Welding lines formation in holes obtained by low pressure injection molding of ceramic parts

(Formação de linhas de solda em furos no processo de injeção a baixa pressão de peças cerâmicas)

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Abstract

This work presents a study to evaluate the process of producing internal holes in ceramic disks produced by low pressure injection molding (LPIM) process. Two process conditions defined as pre-injection and post-injection were used to test the proposition. In the first one the pin cores that produce the holes were positioned in the cavity before the injection of the feedstock; and in the second one, the pin cores were positioned in the cavity, just after the feeding phase of the injection mold. An experimental injection mold designed and manufactured to test both processes was developed to produce ceramic disk with Ø 50 x 2 mm with four holes of Ø 5 mm, equally and radially distributed through the disk. The feedstock was composed of 86 wt% alumina (Al_2O_3) and 14 wt% organic vehicle based on paraffin wax. Heating and cooling systems controlled by a data acquisition system were included in the mold. The results showed that there were no welding lines with the post-injection process, proving to be an option for creating holes in the ceramic parts produced by LPIM. It was observed that best results were obtained at 58 °C mold temperature. The pins extraction temperature was about 45 °C, and the injection pressure was 170 kPa.
Keywords: low pressure injection molding, welding lines, holes.

Resumo

Este trabalho apresenta um estudo para avaliar o processo de produção de furos internos em discos cerâmicos produzidos pelo processo LPIM (moldagem de injeção a baixa pressão). Duas condições de processo definidas como pré-injeção e pós-injeção foram usadas para a proposição. No primeiro, os insertos dos pinos que produzem os furos foram posicionados na cavidade antes da injeção do veículo orgânico; no segundo, os insertos dos pinos foram posicionados na cavidade, logo após a fase de alimentação do molde de injeção. Um molde de injeção experimental projetado e fabricado para testar ambas condições foi desenvolvido para produzir um disco cerâmico com Ø 50 x 2 mm com quatro furos de Ø 5 mm, igualmente e radialmente distribuídos no disco. A matéria-prima foi composta por 86% em massa de alumina (Al_2O_3) e 14% em massa de veículo orgânico à base de parafina. Sistemas de aquecimento e refrigeração controlados por um sistema de aquisição de dados foram incluídos no molde. Os resultados mostraram que não houve linhas de solda com o processo de pós-injeção, provando ser uma opção para criar furos nas peças cerâmicas produzidas pelo LPIM. Observou-se que os melhores resultados foram obtidos a 58 °C como temperatura do molde na injeção. A temperatura de extração dos pinos foi de 45 °C e a pressão de injeção foi de 170 kPa.
Palavras-chave: *injeção a baixa pressão, linhas de soldagem, furos.*

INTRODUCTION

The manufacture of advanced ceramic parts has been challenging around the world, searching ways of making this kind of materials and products useful and profitable in applications that demand reliability in high temperatures [1]. There are different types of manufacturing processes for advanced ceramics, such as uniaxial-pressing, hot pressing, cold isostatic and hot isostatic pressing, slip casting, tape casting, extrusion, injection molding, gel casting or forming [2]. This research focused on the low pressure injection molding (LPIM) process. LPIM has been argued as an alternative process to produce medium and small quantity and high quality complex shapes of ceramic components in comparison to other possible processes such as high pressure injection molding. Some of the main advantages are low investment on machine, mold and low energy consumption for small and complex components production [3, 4]. In this process, the feedstock - composed of a main material, i.e. alumina, and an organic vehicle (or binder), is injected into the mold cavities by air pressure [5], producing the green parts. After that, the organic vehicle is removed by the debinding process, and finally the parts are sintered at 1600 °C. Additionally, this type of process allows the production of more complex geometric shapes, satisfactory dimensional tolerances, the use of different materials, through a traditional polymer injection molding process [4, 6, 7]. Traditionally, researches related to LPIM process have usually approached materials composition and its influence on the final properties [5]. However, due to

the brittle characteristics of the green parts and shrinkage, during and after the cooling cycle, this work has focused on an alternative approach for producing holes in flat geometry parts. In a previous work [8], a study identified the temperature range to extract the core in cup-shaped type parts, i.e. when the feedstock is almost solidified and it has not started the shrinkage process. However, that study revealed one alternative approach to punch the holes just after the injection process, avoiding welding lines. In order to perform this initial study, a disk geometry of Ø 50 mm x 2 mm was chosen to include four holes of 5 mm distributed radially. The disk geometry was chosen as previous work was developed with this geometry, being the process parameters well known. An experimental two cavity mold was designed with two extraction systems: pre and post cavity feeding phase.

The manufacturing of ceramic parts by LPIM process can be divided into four main steps: selection and preparation of feedstock (material); injection molding process; organic vehicle removal; and final sintering [5, 9]. In LPIM, the ceramic powder is mixed with other materials that form the organic vehicle, also called organic binder, which can be composed of elements such as paraffin, waxes and oils, resulting in good fluidity and low viscosity feedstock. Thus, the feedstock has primordial importance in this process [10, 11]. On one hand this provides good viscosity for the mixture; on the other hand, it may become a problem considering its further extraction, in the range of 30 to 40% in volume, usually related to shrinkage, deformations and defects formation [6]. The mixture is heated in the machine tank (between 60 to 100 °C to wax based binders) providing a low viscosity to the feedstock and allowing its transfer to the injection mold by air pressure. Injection pressures are low (between 150 kPa e 1 MPa) when compared to traditional injection molding [12]. The feedstock is cooled in the injection mold producing the solid green parts. After their extraction, these green parts go to the binder removal phase and after that to the final sintering process. The injection phase, i.e. feeding, cooling and extraction of the part, is usually fast, allowing the production of small lots of parts. Green parts are fragile and must be handled with care. If necessary, at this stage, the parts can be adjusted by additional manufacturing process [5]. The main aspects, alone or combined, which may affect the results of this process are: mold design, cooling process, material rheology, injection molding and machine parameters, such as tank temperature, injection pressure, injection feeding time, etc. [5, 13]. The proper control of such variables results in parts free of defects and free of residual stress, which can strongly affect the downstream phases of this process [13, 14]. The temperature difference between the feedstock and mold cavities is very important in this process. Therefore, temperature difference around 50 °C are recommended to produce parts with minimum voids (porosity) and low deformation. Low temperature differences may result in formation of paraffin crystals during the cooling process resulting structural differences between layers from interior

Figure 1: Welding lines in a ceramic part produced by LPIM process.
[Figura 1: Linhas de solda em uma peça cerâmica produzida pelo processo LPIM.]

to the exterior of the part [5, 7].

One of the disadvantages of LPIM is the time required for the binder removal, which is usually long. Depending on the chosen removal technique it might take from 6 to 40 h [15]. The final phase is the sintering process based on a thermal consolidation so that the voids are eliminated with the growing of the grain size [12]. The temperatures at this phase are usually around 1600 °C with heating rate of 1 °C/min and stabilization of 2 h [15]. Another aspect that is critical in LPIM process is the shrinkage associated with the brittle characteristics of the green parts. This is usually a result of waxes addition to composing the feedstock, as it was mentioned before [16]. These waxes and paraffin are melted during the feedstock preparation and the injection phase increasing the volume of the feedstock and resulting in a shrinkage during the cooling of the injection mold and the solidification of the part. Approaches such as the overshot are not possible because of the range of applied pressure (usually under 0.7 MPa). Also, the feeding gate usually solidifies quickly making it difficult to control the pressure [17]. Shrinkage is not a great issue to deal with in most components geometry. However, when dealing with holes produced by inserts (core), this becomes a main issue [8], mainly because of the cracks, voids and welding lines. Welding lines in injection molding usually occur because of the separation of the material flow, enforced by some obstruction and its posterior junction [18-21]. These welding lines are potentialized by the low part of organic vehicle in the feedstock. Welding lines can result in stress concentration and cracks formation, mainly in LPIM that requires posterior debinding and sintering (Fig. 1). This paper presents an alternative approach to mold holes in ceramics parts created by LPIM. The study compares the process of creating holes by some inserts put into the impression cavity before (called here pre-injection) and after (called here post-injection) feeding the cavity.

MATERIALS AND METHODS

Feedstock: the feedstock used in this research was composed of: 86 wt% Almatis Alumina - Al_2O_3, d_{50}=0.4 μm; and 14 wt% of organic vehicle based on paraffin wax (Table I). In order to produce the final feedstock, organic vehicle was mixed and melted at 90 °C, then the alumina was added in small portions. After that, the feedstock was processed during 30 h directly into the injection molding machine (Peltsman MIGL-33), with planetary double blade mixer, applying heating and vacuum. This method was used in previous works with good results [8, 14]. The composition presented in Table I was defined in [22] as a result of experimental work and was defined to provide best rheology properties to the LPIM along with a homogenous mass loss distribution during the debinding step within a wide time range. Specifically, to the organic vehicle, components with low molecular weight were used, avoiding the use of polymers.

Table I - Feedstock composition.
[Tabela I - Composição do veículo orgânico.]

Material	Fraction (wt%)	Fraction (vol%)	Melting temp. (°C)
Alumina (Al_2O_3)	86	55	-
Paraffin	10.5	33.75	49-52
Paraffin microcrystalline	1.4	4.50	82.4
Carnauba wax	1.4	4.50	80-87
Oleic acid	0.42	1.35	16
Stearic acid	0.28	0.90	73

Experimental geometry and injection mold: in order to facilitate the feeding of the cavity, the extraction of the final parts and the result comparison, a disk shape part was used. This geometry was used in [23] with good results. The dimensions were defined in 50 mm diameter and 2 mm thickness. Additionally, to identify and analyze the welding lines formation, four holes of 5 mm were placed radially and uniformly, 15,5 mm from the disk centre (left side of Fig. 2). Fig. 2 also depicts on the right side the pair core/cavity for the geometry which was defined. The core plate had four holes to guide the core pins. The cavity plate had four indentations of 1,5 mm depth to fit the core pins. Venting channels were manufactured with 0.02 mm thickness in order to allow a smoother air escape. Based on this geometry, an experimental injection mold, previously used in [15], was used with some adjustments. Fig. 3 shows a representation of the injection mold. The mold plates were made of aluminum (7075-T6) and had as main function providing structural support and alignment and holding heating, cooling and feeding systems. The heating system was designed based on four Ø 6 x 130 mm cartridge heaters (150 W each) and controlled by a DAQ (Measurement Computing, USB-TC) and thermocouples (K-type), placed at 1 mm of the cavity surface. Temperature

variations were captured every 0.5 s. The cooling system was based on a spiral semi-circular cooling circuit channels with Ø 5 mm machined section to provide uniform cooling of injected parts. Sealing plates with electrolytic cooper were used between the cooling circuit and cavity inserts. A chiller was used on the cooling system with a capacity of 3.5 kW and 8 m^3/min flow rate. Finally, the feeding system was based on a central sprue connecting the injection machine feeding channel to the insert cavity. The mold main cavities were covered with polytetrafluoroethylene (PTFE, Teflon®) to minimize adhesion between part and cavity surface.
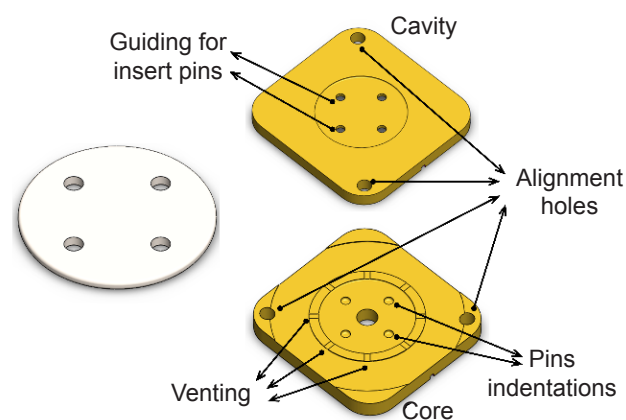

Figure 2: Disk geometry (left) and mold inserts (right) representation.
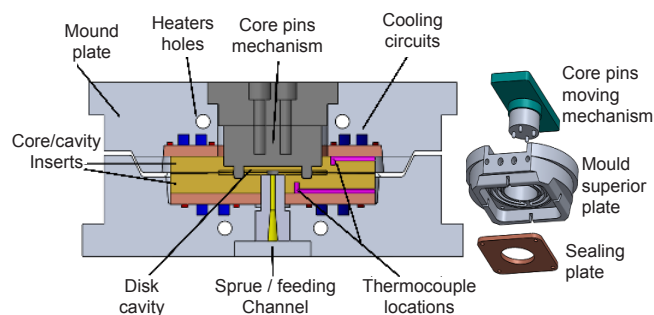*[Figura 2: Representação da geometria do disco (esquerda) e insertos do molde (direita).]*

Figure 3: Experimental injection mold designed and used.
[Figura 3: Molde de injeção experimental projetado e utilizado.]

To test the two approaches proposed in this work, i.e. core pins placed pre and post-injection feeding phase, a moving structure was developed to allow the positioning of the core pins in the cavity (Fig. 3, right). Thus, for the post-injection approach, as shown in the left side of Fig. 4, the core pins were not placed inside the cavity impression during the feeding phase. Just after this phase, the structure was pressed down (Fig. 4, right side) placing the core pins inside the cavity and creating the holes. In this approach, when the temperature of the mold reached 45 °C, the structure was removed again, leaving the shape to the final cooling phase, i.e. until 35 °C. For the pre-injection approach, during the feeding phase, the structure with the core pins stayed

pressed down, providing the obstructions and the cores to shape the holes in the part. In the same manner, the structure was removed when the mold temperature reached 45 °C.
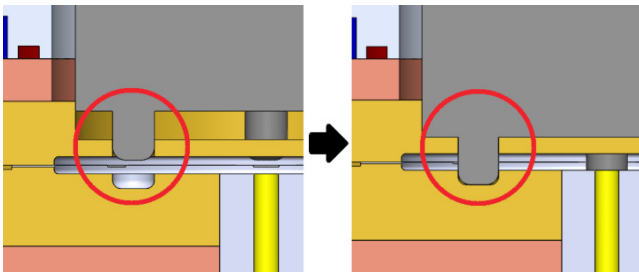

Figure 4: Mechanism for creating pre and post-injection feeding approaches.
[Figura 4: Mecanismo para criar as condições pré e pós-injeção.]

Experiment conditions: the experiments were carried out with the parameters presented in Table II. These parameters were used in previous works [7, 23, 24], proving to be the most recommended parameters for these applications. Thus, the mold temperature at the injection moment was defined at 58 °C, injection pressure around 170 to 200 kPa, injection time (injection and overshot) of 12 s, and final extraction temperature for the part at 35 °C. The temperature of 58 °C as initial injection mold temperature was defined based on initial tests in order to provide a smoother condition of pressing the hole core mechanism inside the cavity just after the injection. The mold temperature usually rose about 2 °C just after the injection while the feedstock temperature was 90 °C. In a similar way, the temperature of 45 °C was defined to extract the hole cores (pins). Lower temperature range increased the force for the extraction of the pins and the formation of cracks, while the higher temperature range resulted in the deformation of the hole. At 45 °C, the part was practically solidified and the force to extract the pins were weaker than the force at 35 °C, when the final extraction of the part was made.

In relation to organic vehicle removal, the heating ramps were the same used in [23] (Fig. 5). It was carried out a process of debinding by wicking at 250 °C during 36 h, followed by a debinding process at 1000 °C for 1 h, using a Sanchis B CER muffle furnace. Final sintering was

Table II - Injection molding process parameters.
[Tabela II - Parâmetros definidos para o processo de injeção.]

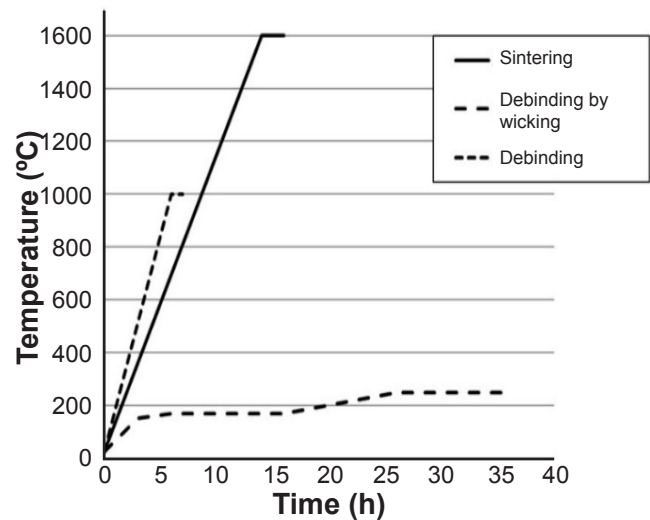| Parameter	Value
Feedstock temperature	90 °C
Injection mold temperature	58 °C
Pin core insertion temperature	60 °C
Pin core extraction temperature	45 °C
Injection mold temperature on extraction	35 °C
Injection pressure	0.17-0.20 MPa
Injection time	12 s

Figure 5: Heating curves for debinding and sintering.
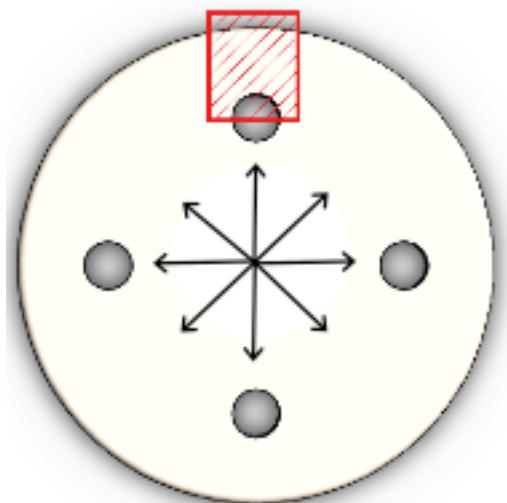*[Figura 5: Ciclos térmicos para extração de ligante e sinterização final.]*

Figure 6: Defined region for weld line and crack formations.
[Figura 6: Região definida para análise da formação das linhas de solda e trincas.]

performed at 1600 °C during 2 h in a Termolab MLR17 furnace, with Kanthal Super heating elements (molybdenum disilicide) and oxidizing atmosphere. At the end of this process the samples were left inside the furnace switched off for cooling until the temperature of 250 °C, when the furnace door was opened. This final cooling process took around 24 h and was important to avoid residual stress in the final parts.

Final holes analysis: the final analysis of the cracks or welding lines formation was based on visual observations with an optical microscope, Zeiss Axio Binocular. For this phase, it was defined the area depicted in Fig. 6. This region was chosen because the feeding point is placed at the centre of the part providing a uniform flow of material to the cores and forming the welding lines. Both final diameter and circularity of the hole were measured using an optical

measurement equipment, Hexagon Optiv Lite OLM 3020. Two strategies of measuring were used: one by windows (based on the hole geometry from which the circle is defined) and one by points (3 points must be selected and a circle is defined based on them). Twenty-five samples were injected.

RESULTS AND DISCUSSION

Welding lines analysis

Pre-injection samples: for the optical analysis of the pre-injection condition, seven samples were selected (A1, A2, A12, A18, A19, A23 and A24). These samples presented the best superficial conditions in the holes areas. All samples, in this case, presented welding lines in the four holes, due to the material flow obstruction. The welding lines appeared after

the obstruction following the contour of the hole core (Fig. 1). Fig. 7 depicts the welding line formation to the 4 stages of ceramic processing, i.e. green part (Fig. 7a), debinding by wicking (Fig. 7b), debinding at 1000 ºC (Fig. 7c) and final sintering at 1600 ºC (Fig. 7d). Under this condition, i.e. pre-injection, there was also the presence of cracks in some holes. In this case, the cracks appeared during the first stage, i.e. green parts (Fig. 8). On the other hand, the superficial finishing for this condition, i.e. walls and contour of the holes, presented good results.

Post-injection samples: using the same criteria of the previous condition also seven samples were selected for the optical analysis: A7, A8, A10, A15, A21, A22 and A25. In this case, no welding lines or cracks were identified on the hole perimeters. However, the surface fishing of the holes presented some burrs. These burrs were result of the

Figure 7: Pre-injection condition welding lines: (a) green part; (b) debinding by wicking; (c) debinding at 1000 ºC; (d) sintering at 1600 ºC.
[Figura 7: Linhas de solda na condição pré-injeção: (a) peça verde; (b) extração de ligante por leito de pó; (c) extração de ligante a 1000 ºC; (d) sinterização a 1600 ºC.]
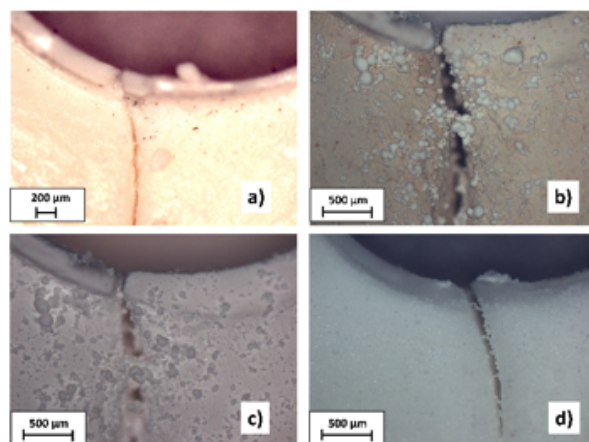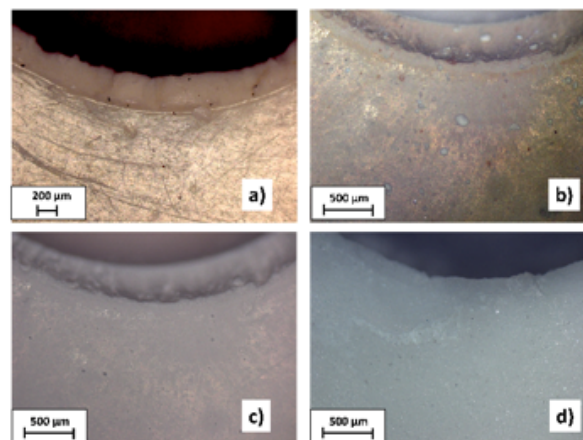

Figure 9: Post-injection condition with no welding lines: (a) green part; (b) debinding by wicking; (c) debinding at 1000 ºC; (d) sintering at 1600 ºC.
[Figura 9: Condição pós-injeção sem linhas de solda: (a) peça verde; (b) extração de ligante por leito de pó; (c) extração de ligante a 1000 ºC; (d) sinterização a 1600 ºC.]

Figure 8: Pre-injection condition cracks formation: (a) green part; (b) debinding by wicking; (c) debinding at 1000 ºC; (d) sintering at 1600 ºC.
[Figura 8: Formação de trincas na condição pré-injeção: (a) peça verde; (b) extração de ligante por leito de pó; (c) extração de ligante a 1000 ºC; (d) sinterização a 1600 ºC.]
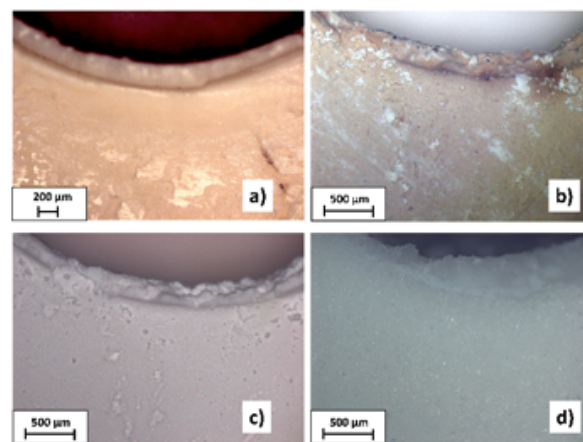

Figure 10: Post-injection condition with burrs formation: (a) green part; (b) debinding by wicking; (c) debinding at 1000 ºC; (d) sintering at 1600 ºC.
[Figura 10: Formação de rebarbas na condição pós-injeção: (a) peça verde; (b) extração de ligante por leito de pó; (c) extração de ligante a 1000 ºC; (d) sinterização a 1600 ºC.]

adhesion between the hole core (insert) surface and the ceramic material, created by the insertion and/or extraction movements. In [15], the core pins were coated with TPBU (Teflon®) to avoid the adhesion problems with good results. This was not tested in this work. Figs. 9 and 10 show the results for this condition. In Fig. 10 the presence of burrs can be observed.

Dimensional analysis

Measurements of the hole diameters and circularities was performed in a profile projector by two different strategies as mentioned above: by points and by window. Measurements were taken in the four stages: green parts, debinding by wicking, debinding at 1000 ºC and final sintering at 1600 ºC. Reference diameters of the 4 pins in the mold insert were also measured with a calliper Mitutoyo (resolution 0.05 mm). A value of diameter of 5.15±0.05 mm was found and used as reference (dm) to compare with the values of the holes obtained in the parts (df). Based on these values the shrinkage of the holes was calculated (Eq. A) and is presented in Table III.

$$\%Re = \frac{dm - df}{dm} \cdot 100\% \qquad (A)$$

For the pre-injection condition, the average value of hole diameter was 5.03 mm, with a standard deviation of 0.03 mm and the shrinkage of the green parts was around 2.2%. This value was the expected considering the feedstock used in [23]. For the post-injection condition the average value of diameter was 5.05 mm, with a standard deviation of 0.05 mm and a shrinkage value of 1.93%. In this case, although the difference was not significant, the smaller shrinkage can be due to the material compaction during the penetration of the core pins (Table III). The average value of the hole diameter was a composition of the diameters obtained by the two strategies of measurement, i.e. points and window. The diameter and shrinkage averages were calculated during the four phases of the ceramic material post-processing, i.e. green parts, debinding by wicking, debinding at 1000 ºC and final sintering at 1600 ºC (Table III). The shrinkage was calculated considering the diameter of the core pins, i.e. 5.15 mm. The diameter averages to the post-injection condition were slightly larger than the pre-injection condition. However, considering the final

Table IV - t-test's statistic for diameters.
[Tabela IV - Análise estatística t-test para os diâmetros dos furos.]

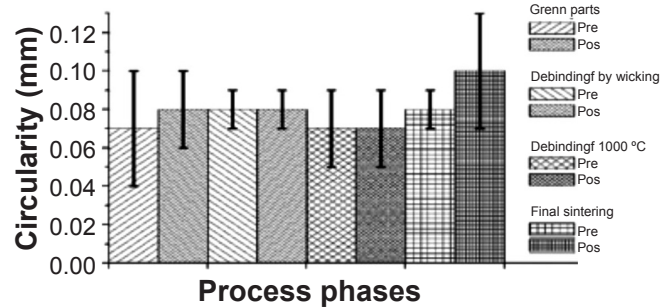| Parameter	Pre-injection holes mean diameter	Post-injection holes mean diameter
Mean (mm)	5.031	5.055
Variance (mm^2)	0.0014	0.0008
Sample size	10	9
Freedom degree	17	
Stat t	-1.58	
P(T≤t)	0.07	
t-value	1.74	

Figure 11: Circularity of the holes for pre and post-injection conditions during the four processing phases.
[Figura 11: Circularidade dos furos nas condições pré e pós-injeção durante as 4 fases de processamento.]

sintering, there were no significant differences, due to the extraction of the organic material (16% in volume of the feedstock). As mentioned before, the diameter for the post condition suffers from the insertion and extraction of the core pins, where some material was removed by the adhesion. Pre-injection holes mean diameters were compared with post-injection holes by t-test statistic with a significance level of 0.05 as shown in Table IV. Within the uncertainties of the measurements, acceptance of the hypothesis H$_o$ was attained, indicating no difference between the averages of the holes in the two experimental conditions, i.e. the post-injection drilling process does not affect the diameter and contraction results.

Fig. 11 depicts the average values of circularity for

Table III - Average of hole diameter and shrinkage after the four processing phases.
[Tabela III - Média do diâmetro e da retração dos furos nas 4 fases de processamento.]

Processing phase	No. of samples	Pre-injection		Post-injection	
		Diameter (mm)	Shrinkage	Diameter (mm)	Shrinkage
Green part	19	5.034	2.2%	5.030	1.9%
Debinding by wicking	13	4.939	4.1%	5.003	2.9%
Debinding at 1000 ºC	13	4.908	4.7%	4.952	3.8%
Sintering 1600 ºC	7	4.293	16.6%	4.306	16.4%

each phase of the ceramic part processing, considering the two injection conditions, i.e. pre and post-injection. Post-injection condition presented higher values of circularity for green parts and final sintering phase, with also higher dispersion. As mentioned before, the finishing surface of the pre-injection condition was better than the post-injection condition, therefore this is also perceived in the circularity of the holes.

CONCLUSIONS

This work presented an alternative approach to producing holes in LPIM ceramic parts, avoiding the formation of welding lines and cracks. The suggested alternative is based on the positioning of core pins just after the feeding phase of the injection process. The injection mold with an experimental extraction system was designed and manufactured providing reliable results. In relation to the injection molding process parameters, best results were found for injection mold temperature at the feeding phase of 55 ºC for pre-injection condition and 58 ºC for post-injection condition. Moreover, the pins extraction temperature of 45 ºC was considered the ideal for both conditions. For the pre-injection conditions, 70% of the holes presented welding lines and cracks while none of the samples obtained with the post-injection condition presented welding lines. However, they showed lower quality on the surface finishing of the holes than the pre-injection condition. In order to minimize this, further study is set to repeat the experiment coating the core pins with Teflon®. Another suggestion for future work is manufacturing parts with different hole geometries and dimensions, using the post-injection drilling system, in order to test different joint of flow material formation. Through the statistical analysis of the hole diameters obtained in each experimental conditions, it was possible to verify that the post-injection drilling process does not affect the diameter and contraction results.

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REFERENCES

[1] J. Rödel, A.B.N. Kounga, M. Weissenberg-Eibl, D. Koch, A. Bierwisch, W. Rossner, M.J. Hoffmann, J. Eur. Ceram. Soc. **29**, 9 (2009) 1549.
[2] R.A. Barbieri, C.A. Perottoni, J.R. Zorzi, Int. J. Appl. Ceram. Technol. **9**, 3 (2012) 599.
[3] W. Bauer, M. Gomez, V. Valcárcel, C. Cerecedo, F. Guitián, M. Peltsman, J.E. Zorzi, Ceram. Ind. **156**, 5 (2006) 22.
[4] T. Moritz, R. Lenk, Powder Injection Molding Int. **3**, 3 (2009) 23.
[5] E. Medvedovski, M. Peltsman, Adv. Appl. Ceram. **111**, 5-6 (2012) 333.
[6] R.M. German, A. Bose, *Injection molding of metals and ceramics*, Metal Powder Ind. Fed., New Jersey (1997).
[7] J.E. Zorzi, C.A. Perottoni, J.A.H. Jornada, Ind. Ceram. **23**, 1 (2003) 47.
[8] W.B. Souza, C.A. Costa, C.R. Altafini, M.A. Luciano, *in* 7th Int. Conf. Polym. Moulds Innovations, K. Ragaert, L. Delva, L. Cardon, Ed., Ghent (2016) 126.
[9] H. Zhang, R.M. German, A. Bose, Int. J. Powder Metall. **26**, 3 (1990) 217.
[10] M. Müller, W. Bauer, H.J. Ritzhaupt-Kleissl, *in* 4M Conf., S. Dimov, W. Menz, Ed., Elsevier, Amsterdan (2005) 203.
[11] R. Knitter, J. Haußelt, B. Loebbecke, J. Eur. Ceram. Soc. **29** (2009) 1595.
[12] D.W. Richerson, *Modern ceramic engineering: properties, processing, and use in design*, 3rd Ed., Taylor & Francis, Boca Raton (2006).
[13] J.V. Zampieron, J.P. Soares, F. Mathias, J.L. Rossi, F. Ambrozio, Key Eng. Mater. **189-191** (2001) 610.
[14] A.R. Erickson, H.E. Amaya, Int. Conf. Mater. Powder Techn., DGM, New York (1993) 145.
[15] C.A. Costa, C.R. Altafini, F.R. Visioli, A.P. Baccin, J. Mater. Design App. **231**, 1-2 (2017) 187.
[16] E. Harima, "Estudo da moldagem de pós de alumina por injeção com ênfase na formulação e remoção de ligante", Tese Dr., Un. Fed. Santa Catarina, Florianópolis (2003).
[17] M.J. Edirisinghe, J.R.G. Evans, Int. J. High Technol. Ceram. **2**, 4 (1986) 249.
[18] H. Rees, *Mold engineering*, Hanser/Gardner Publications, New York, U.S. (1995).
[19] S. Krug, J.R.G. Evans, J.H.H. Ter Maat, Br. Ceram. Trans. **98**, 4 (1999) 178.
[20] C.H. Wu, W.J. Liang, Polym. Eng. Sci. **45**, 7 (2005) 1021.
[21] V.A. Krauss, E.N. Pires, A.N. Klein, M.C. Fredel, Mater. Res. **8**, 2 (2005) 187.
[22] J.E. Zorzi, C.A. Perottoni, J.A.H. Jornada, Cerâmica **50**, 315 (2004) 202.
[23] E. Thomazi, C.A. Costa, *in* 6th Int. Conf. Polym. Moulds Innovations, A.J. Pontes, A.S. Pouzada, Ed., Guimarães (2014) 77.
[24] F.M. Pasquali, C.A. Costa, *in* 6th Int. Conf. Polym. Moulds Innovations, A.J. Pontes, A.S. Pouzada, Ed., Guimarães (2014) 303.