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Received: April 8, 2014 - Revised: November 30, 2014

A study is reported on Langmuir and Langmuir-Blodgett (LB) films obtained by adding different percentages of stearic acid (SA) to poly(3-octylhiophene) (P3OT). Surface pressure–mean molecular area (π –A) isotherms, ultraviolet-visible (UV-Vis) absorption spectrum, atomic force microscope (AFM) and dc electrical measurements have been used for characterization. From π –A isotherms, the area per monomer for condensed films (of SA) increased according to P3OT amount. Good transfer ratios were obtained for LB deposition from P3OT/SA films with higher percentage of SA. LB films revealed linear growth as monitored by UV-Vis absorption spectra. Electrical characterization showed higher conductivity for LB film with 54 wt.% of SA. AFM images of the same film revealed high roughness and presence of large size agglomerates. Finally, interdigitated electrodes (IDE) coated with P3OT/SA LB films showed to be sensitive to the presence of volatile organic compounds.

Keywords: Langmuir, Langmuir-Blodgett, polythiophenes, electrical characterization, sensors

1. Introduction

Semiconducting conjugated polymers have drawn much attention due to their unique electrical and optical properties, reasonable chemical stability, and easy processability¹. These unique properties of conjugated polymers enable the development of several electronic devices, such as Organic Light Emitting Diodes (OLEDs), photovoltaic cells, transistors, sensors and biosensors²⁻⁵. Among semiconducting polymers, polythiophene and its derivatives display thermal and environmental stability⁶. Since regioregular poly(3hexylthiophene-2,5-diyl) was introduced7, regioregular polythiophenes have been investigated, concerning their role as an effective hole transferring semiconducting polymers. In the late 90's, advances in the synthesis and assembly of other poly(3-alkyllthiophene-2,5-diyl) (P3AT) were reported⁸.

Polythiophene films can be fabricated by different methods⁹ for further optoelectronic applications. In this context, Langmuir-Blodgett (LB) is a technique that allows the fabrication of highly ordered thin films, which properties can be monitored at molecular level^{10,11}. Two critical parameters that determine the electrical characteristics of polymer devices are their chemical composition and solid nanostructure¹². The electrical properties of these materials are drastically altered due to electronic coupling between polymer chains, which is determined by the type of packing and conformation in the solid state^{12,13}. Thus, the development of effective deposition methods that control material organization at nanoscale, e.g., the LB and the layer-

by-layer (LbL) techniques, is extremely important. Since polythiophenes exhibit low amphiphilicity, an amphiphilic long chain fatty acid (viz. Stearic Acid, SA) has been used to assist the formation of high quality polythiophene monolayer and multilayer LB films¹⁴⁻¹⁶.

The awareness of effects that contaminants indoor atmospheres can have in health has been increasing over the last 25 years. This has caused a development of international standards, which are concerned with indoor pollutants¹⁷. Thus, there is a need to develop electronic noses as devices for detection of gas and volatile organic compounds (VOCs), with low cost, portability, and operating at low power¹⁸. The regioregular polythiophenes are good alternatives for sensors fabrication in relation to other semiconducting polymers. This can be attributed to the way they organize, since small changes in conformational order and solidstate organization should cause significant changes in conductivity, promoting excellent sensitivity¹⁹.

In this paper, we report the fabrication of P3OT LB films by adding SA. Moreover, through optical and electrical characterization we analyzed the multilayer films. Finally, P3OT/SA LB films were tested toward the detection of VOCs showing good technological potential for the development of gas sensors.

2. Experimental

2.1. Materials

Regioregular poly(3-octylthiophene) (P3OT), with molecular weight of 25,000 g.mol⁻¹ and regioregularity over 90% was used in this report. The repeat unit of P3OT

molecule structure is shown in Figure 1a. Both, P3OT and SA (Figure 1b) were purchased from Sigma-Aldrich and used as received. Chloroform was purchased from Synth, and dichloromethane, tetrahydrofurane (THF), toluene were purchased from Vetec. Brazilian Synchrotron Light Laboratory provided golden interdigitated electrode (IDE) in glass substrate fabricated by photolithography, with 25 pairs of digits, 100 nm height, 8 mm length, 100 μ m width and 100 μ m separation. BK7 glass slides were left over HMDS (hexamethyldisilazane) vapor in order to create a hydrophobic surface.

2.2. Langmuir and LB films

Surface pressure vs. mean molecular area (π -A) isotherms of neat P3OT and P3OT/SA mixtures were obtained using a KSV 5000 Langmuir trough. Different weight percentages of SA were used to prepare mixed LB films. The percentage of the first fabricated film was 74 wt.%., and then decreased this percentage (54, 34 and 14 wt.%), keeping the solution concentration at 0.38 mg/mL in chloroform, in order to analyze the influence of SA addition to obtain high quality LB films²⁰. The Langmuir films were obtained by spreading a 250 µl solution of P3OT and P3OT/SA dissolved in chloroform onto ultrapure water (supplied by a Millipore system, resistivity 18.5 M Ω ·cm). A period of 15 min was allowed for the evaporation of solvent before starting compression. The π -A isotherms of P3OT/ SA were recorded at a barrier compression of 15 cm²/min and at 22 °C. For electrical characterization, 25 monolavers were transferred onto IDE coated glass substrates, by vertical dipping and for comparison by horizontal lifting (Langmuir-Schaefer - LS) methods. For optical characterization, Langmuir films were deposited onto hydrophobic glass substrates. LB films were transferred with surface pressure (corresponding to the condensed phase of Langmuir films), dipping speed and transfer ratio (TR) for downstrokes and upstrokes presented in Table 1. Good transfer ratios were obtained for LB deposition from P3OT/SA films with 74, 54 and 34 wt.% SA, resulting in high quality and uniform films. While for the other films (14 and 0 wt.%) was not possible to grow LB films.

2.3. Instrumentation and characterization

UV-Vis absorption spectra were obtained by Varian Cary 50 spectrophotometer. Concerning morphology, atomic force microscopy (AFM) studies of LB and LS P3OT/ SA films were carried out in an area of 30x30 µm² using contacting mode in Nanosurf microscope, model EasyScan



Figure 1. Structure of a) P3OT and b) SA molecule repeat unit.

2. Direct current (dc) electrical characterization of films was performed using Keithley 238 source measure unit. Current *vs.* voltage (I *vs.* V) curves were obtained to analyze samples reproducibility, to evaluate films conductivities, and to make a comparison among the fabricated films.

2.4. VOCs sensors

The electrical responses of P3OT/SA LB films upon VOCs exposure (dichloromethane, THF and toluene) were carried out using dc measurements (I vs. t), with fixed applied voltage (5V). The sensors were assayed in a homemade test chamber in the presence of VOCs in dynamic flow, dragged by nitrogen. The flow was kept constant at 60NL/h (NL is a unit of mass for gases equal to the mass of 1L at a pressure of 1atm and at a standard temperature) during all the experiments, as verified by the flow meter. For the baseline, devices were left in the presence of nitrogen flux by 15 minutes, after that the VOC is carried by the nitrogen for another 15 minutes, then returning to baseline. Therefore, the response can be seen as the current changes according to the VOCs presence. Prior to data acquisition, the sensing units were left under pure nitrogen flow for 30 min in the chamber to reach a stable reading.

3. Results and Discussion

3.1. π -A isotherms

The π -A isotherms of P3OT (neat and mixed with different SA wt%) are shown in Figure 2. It is possible to observe a transition in the liquid phase, indicating a molecular reorganization of the monolayer. Regarding polythiophenes, if thiophene rings are oriented in parallel to water surface, the area per repeat unit would be around 28 Å^{2,21}. While their expected area, according to Corey–Pauling–Koltum (CPK) model, is 14.7 Å², if the thiophene rings are oriented perpendicular to the water surface, with the backbones along the surface^{21,22}. However, the area

Table 1. P3OT LB films deposition parameters.

Wt.%ofSA	74	54	34
Deposition surface pressure (mN/m)	30	30	35
Speedupstroke (mm/min)	10	9	0.7
SpeedDownstroke (mm/min)	10	7	0.5
Averagetransferratio	0.9	0.9	0.7



Figure 2. Surface pressure vs. area of P3OT/SA LB films.

per repeat unit of neat P3OT found is about 5.4 Å². This is an indication that P3OT did not form a true monolayer, with the polymer forming aggregated and/or being coiled during the compression, in a way that only part of repeat units of main chain is in contact with water surface, due to its rigidity. For instance, the stacking forming bilayers for Langmuir monolayers of polymers has been reported^{16,23}. Therefore, it was not possible to fabricate good quality LB films of neat P3OT. From the displayed π -A isotherms of the P3OT/SA mixtures, as has already been observed for regioregular P3ATs^{14,24}, the mean molecular area increases with the increase of P3OT content, and so does the surface pressure. According to previous reports, SA molecules should have mean molecular area of 22 Å^{2,25} at a surface pressure of 50 mN/m. The P3OT/SA films with higher SA percentages (74 and 54 wt.%) revealed an area per repeat unit close to SA area. Thus, it seems that SA molecules are oriented perpendicular to water surface and they work as a matrix in which P3OT is lying down, forming aggregated islands on the SA film. Langmuir film with 34 wt.% SA showed an area about 34 Å², leaving 12 Å² for P3OT. This indicates that thiophene rings are perpendicular to the water surface and located on the side of SA molecules. The lowest SA percentage (14 wt.%) presents area of 49 Å², remaining 27 $Å^2$ to the polymer. In this case, it seems that P3OT probably is oriented parallel to water surface.

3.2. UV-Vis absorption

The degree of intra and interchain ordering in solutions, in LB and LS films of neat and mixed P3OT, can be examined from their optical absorption spectra. UV–Vis spectra of P3OT/54SA in chloroform and LB film are shown in Figure 3. For the solution spectrum, it is possible to observe an absorption band centered at about 450 nm, which is attributed to π - π * transition²⁶. LB film presents an organized structure, since films of high ordering display a red-shifted π - π * absorption band with more pronounced vibrational structure. This structure also shows the presence of longer conjugation lengths, which are supported by the literature^{27,28}. The λ_{max} for the main peaks of mixed LB films (74, 54 and 34 wt.% SA) are shown in Table 2. The UV-Vis spectrum of P3OT/74SA LB film presents the lowest redshift when compared to the other LB film spectra, consistent with the shorter conjugation due to the higher amount of SA.

The UV–Vis absorption measurement is usually used to verify the reproducibility of the transfer process²⁹. In Figure 4, it is possible to observe the absorbance spectra of mixed LB films according to the number of layers deposited onto BK7 glasses. This growth was found to be practically linear, corroborating with the uniformity of the LB films deposited onto the substrate.

3.3. Dc electrical characterization

Figure 5 shows electrical characteristics (I *vs.* V) of mixed P3OT/SA LB and LS films onto IDE substrates. It is possible to observe that the curves present an ohmic



Figure 3. UV–vis absorption spectra of P3OT/54SA solution and LB film (11 layers).



Figure 4. Absorbance spectra of P3OT/54SA LB films. Inset: the relationship between the absorbance (556 nm) and the number of transferred layers.



Figure 5. I vs. t curves for P3OT/SA LB and LS films.

Table 2. Wavelength for bands and shoulder for P3OT/SA Lb films with 11 layers.

P3OT/SA	Band 1 λ(nm)	Band 2 $\lambda_{máx}(nm)$	shoulder λ(nm)
P3OT/74SA	523	554	593
P3OT/54SA	526	556	595
P3OT/34SA	526	556	596

behavior, which was expected due to the parallel contact Au/P3OTfilm/Au, causing the absence of a potential barrier. Thereby, the resistance, R, can be estimated from the I vs. V measurements carried out on IDE. This quantity is calculated by the data linear fit (, once the slope is the inverse of R. The conductivity, σ , is the inverse of resistivity, ρ , which can be found through the relation, where L is the conductor length and A the cross section area. Thus, Olthuis et al.³⁰ developed a method of obtaining the conductivity from IDE devices. It was introduced the cell constant, which is equivalent to L/A ratio. The cell constant, κ , is determined by the sensor geometry, and takes into account number, length, spacing and height of digits, considering the dimension of IDE. The cell constant for this study was established to be 5.1 m⁻¹. Table 3 presents dc electrical conductivity, σ_{dc} , values determined for the P3OT films. It is possible to observe that σ_{dc} presents maximum value for LB film with 54 wt.% SA, showing how the amount of SA influences on the conductivity. As sake of comparison, the electrical results of neat P3OT LS film are presented. Therefore, there

 Table 3. Electrical conductivities for LB and LS films with different

 Stearic Acid Percentages.

wt.%of SA	Conductivitys (S/m)		
	РЗОТ		
74	1.25 · 10-6		
54	$2.66 \cdot 10^{-6}$		
34	1.67 10-6		
0 LS	$1.39 \cdot 10^{-6}$		
54 LS	$1.95 \cdot 10^{-6}$		

is an optimum result, since the insulating behavior of SA³¹ may be interfering in one of the cases (74wt.%), and film quality and organization are compromised because there is not SA enough to enhance LB deposition (34 wt.%). Then, for comparison with these results, LS films were prepared of neat and P3OT/SA (54 wt.%), in order to investigate the influence of the deposition technique on the results. The σ_{dc} of neat P3OT LS is similar to P3OT/SA(74 wt.%) and the σ_{dc} of P3OT/SA(54 wt.%) LS is lower than P3OT/SA(54 wt.%) LB film. These outcomes demonstrate that LB film ordered structure is responsible for the improvement on conductivity. Kumar et al.³² found a σ_{dc} of 1.1×10^{-6} S·m⁻¹ for neat P3OT deposited by spin-coating technique onto gold electrodes evaporated on both sides, forming a diode-like structure (Au/P3OT/Au), and conductivity of 8.2 \times 10⁻⁴ S·m⁻¹ when doped. Most results of literature presents electrical measurements for doped polythiophene films^{33,34}, which extensively increases the conductivity.

3.4. AFM morphology

To support the electrical outcomes, AFM measurements in contacting mode were performed on LB and LS films with 54 wt.%, since they revealed higher σ_{de} values. This characterization is important to observe the morphology significance on the results, and evaluate the topographic feature of the P3OT/54SA films at the molecular level. It is noteworthy the homogeneity of IDE substrate that revealed root mean square (rms) roughness of 2.8 nm. AFM topographic images of 30×30 μ m² for P3OT/54SA films are displayed in Figure 6. In LB film image (Figure 6a) it is possible to observe the presence of agglomerates with



Figure 6. AFM topographic images of a) LB and b) LS P3OT/54SA films with 25 layers.

large size and the roughness found for this film revealed a high value (143 nm). LS film image (Figure 6b) also shows the presence of agglomerates, however, they seem to be smaller and in greater amount, the roughness obtained exhibited considerable decrease (80.3 nm) when compared to the LB film. These images support the existence of P3OT aggregated islands on top of a SA Langmuir film. Thus, it seems that higher roughness lead to higher conductivity values, corroborating previous work³⁵.



Figure 7. Normalized current responses for the different analytes tested as function of time.

3.5. VOCs sensors

Normalized current changes for P3OT/54SA sensors upon exposure to different VOCs are shown in Figure 7. The films with other percentages presented similar results, changing only intensities. As shown by Arshak et al.³⁶ through normalization, it is possible to analyze and compare data with different initial currents. Thus, normalizations were carried out using the sensor current immediately before the passage of VOCs (t ~15 min). For all vapors tested, the sensors response to VOCs was almost immediate and it was obtained a negative response, demonstrating that polythiophene films conductivity decreases in the presence of VOCs, as confirmed by previous studies^{19,37}. THF vapor showed the maximum response, while dichloromethane vapor presented an intermediary response, but its current starts increasing before turning-off the vapor. Toluene, non-polar and weak solvent for P3OT, showed the minimum response. The sensor currents did not return to the initial value after the passage of VOCs, even being in the presence of only nitrogen for 15 minutes. THF sensor shows the less reversible process among the three VOCs. On the other hand, dichloromethane sensor process is almost entirely reversible.

The negative behavior can be attributed to the swelling of polymer chains, which is coherent due to the fact tested VOCs are P3OT solvents. The presence of analyte molecules among polymeric chains are capable of modifying π - π * interactions among the molecules of conjugated



Figure 8. Normalized currents saturation of P3OT/SA (54wt.%) LB films in the presence of VOCs.

polymers. These interactions occur due to the aggregation phenomenon, tendency of conjugated chromophores remain together in solution and especially in film form³⁸. This swelling may increase the distance between polymer chains, and therefore, decrease the hopping conduction. This process can induce a conformational change in polymer chains and in some cases cannot be fully reversed with the presence of only nitrogen flow. To summarize, once THF is a polar analyte and a strong P3OT solvent, it can penetrate the film and expand the space between polymeric chains, decreasing the conductivity, and show a higher response.

Aiming test sensors selectivity, one P3OT/54SA LB film was exposed to each of the three VOCs individually, in order to analyze if after detecting one of the vapors, the sensor could still detect the others, and if it does, verify a possible modification caused by this process. For the selectivity test it was used all the possible sequences of the VOCs to check eventual changes in the sensor response. From the results, it could be noticed that sensors can detect one VOC, even after have detected others before and showing the same behavior previously presented. Hence, the P3OT/54SA sensor exhibited vapor selectivity for the tested VOCs. Some further tests were performed in order to verify the capacity of the sensors response to the VOCs after the first exposure.

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Figure 8 shows the results for the P3OT/54SA sensor under THF, toluene or dichloromethane VOCs. Despite the lack of full recovery, the sensors have sensitivity to a new passage of vapor. It is probably related to the fact that sensitive areas were not be completely inactivated, but decreased with the continued exposure to VOCs.

4. Conclusions

It was possible to fabricate a good quality of P3OT/ SA LB films, which was also very reproducible. These LB films revealed higher dc conductivity values than LS films, showing that the organization and roughness are more favorable for the electrical properties. Among LB films, P3OT/SA with 54wt.% SA showed the best result due to the optimization between the quality of deposition and the semiconducting characteristics of P3OT, according to SA percentage. The P3OT/SA films demonstrated promising results for their use as VOCs sensors.

Acknowledgments

The authors are grateful for the financial support of CAPES, FAPESP, CNPq, INEO and LNLS (Brazil) for providing interdigitated electrodes (project LMF 16557).

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