortion, which are capable of promoting the formation of intermediate energy levels within the gap [30]. According to the literature, in the diffuse reflectance study, Mo_2C presented a band gap energy of 1.1 eV [29]. In another study, the Mo_2C presented a band gap energy of 3.85 eV. This difference in E_{gap} values is associated with the degree of organization of the crystal structure of molybdenum carbide material [30].

Photocatalysis characterization: to determine the reaction time (RT) and pH solution for our samples, studies varying the RT from 1 up to 5 h for distinct pH solution (2.0, 7.0, and 9.0) were performed. From these results, we established that the reaction time of 1.0 h and the pH of the solution at 2.0 and 9.0 were the optimal parameters for the study. The representative results of photocatalytic degradation

of the Maxilon Blue GRL 300 textile dye in the presence of the molybdenum carbide catalyst at different reaction times and pH of the solution showed a strong decrease in absorbance at low reaction time values, followed by near-constant value as the reaction time increased. Similar results were observed for the samples with distinct pH values. Fig. 4 shows the UV-visible spectra for the carbide tested (Mo_2C and $\text{Mo}_2\text{C-Ni}$) in 50 ppm dye solutions at pH 2.0 and 9.0 in the presence of tungsten light. The solutions were analyzed with and without carbide-type materials in order to compare the absorbance results found after the reaction time of 1.0 h, so it was verified the decrease in the concentration of the dye solution from the 50 ppm concentration matrix solution. With the results obtained in the spectroscopy analysis in the UV-visible region, it was possible to calculate the final concentrations and yields of each solution after a reaction time of 1 h for Mo_2C , $\text{Mo}_2\text{C-Ni5\%}$, and $\text{Mo}_2\text{C-Ni10\%}$; these values are listed in Table II.

The principle and mechanism of photocatalysis have been discussed [13, 15, 19]. The irradiation of Mo_2C and $\text{Mo}_2\text{C-Ni}$ by a light source caused the excitation of the electrons located in the valence band to the conduction band. As a result, holes (h^+) and electrons (e^-) were generated in the valence and conduction bands, respectively. The generated h^+ was derived from the hydroxyl radicals in reaction with water. These hydroxyl radicals caused the degradation of organic molecules, often from dyes. The photogenerated electrons were attached to oxygen in the form of superoxide radicals and subsequent conversion to hydroxyl radicals. Usually, the efficiency of photocatalysts is characterized by factors such as the adsorption capacity, the electron-hole pair production, and the recombination time of the electron-hole pair. The latter has an important factor in the overall performance of the photocatalyst material [19]. The photocatalysis process, with activation through visible light for various materials, such as Mo_2C , was studied in distinct reports [32-34], demonstrating the efficiency of this process for H_2 evolution.

The photocatalytic activity measured for all samples showed a considerable reduction in the final concentrations of the dye solutions, compared to the initial 50 ppm solution, both in acidic and basic media. For the samples of Mo_2C and $\text{Mo}_2\text{C-Ni5\%}$, we observed higher reaction yields in comparison with $\text{Mo}_2\text{C-Ni10\%}$. In particular, the Mo_2C sample showed the same level of conversion compared to $\text{Mo}_2\text{C-Ni5\%}$, considering the calculated error range. For the $\text{Mo}_2\text{C-Ni10\%}$ sample, it can be considered that there was a limit to the bulk quantity of dopant for the activation of the material and dye degradation occurring in both acidic and basic environments. The carbides studied in this paper had the characteristic of nanometric materials, proved by the analysis of crystallite size, which added ease of dispersion in the solution, causing difficulties in removing them from the suspension, which may have impaired the UV-visible analysis; even if the samples were centrifuged, there would still be particles in suspension. In general, we can say that the photocatalytic behavior of the samples is connected with

the structural characteristics of the materials. The insertion of Ni-dopant in the Mo_2C structure, identified through XRD analysis, increased the formation of particle clusters (see SEM images), causing an increase in the band gap energy of the materials. Therefore, it positively influenced the photocatalytic degradation of the dye.

CONCLUSIONS

The process of producing Ni-doped molybdenum carbides in a fixed bed reactor proved to be efficient. According to the results obtained through XRD and SEM analyses, it was possible to observe the presence of the characteristic peaks of Mo_2C in all samples and the change in the morphology of the particle agglomerates with the addition of Ni. The mean crystallite size calculations indicated that the particles had nanometer sizes ranging from 34.7 to 37.5 nm, calculated by the HWL method from the X-ray diffraction peaks. The Maxilon Blue GRL 300 was degraded upon irradiation with tungsten light in the presence of all catalysts. Tests carried out in both acidic and basic pH demonstrated the efficiency of the photocatalysis process for the synthesized carbides, achieving a removal performance of up to 92% at the final concentration of the dye. This study revealed a new system of a low-cost catalyst with potential application in the photocatalytic degradation of dyes.

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