

Franciele Longaray Bernard^a, Leonardo Moreira dos Santos^a, Fabiana Willers Cobalchina^a,

Marina Barbieri Schwab^a, Sandra Einloft^{a,b}*¹⁰

^aEscola Politécnica, Pontificia Universidade Católica do Rio Grande do Sul (PUCRS), Porto Alegre, RS, Brasil ^bPrograma de Pós-Graduação Engenharia e Tecnologia de Materiais (PGETEMA), Pontificia Universidade Católica do Rio Grande do Sul (PUCRS), Porto Alegre, RS, Brasil

Received: December 7, 2018; Revised: August 01, 2019; Accepted: August 15, 2019

The development of both versatile and inexpensive sorbents for CO_2/CH_4 separation has become one of the greatest challenges to the environment and natural gas processing. This study reports the preparation and characterization of polyurethane (PU)/ cellulose based poly(ionic liquid)(CPIL) composites for CO_2/CH_4 separation. PU matrix was reinforced with CPIL in the range of 10-30 wt%. Several characterization techniques (TGA, DSC, DMTA and FESEM) were used to study the physical properties of composites when the PU matrix is reinforced with cellulose based poly (ionic liquids) (CPIL) up to 30%. CO_2 sorption, reusability and CO_2/CH_4 selectivity were assessed by pressure-decay technique. Results showed that CPIL addition in PU matrix promoted the increase in both thermal stability and mechanical properties when compared to PU. The best result for CO_2 sorption (35.0 mgCO_2/g) was obtained for PU/CPIL-TBP 10% which presented a higher sorption value when compared to PU (24.1 mgCO_2/g) and PU/CELLULOSE 10% (26.8 mgCO_2/g). PU/ CPIL-TBP 20% demonstrated higher CO_2/CH_4 selectivity. PU/CPIL composites appear as promissory materials for CO_2 capture. These compounds combine the benefits of ionic liquids (ILs) (high ionic conductivity, chemical and thermal stability) and cellulose (thermal stability) with PU properties (mechanical stability, processing and tunable macromolecular design).

Keywords: Carbon dioxide, natural gas, composites, polyurethanes, cellulose-based poly(ionic liquids).

1. Introduction

Carbon capture technologies from natural gas appear as one of main strategies to mitigate global warming, meet fuel performance requirements for a given application and prevent corrosion problems in pipeline¹. The development of both versatile and inexpensive sorbents for CO₂ capture has been one of the most relevant challenges in this field².

Chemical absorption processes using aqueous alkanolamine solutions have been extensively used in industry during the recent years^{3–5}. Using these solvents in capture processes have shown some drawbacks, including amine degradation/ evaporation, equipment corrosion and high energy penalty for solvent regeneration^{3–5}. It has been demonstrated that the use of solid adsorbents e.g. Poly(ionic liquid)s (PILs) present advantages compared to aqueous alkanolamine solutions as such as elimination of corrosion problems, reversible CO₂ sorption/desorption performance and low energy for sorbent regeneration^{2,6–11}. However, high cost is the main drawback as compared with alkanolamines.

Poly(ionic liquid)s (PILs) are an emerging new subclass of polyelectrolytes containing each repeated ionic unit connected

through a polymeric backbone to form a macromolecular structure^{12,13}. PILs syntheses are generally performed via direct ionic liquid monomer polymerization or chemical modification of existent polymers through ion exchange^{11–15}.

Our previous work¹⁶ reported the chemical modification of cellulose fibers extracted from rice husk with different ionic liquid cations (imidazolium, phosphonium, ammonium and pyrrolidinium). These cellulose based poly(ionic liquids) (CPIL) obtained from agroindustrial residues (rice husk) can be a promising alternative for CO₂ capture because they combine the benefits of ILs (high ionic conductivity, chemical and thermal stability) with waste reduction. The countercation introduction into cellulose structure promoted CO_2 sorption increase and a completely reversible CO_2 sorption/desorption process.

Polyurethanes (PUs) are an important class of polymers containing urethane group as the major repeating unit. Nevertheless other groups such as esters, urea, ethers and aromatic may also be present in the structure. PUs are versatile materials and widely used in industry^{17,18}. Literature describes that the introduction of polar groups (usually O, N) into the polymer structure may be an effective way to promote CO₂ affinity^{19,20}.

PU formulation can be tuned to deliver desirable properties in specific applications²¹. Furthermore, composites may also be used as alternative to improve mechanical and thermal properties of PU^{21–23}. The use of cellulose as reinforcing filler in PUs to promote improvements of both mechanical and thermal properties have been reported^{21–23}. Cellulose based poly(ionic liquid) use aiming to obtain PU/CPIL composites are not described in literature needing further studies.

Herein, we report for the first time the preparation and characterization of PU/CPIL composites. The effect of cation present in CPILs structure (1-butyl-3-methylimidazolium (BMIM), tetrabutylammonium (TBA), tetrabutylphosphonium (TBP) and 1-butyl-1-methylpyrrolidinium (BMPYRR) as well as CPIL concentration on the CO₂ sorption capacity, and both thermal and mechanical properties of the composites were studied. Our study also investigated CO₂/CH₄ separation.

2. Experimental

2.1 Materials

Synthesis and characterization of cellulose-based poly(ionic liquids) (CPILs) (1-butyl 3-methylimidazolium - CPIL-BMIM, tetrabutylammonium-CPIL-TBA, tetrabutylphosphonium-CPIL-TBP and 1-butyl-1-methylpyrrolidinium-CPIL-BMPYRR), were previously described by our group ¹⁶. ILs cations TBPB, TBAB, BMPYRR and BMIM were inserted into the cellulose structure in contents of 1.16×10^{-4} mol/g, 0.68×10^{-4} mol/g, 1.57×10^{-4} mol/g and 0.5901×10^{-4} mol/g, respectively¹⁶. Surface area values obtained for CPLIs TBP, TBA, BMPYRR and BMIM were extremely small (0.6430 m²/g, 1.7370 m²/g, 0.5901 m²/g, 0.6984 m²/g respectively¹⁶.

PU was synthesized following procedures adapted from literature²⁴. Poly(tetrahydrofuran) polyol (PTMG-2000 g/ mol, Sigma aldrich) (0.09 mol) and 0.1% wt of dibutyltin dilaurate (DBTDL, 95% Sigma aldrich) were dissolved in methyl ethyl ketone (50 mL) (MEK, P.A Dinâmica) in a five-necked flask. Then, hexamethylene diisocyanate (HDI, \geq 98.0%, Sigma Aldrich) (0.153 mol) was slowly added to the reaction mixture by a dropper funnel and stirred at 60°C for 90 min to obtain NCO-terminated polyurethane prepolymer. The NCO/OH ratio of 1.7 (0.153 mol HDI/0.09 mol OH) was used. The solid content was quantified as well as the polymer NCO free isocyanate (%NCO). In order to neutralize the NCO free isocyanate content, the system was cooled to 55°C and chain extender (1,6-Hexanediol) was charged into the reactor. Molecular weight (Mn) of 68000 g/mol and DPI = 1.5 was acquired from a Gel Permeation Chromatograph (GPC), equipped with a Waters 1515 pump and a Waters 2412 refractive index detector, using THF as eluent at a flow rate of 1 mL/min. PU/CPIL composites were labeled as PU/CPIL-X-Y, where X is cation, Y is CPIL concentration, For example, PU/CPIL-TBP-10% means TBP cation, 10%wt CPIL in PU matrix.

2.2 Preparation of composites

The PU and CPILs (PU/CPILs) were mixed in different ratios to obtain dry films of 0.15 mm thickness with CPILs content ranging up to 30 wt%. CPILs and PU chemical structures can be seen in Fig.1. Initially, 30%wt PU solution was prepared by dissolving 7.5 g PU into 25 ml dimethylformamide (DMF, P.A Dinâmica) via magnetic stirring until PU was completely dissolved (Fig.2I). In another glass bottle a suspension of CPILs was obtained using magnetic stirring in dimethylformamide for 6 h (Fig.2II). The mixtures were magnetically stirred for one day (Fig.2III). They were then sonicated by means of high power ultrasound disperser during 20 min (Fig.2IV). Finally, films with a thickness close to 0.15 mm were produced by casting and dried under vacuum at 60°C for 72 h (Fig.1IV).

2.3 Characterization

Samples were characterized by Universal Attenuated Total Reflectance sensor (UATR-FTIR) using a Perkin-Elmer Spectrum One FTIR Spectrometer, 4000 - 650 wavenumber range. Field emission scanning electron microscopy (FESEM) was performed using a FEI Inspect F50 equipment in secondary electrons (SE) mode. Differential Scanning Calorimetry (DSC) thermograms were attained by using a TA Instrument Q20 differential scanning calorimeter in the range of -90°C-170°C, or 200°C at a heating rate of 10°C/min under nitrogen. Thermogravimetric Analysis (TGA) was performed using a TA Instrument SDT-Q600 between 25°C and 600°C at a heating rate of 20°C/min in a nitrogen atmosphere. Tensile tests (stress x strain curves) were carried out at 25°C with rectangular shape films (12 mm long; 7 mm wide) with a thickness close to 0.15 mm, on a DMTA equipment (model Q800, TA Instruments) with 1 N/min. The Young Modulus of materials was determined according to ASTM D638. The analyses were carried out in triplicate.

2.4 Sorption experiments

2.4.1 CO₂ sorption measurements

A dual-chamber gas sorption cell was used to measure CO2 sorption by pressure-decay technique²⁵. The experiments were carried out in triplicate. Samples (Ws≈1g) were previously degassed at 70°C (343.15K) during 2 h. CO₂ sorption measurements were carried out at 25° C (298.15 K) and 0.1 MPa. A detailed description of sorption apparatus and measuring procedure can be found in previous works ²⁶⁻²⁸. Five CO₂ sorption/desorption cycles were also performed in PU/CPIL composites. CO₂ sorption was evaluated at 25°C (298.15 K) and 0.1 MPa with desorption following each cycle using heating 70°C (343.15K) during 2h.





Figure 1. CPILs (a) and PU (a) chemical structure.



3

Figure 2. Preparation scheme of composites

2.4.2 CO_2/CH_4 separation selectivity

The separation of CO₂ from CO₂/CH₄ gas mixture (35 mol % of CO₂ and CH₄ balance) also was determined using a dual-chamber gas sorption cell by pressure-decay technique which has been previously described in detail ^{27,29-31}. Samples (Ws≈1g) were also previously degassed at 70°C (343.15K) during 2 h. Selectivity experiments were also carried out in triplicate at 25°C and 2.0 MPa.

3. Results and Discussion

PU/CPIL-TBP composites were chosen to be characterized and to study the effects of CPIL addition in PU on both thermal and mechanical properties due to higher CO_2 sorption capacity compared to PU. The PU/ CELLULOSE 10% composite was also characterized for comparative purposes.

FTIR analysis was used to identify polymer formation and the effect of CPIL presence in PU matrix. The FTIR spectra of PU, PU/CELLULOSE-10% and PU/CPIL-TBP composites are shown in Fig.3. The completion of urethane reaction can be observed by the absence of band at about 2270 cm⁻¹ corresponding to free NCO group stretching vibration ^{32,33}. The N-H absorption region in spectrum indicates the presence of band at about 3322 cm⁻¹ associated with the N-H stretching of bonded hydrogen34,35. Band area at 3322 cm⁻¹ tends to increase with both the incorporation and concentration of fillers in PU matrix. Bands at about 1720 and 1693 cm⁻¹ are attributed to "free" C=O stretching and 'bonded hydrogen' C=O stretching, respectively³⁴. The band area increase at about 1693 cm⁻¹ upon the filler addition suggests a hydrogen bond formation increase. Band area at about 1693 cm-1 of PU/CPIL-TBP is higher than PU/CELLULOSE 10%. FTIR analysis also showed others characteristic PU and cellulose adsorption bands³⁶: 2936 - 2840 cm-1 (C-H), 1532 cm-1 (HN), 1246 cm-1 (C-N and C-O of urethane), 1100 cm-1 (C-O-C).

PU and PU composites thermal stability was investigated by thermogravimetric analysis (TGA) (Fig.4). PU thermal stability tended to increase with the addition of cellulose/ CPIL in PU matrix. TGA curves showed two stages of thermal degradation. The first thermal event is due to degradation of both hard segment (urethane bond)^{37–39} and cellulose/ CPIL structure¹⁶. T_{1onset} for PU, PU/CELLULOSE 10%, PU/ CPIL-TBP 10%, PU/CPIL-TBP 20% and PU/CPIL-TBP 30% occurred at 306.6°C, 344.2°C, 311.1°C, 311.0°C and 333.2°C, respectively. The second weight loss is associated to dissociation of soft segments (PTMG polyol)⁴⁰. T_{2 onset} for PU, PU/CELLULOSE 10%, PU/CPIL-TBP 10%, PU/ CPIL-TBP 20% and PU/CPIL-TBP 30% occurred at 450.6°C, 456,13°C, 458.6°C, 451.5°C and 447.6°C, respectively.



Figure 4. TGA thermograms of PU and PU composites.

DSC curves (Fig. 5) obtained for PU and PU composites showed an endothermic peak (T_m) (PU =19°C, PU/CELLULOSE 10% = 12°C, PU/CPIL-TBP 10% = 13°C, PU/CPIL-TBP 20%= 21°C and PU/CPIL-TBP 30%= 21°C) associated to the melting of crystalline microphase due to molecular weight (Mn= 2000) of soft segments (PTMG) promoting crystallization^{41,42}.



Figure 3. FTIR spectra of PU and PU composites



Figure 5. DSC thermograms obtained for PU and PU composites.

DSC curves also showed an exothermic peak related to crystallization of microphase (T_c) was also observed for all samples (PU=-32.0°C, PU/CELLULOSE 10%=-29.0°C, PU/CPIL-TBP 10%=-30.0°C, PU/CPIL-TBP 20%=-31°C and PU/CPIL-TBP 30%= - 29°C). Melting and crystallization enthalpy tended to decrease with the addition of cellulose/CPIL in PU matrix (PU $\Delta H_m = 36.5 \text{ J/g}$ and $\Delta H_c = 36.4 \text{ J/g}$; PU/CELLULOSE 10% $\Delta H_m = 18.4 \text{ J/g}$ and $\Delta H_c = 14.6 \text{ J/g}$; PU/CPIL-TBP 10% $\Delta H_m = 20.4 \text{ J/g}$ and $\Delta H_c = 17.8 \text{ J/g}$; PU/CPIL-TBP 20% $\Delta H_m = 23.3 \text{ J/g}$ and $\Delta H_c = 15.2 \text{ J/g}$; PU/CPIL-TBP 30% $\Delta Hm = 15 \text{ J/g}$ and $\Delta H_c = 9.24 \text{ J/g}$.

Homogeneous structures with a relatively uniform dispersion were obtained in PU composite filled with a fiber concentration up to 10 wt% (Fig.6). Holes (pores) were also observed after the filler addition in PU matrix. Tensile properties and Young moduli are presented in Figs 7 and 8. Mechanical properties increased with the filler addition in PU matrix. PU/CPIL-TBP 10% showed higher Young moduli than PU/CELLULOSE 10%. The presence of hydrogen bonding tends to enhance the mechanical properties of PU^{43,44}. FTIR band areas associated with the bonded hydrogens in PU/CPIL-TBP were higher than PU/CELLULOSE 10% (see Fig.3). However, the Young moduli reduced from 35 MPa in PU/CPIL-TBP 10% to 32.7 MPa in PU/CPIL-TBP 20%, probably due to homogeneity reduction observed by FESEM analysis (Fig.6). PU/CPIL-TBP 30% demonstrated fragility to perform tensile tests.

PU and PU composite CO_2 sorption capacity at 0.1 MPa and 298.15 K is shown in Fig.9. PU exhibited a CO_2 sorption capacity of 24.1mg CO_2 /g (1 bar) due to polar groups into the polymer structure that may promote CO_2 affinity ^{19,20}. CO_2 sorption increased after the filler addition in PU matrix. The cellulose structure also has polar groups (ether, ester and hydroxyl groups) that may promote interactions with $CO_2^{16,26}$. PU/CPIL CO_2 sorption capacity was higher than PU/CELLULOSE 10%, indicating that the carboxylate ion and IL countercation present in CPIL promote CO_2 sorption in PU matrix. The best result was found for PU/ CPIL-TBP 10% (35.0 mg CO_2 /g). Computational studies showed that TBP cation exhibit weaker coordination of carboxyl group promoting CO_2 sorption through electrostatic binding of CO_2 and carboxylate ion ^{16,45}.



Figure 6. Micrographs obtained for PU and PU composites. (a)PU, (b) PU/CELLULOSE 10%, (c) PU/CPIL-TBP 10%, (d) PU/CPIL-TBP 20% and (d) PU/CPIL-TBP 30%





Figure 7. Stress/strain curves obtained for PU and PU composites.







The effect of filler concentration increase in PU matrix on CO_2 sorption capacity is shown in Fig.10. CO_2 sorption decreased with the increasing content of CPIL-TBP in PU matrix. This behavior can be associated with increase of hydrogen bonds (see Fig.3) and/or homogeneity reduction of PU composite (see Fig.6) that may reduce PU- CO_2 interactions.



Figure 10. PU/CPIL - TBP 10% PU/CPIL - TBP 20% PU/CPIL - TBP 30% **Figure 10.** PU/CPIL-TBP composites CO_2 sorption capacity values at 0.1 MPa and 298.15 K.

Table 1 presents the comparison of CO_2 sorption capacity of PU/CPIL - TBP -10% with competitive PILs described in literature. At comparable temperatures and pressures, PU/CPIL-TBP 10% CO_2 sorption capacity is higher when compared to reported PILs. Results suggest that PU/CPIL-TBP 10% presents potential for CO_2 capture.

PU/CPIL-TBP 10% stability was evaluated over five CO_2 sorption/desorption cycles (Fig.11). PU/CPIL-TBP 10% sorption capacity was constant for all cycles indicating that PU/CPIL-TBP 10% sorbent offers necessary stability for CO_2 capture processes. This result evidences a typical behavior of a physical sorbent. Moreover, the advantage in this case is the low temperature needed for desorption process.

PU composites CO_2/CH_4 selectivity results are presented in Fig.12. PU/CPIL-TBP composites showed higher selective response than PU/ CELLULOSE 10%. Preferential affinity of CO₂ for PU/CPIL-TBP composites compared to PU/ CELLULOSE 10% is probably due to strong interactions between CO₂ and carboxylate ion of CPIL -TBP ⁴⁵.

Table 1. Comparison of PU/CPIL - TBP -10% CO, sorption capacity values with competitive PILs.

Sorbent	CO2 sorption (mg/g)	Conditions (P, T)	Ref.
P[VBIH][PF ₆]	3.22	0.079 bar, 295.15K	6
P6 [BIEMA][acetate]	12.46	0.1 MPa, 278.15K	46
P[VBTEA][PF ₆]	14.04	0.1 MPa, 278.15K	47
PIL-8.1.BF ₄	24.76	0.1 MPa, 273.15K	48
PU-TBP	15.70	0.082 MPa, 303.15K	45
PUIS-02-TBP	26.70	0.082 MPa, 303.15K	28
PU/CPIL-TBP-10 %	35.00	0.082 MPa, 298.15K	This study



Figure 11. PU/CPIL-TBP 10% CO₂ sorption/desorption cycles.



Figure 12. PU composites CO_2/CH_4 selectivity values at 2.0 MPa and 298.15 K.

Results provide evidence that the addition of CPIL-TBP can increase the selectivity of PU for CO₂ over CH₄. PU/CPIL-TBP 20% composite presented higher CO₂/CH₄ selectivity as compared with polyurethane-based poly(ionic liquid) s (PILPC-TBP: CO₂/CH₄ selectivity of ~2.22 at 303.15 K and 2 MPa)⁴⁹ and PU foam/ILs composite (PUF BF4 40: CO₂/CH₄ selectivity of 1.42 at 3 MPa)²⁹.

4. Conclusions

PU/ CPIL composites for CO₂ capture from natural gas were successfully prepared. PU/CPIL composites showed improvement in both thermal stability and mechanical properties when compared to PU. The best CO₂ sorption result was obtained for PU/CPIL - TBP -10%. PU/CPIL -TBP 10% CO₂ sorption capacity is highest then competitive PILs described in literature. PU/CPIL - TBP 20% Selective capacity for CO₂/CH₄ was higher than PU/CPIL - TBP 10%.

5. Acknowledgments

Sandra Einloft thanks CNPq for research scholarship.

6. Reference

- Kazemi A, Joujili AK, Mehrabani-Zeinabad A, Hajian Z, Salehi R. Influence of CO₂ residual of regenerated amine on the performance of natural gas sweetening processes using alkanolamine solutions. *Energy and Fuels*. 2016;30(5):4263-4273.
- Zulfiqar S, Sarwar MI, Mecerreyes D. Polymeric ionic liquids for CO₂ capture and separation: potential, progress and challenges. *Polymer Chemistry*. 2015;6(36):6435-6451.
- McGurk SJ, Martin CF, Brandani S, Sweatman MB, Fan X. Microwave swing regeneration of aqueous monoethanolamine for post-combustion CO₂ capture. *Applied Energy*. 2017;192:126-133.
- Yu CH, Hung CH, Tan CS. A review of CO₂ capture by absorption and adsorption. *Aerosol and Air Quality Research*. 2012;12:745-769.
- Sanz-Pérez ES, Murdock CR, Didas AS, Jones CW. Direct capture of CO₂ from ambient air. *Chemical Reviews*. 2016;116(19):11840-11876.
- Tang J, Sun W, Tang H, Radosz M, Shen Y. Enhanced CO₂ absorption of poly(ionic liquid)s. *Macromolecules*. 2005;38(6):2037-2039.
- Tang J, Tang H, Sun W, Radosz M, Shen Y. Low-pressure CO₂ sorption in ammonium-based poly(ionic liquid)s. *Polymer*. 2005;46:12460-12467.
- Tang J, Tang H, Sun W, Plancher H, Radosz M, Shen Y. Poly(ionic liquid)s: a new material with enhanced and fast CO₂ absorption. *Chemical communications*. 2005;(26):3325-7.
- Blasig A, Tang J, Hu X, Shen Y, Radosz M. Magnetic suspension balance study of carbon dioxide solubility in ammoniumbased polymerized ionic liquids: Poly(*p*-vinylbenzyltrimethyl ammonium tetrafluoroborate) and poly([2-(methacryloyloxy) ethyl] trimethyl ammonium tetrafluoroborate). *Fluid Phase Equilibria*. 2007;256(1-2):75-80.
- Privalova EI, Mäki-Arvela P, Murzin DY, Mikkhola JP. Capturing CO₂: conventional versus ionic-liquid based technologies. *Russian Chemical Reviews*. 2012;81(5):435-457.
- Einloft S, Bernard FL, Vecchia FD. Capturing CO₂ with Poly(Ionic Liquid)s. In: Eftekhari A, editor. *Polymerized Ionic Liquids*. Belfast: Royal Society of Chemistry; 2017. p. 489-514.
- Yuan J, Mecerreyes D, Antonietti M. Poly(ionic liquid)s: An update. Progress in Polymer Science. 2013;38(7):1009-1036.
- Yuan J, Antonietti M. Poly(ionic liquid)s: Polymers expanding classical property profiles. *Polymer*. 2011;52(7):1469-1482.
- Tomé LC, Marrucho IM. Ionic liquid-based materials: a platform to design engineered CO₂ separation membranes. *Chemical Society Reviews*. 2016;45(10):2785-2824.
- Shaplov AS, Morozova SM, Lozinskaya EI, Vlasov PS, Gouveia ASL, Tomé LC, et al. Turning into poly(ionic liquid)s as a tool for polyimide modification: synthesis, characterization and CO₂ separation properties. *Polymer Chemistry*. 2016;7(3):580-591.

- Bernard FL, Rodrigues DM, Polesso BB, Donato AJ, Seferin M, Chaban VV, et al. New cellulose based ionic compounds as low-cost sorbents for CO₂ capture. *Fuel Processing Technology*. 2016;149:131-138.
- Akindoyo JO, Beg MDH, Ghazali S, Islam MR, Jeyaratnam N, Yuvaraj AR. Polyurethane types, synthesis and applications – a review. *RSC Advances*. 2016;6:114453-114482.
- Chattopadhyay DK, Webster DC. Thermal stability and flame retardancy of polyurethanes. *Progress in Polymer Science*. 2009;34(10):1068-1133.
- Tomasko DL, Li H, Liu D, Han X, Wingert MJ, Lee LJ, et al. A review of CO₂ applications in the processing of polymers. *Industrial* and Engineering Chemistry Research. 2003;42(25):6431-6456.
- 20. Gabrienko AA, Ewing AV, Chibiryaev AM, Agafontsev AM, Dubkov KA, Kazarian SG. New insights into the mechanism of interaction between CO₂ and polymers from thermodynamic parameters obtained by *in situ* ATR-FTIR spectroscopy. *Physical Chemistry Chemical Physics.* 2016;18(9):6465-6475.
- Hadjadj A, Jbara O, Tara A, Gilliot M, Malek F, Maafi EM, et al. Effects of cellulose fiber content on physical properties of polyurethane based composites. *Composite Structures*. 2016;135:217-223.
- 22. Fortunati E, Luzi F, Janke A, Häußler L, Pionteck J, Kenny JM, et al. Reinforcement effect of cellulose nanocrystals in thermoplastic polyurethane matrices characterized by different soft/hard segment ratio. *Polymer Engineering and Science*. 2017;57(6):521-530.
- Vaithylingam R, Ansari MNM, Shanks RA. Recent advances in polyurethane-based nanocomposites: a review. *Polymer-Plastics Technology and Engineering*. 2017;56(14):1528-1541.
- 24. Lei W, Fang C, Zhou X, Cheng Y, Yang R, Liu D. Morphology and thermal properties of polyurethane elastomer based on representative structural chain extenders. *Thermochimica Acta*. 2017;653:116-125.
- Koros WJ, Paul DR. Design considerations for measurement of gas sorption in polymers by pressure decay. *Journal of Polymer Science: Polymer Physics Edition.* 1976;14:1903-1907.
- Bernard FL, Duczinski RB, Rojas MF, Fialho MCC, Carreño LÁ, Chaban VV, et al. Cellulose based poly(ionic liquids): Tuning cation-anion interaction to improve carbon dioxide sorption. *Fuel*. 2018;211:76-86.
- Duczinski R, Bernard FL, Rojas MF, Duarte E, Chaban V, Vecchia FD, et al. Waste derived MCMRH- supported IL for CO₂/CH₄ separation. *Journal of Natural Gas Science and Engineering*. 2018;54:54-64.
- Bernard FL, Polesso BB, Cobalchini FW, Chaban VV, Nascimento JF, Dalla Vecchia F, et al. Hybrid alkoxysilane-functionalized urethane-imide-based poly(ionic liquids) as a new platform for carbon dioxide capture. *Energy and Fuels*. 2017;31(9):9840-9849.
- 29. Rojas MF, Miranda LP, Ramirez AM, Quintero KP, Bernard F, Einloft S, et al. New biocomposites based on castor oil polyurethane foams and ionic liquids for CO₂ capture. *Fluid Phase Equilibria*. 2017;452:103-112.
- 30. Azimi A, Mirzaei M. Experimental evaluation and thermodynamic modeling of hydrate selectivity in separation of CO₂ and CH₄. *Chemical Engineering Research and Design.* 2016;111:262-268.

- Campbell S, Bernard FL, Rodrigues DM, Rojas MF, Carreño LÁ, Chaban VV, et al. Performance of metal-functionalized rice husk cellulose for CO₂ sorption and CO₂/N₂ separation. *Fuel*. 2019;239:737-746.
- Deligöz H, Yalcınyuva T, Özgümüs S. A novel type of Si-containing poly(urethane-imide)s: synthesis, characterization and electrical properties. *European Polymer Journal*. 2005;41(4):771-781.
- 33. Song J, Chen G, Ding Y, Shi J, Liu Y, Li Q. Preparation and characterization of epoxy resin modified with alkoxysilanefunctionalized poly(urethane-imide) by the sol-gel process. *Polymer International*. 2011;60(11):1594-1599.
- Coleman MM, Lee KH, Skrovanek DJ, Painter PC. Hydrogen bonding in polymers. 4. Infrared temperature studies of a simple polyurethane. *Macromolecules*. 1986;19(8):2149-2157.
- Zhang M, Hemp ST, Zhang M, Allen Junior MH, Carmean RN, Moore RB, et al. Water-dispersible cationic polyurethanes containing pendant trialkylphosphoniums. *Polymer Chemistry*. 2014;5(12):3795-3803.
- 36. Lee HT, Wu SY, Jeng RJ. Effects of sulfonated polyol on the properties of the resultant aqueous polyurethane dispersions. *Colloids and Surfaces A: Physicochemical and Engineering Aspects.* 2006;276(1-3):176-185.
- 37. Cervantes-Uc JM, Espinosa JIM, Cauich-Rodríguez JV, Ávila-Ortega A, Vázquez-Torres H, Marcos-Fernández A, et al. TGA/ FTIR studies of segmented aliphatic polyurethanes and their nanocomposites prepared with commercial montmorillonites. *Polymer Degradation and Stability*. 2009;94(10):1666-1677.
- Petrović ZS, Zavargo Z, Flyn JH, Macknight WJ. Thermal degradation of segmented polyurethanes. *Journal of Applied Polymer Science*. 1994;51(6):1087-1095.
- Barikani M, Fazeli N, Barikani M. Study on thermal properties of polyurethane-urea elastomers prepared with different dianiline chain extenders. *Journal of Polymer Engineering*. 2013;33(1):87-94.
- Pashaei S, Siddaramaiah, Syed AA. Thermal degradation kinetics of polyurethane /organically modified montmorillonite clay nanocomposites by TGA. *Journal of Macromolecular Science*, *Part A*. 2010;47(8):777-783.
- 41. Król P. Synthesis methods, chemical structures and phase structures of linear polyurethanes. Properties and applications of linear polyurethanes in polyurethane elastomers, copolymers and ionomers. *Progress in Materials Science*. 2007;52(6):915-1015.
- Kojio K, Nakashima S, Furukawa M. Microphase-separated structure and mechanical properties of norbornane diisocyanatebased polyurethanes. *Polymer*. 2007;48(4):997-1004.
- Han W. Synthesis and properties of networking waterborne polyurethane/silica nanocomposites by addition of poly(ester amine) dendrimer. *Polymer Composites*. 2013;34(2):156-163.
- 44. Wang Y, Cheng Z, Liu Z, Kang H, Liu Y. Cellulose nanofibers/ polyurethane shape memory composites with fast water-responsivity. *Journal of Materials Chemistry: B.* 2018;6(11):1668-1677.
- 45. Bernard FL, Polesso BB, Cobalchini FW, Donato AJ, Seferin M, Ligabue R, et al. CO₂ capture: Tuning cation-anion interaction in urethane based poly(ionic liquids). *Polymer*. 2016;102:199-208.

- 46. Privalova EI, Karjalainen E, Nurmi M, Mäki-Arvela P, Eränen K, Tenhu H, et al. Imidazolium-based poly(ionic liquid)s as new alternatives for CO₂ capture. *ChemSusChem.* 2013;6(8):1500-1509.
- 47. Yu G, Li Q, Li N, Man Z, Pu C, Asumana C, et al. Synthesis of new crosslinked porous ammonium- based poly(ionic liquid) and application in CO₂ adsorption. *Polymer Engineering and Science*. 2014;54(1):59-63.
- Morozova SM, Shaplov AS, Lozinskaya EI, Mecerreyes D, Sardon H, Zulfiqar S, et al. Ionic polyurethanes as a new family of poly(ionic liquid)s for efficient CO₂ capture. 2017;50(7):2814–2824.
- Bernard FL, Santos LM, Schwab MB, Polesso BB, Nascimento JF, Einloft S. Polyurethane-based poly(ionic liquid)s for CO₂ removal from natural gas. *Journal of Applied Polymer Science*. 2019;136(20):47536.