Bioactive TiO₂ Fibers Prepared by Solution Blow Spinning: A Promising Approach for **Microbial Control**

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PEO/TiP fibers were obtained using the Solution Blow Spinning (SBS) apparatus and heat treated to produce $\rm TiO_2$ fibers. The morphological and structural characteristics were assessed using SEM and X-ray diffraction. The fibers, with a thickness of 12 μm, showed a change in crystalline structure with heat treatment. At temperatures as low as 800 °C, only the anatase phase was identified, while at 900 °C, both anatase and rutile phases coexisted. The addition of TiP to the polymer matrix reduced the initial breakdown temperature, and the DSC curves showed exothermic peaks due to the amorphous phase transition to TiO₂/anatase. The fibers' photocatalytic capacity was tested, revealing that TiO₂fibers in the anatase phase achieved 97% degradation of Rhodamine-B dye in 40 minutes. The study found that the biocide efficacy of TiO₂-fibers depends on their heat treatment. Fibers with anatase/rutile or pure rutile phases did not show significant efficiency. However, fibers treated at 600°C with pure anatase phase were more effective in eliminating E. coli and total coliforms. Finally, we can state that the TiO₂ fibers obtained in this work using the SBS technique can be used to produce filters to purify water contaminated by pathogens dangerous to human health or even to purify the air.

Keywords: TiO₂*fibers, solution blow spinning, photocatalytic property, PEO/TiP.*

1. Introduction

Photocatalysis has emerged as a promising technology for degrading organic pollutants, dyes, and antibacterial applications¹⁻⁴. Currently, this technology is widely used in water treatment⁵⁻⁸. Several broadband semiconductors, such as TiO_2^9 , WO_3^{10} , Cu_2O^{11} , and CdS^{12} have been employed in photocatalytic processes to achieve this goal. In semiconductors $TiO₂¹³⁻¹⁸$ is widely used due to its environmental friendliness, high removal rate⁹, low toxicity, low cost¹⁹, and high photocatalytic activity²⁰. Photocatalytic fiber semiconductors have recently been used for wastewater treatment due to their adjustable pore architectures, huge specific surface areas, and nanoscale and micrometer diameters²¹⁻²⁷. Fibers may be created using various techniques, including electrospinning, phase separation, self-assembly, and stretching^{28,29}.

Electrospinning is a well-established technique for polymeric production of fibers from melt polymers or polymer solutions25,30-32. The method depends on the polymer solution's dielectric properties and high-voltage electric field to produce the polymer fibers.

The amount of "free" charge that can be induced in a polymer solution during electrospinning is represented by the dielectric constant of the solvent. Low dielectric constant polymer-solvent solutions limit the initiation of the whipping instability and thinning of the polymer jet^{33,34}. Because of these constraints, the SBS technique has recently been highlighted in producing polymeric micro/nanfibers³⁵⁻⁴¹.

Compared with electrospinning, the SBS process does not depend on the dielectric constant; fibers can be deposited onto any type of substrate or collector, and the fiber production rate is many times higher, thus having better commercial production potential³⁵. As a result, this study aimed to create $TiO₂$ photocatalytic microfibers by thermally treating polyethylene oxide/isopropoxide (PEO/ TiP) fibers generated by the SBS apparatus. The influence of thermal treatment on the morphology and crystalline structure of TiO_2 microfibers was examined first, followed by an assessment of the photocatalytic activity of microfibers as function of the degradation of the Rhodamine-B (RhB) dye under UV light. The efficiency of $TiO₂$ fibers in the death of bacteria such as E. coli and total coliforms was also evaluated. The results of this study can provide valuable information on the use of $TiO₂$ fibers in various applications, such as water treatment contaminated with pathogens harmful to human health and air purification by pathogens harmful to human *e-mail: fernando.r.paula@.unesp.br health and air purification.

2. Experimental Procedure

2.1. Materials

All chemicals used in this work were purchased from Sigma-Aldrich and used as received: polyethylene oxide powder (PEO) (Average Mw = 5,000,000) titanium (IV) isopropoxide (TiP) (\geq 95.0%), RhB dye (\geq 95.0%), chloroform $(\geq 99.5\%)$, and absolute ethanol ($\geq 95.0\%$).

2.2. Extraction of pure titanium dioxide fibers

The precursor solution was prepared by adding 0.2 g of PEO to a mixture of 3.0 mL of absolute ethyl alcohol and 5.0 mL of chloroform. The mixture was stirred for 40 minutes at room temperature (25 $^{\circ}$ C). Then, 2.0 mL of TiP was added to the PEO solution which remained stirred for 40 minutes. The final solution was placed in disposable syringes connected to an 18G spinal needle and attached to the injection system (Figure 1). The most optimal conditions for forming of the microfiber were as follows: injection rate of 0.3 L min−1, work distance of 35 cm, and collector speed of 60 rpm. No gas was used to draw the solution to the collector. The (PEO/TiP) fibrous composites were thermally treated at 600, 700, 800, 900, and 1000 °C to remove the polymer component to yield $TiO₂$ fibers.

2.3. Characterization

The crystallinity and phase of the produced TiO_2 fibers were investigated using a Shimadzu X-ray diffractometer (model XRD-6000), Cu K α (1.54056 Å). The Scherrer equation and the X-ray diffraction pattern were used to calculate the average crystallite size. A scanning electron microscope (SEM, Zeiss EVO LS15) with a voltage range of 5.00 kV to 20.00 kV was used to investigate the morphological structure. Using the image analysis tool "Image J 1.45," the average diameter of the fibers was estimated. Ten milligrams of the sample were used for the thermogravimetric analysis, which was carried out using an SDT model Q 600 from TA instruments. The samples were heated between 25 and 800 °C at 10 °C min⁻¹ in a nitrogen environment with a flow rate of 100 mL min-1. The Varian Cary 50 Scan equipment captured UV-vis spectra to examine the peak decrease in maximum dye absorption.

2.4. Photocatalytic principles and activity evaluation

The band gap, which forms the foundation for heterogeneous photocatalysis, is the area between the semiconductor material's valence (VB) and conduction (CB) bands. Semiconductor activation may occur depending on the energy of the photons from artificial or natural light. According to Equation 1, an electron is promoted from VB to CB by absorption of a photon with energy larger than or equal to the band gap, causing a vacancy $(h+)$ to form in the VB. The positive potential of these vacancies might range from $(+2.0 \text{ to } +3.5 \text{ eV})$ depending on the semiconductor⁴². Figure 2 depicts a diagram showing how UV light activates TiO2 microfibers so that they can interact with contaminants such as RhB.

In the photocatalysis process, contaminants can be degraded when water molecules adsorbed on the surface of the semiconductor generate OH radicals (Equations 2 and 3)^{43,44}. In addition to the OH radicals, the degradation can also occur through oxygen derivatives that are formed when electrons are captured in the system (Equation 4)⁴⁴.

$$
TiO2 \rightarrow TiO2 (eCB- + hVB+)
$$
 (1)

$$
h^{+} + H_{2}O_{adsorption} \rightarrow HO^{+} + H^{+}
$$
 (2)

$$
h^{+} + OH_{adsorption}^{-} \rightarrow HO
$$
 (3)

$$
e^- + O_2 \rightarrow O_2^- \tag{4}
$$

In the development of this work, the photocatalytic activity of samples was evaluated in a photocatalytic reactor equipped with a UV mercury lamp (250 W) and a magnetic stirrer. For this purpose, 80 mg of TiO_2 fibers were added to 100 mL of RhB aqueous solution (10 mg/L). The mixture was stirred for 15 minutes at room temperature (25°C) in the absence of light. To start the photocatalysis process, the UV light was turned on, and every 10 minutes, 3 mL aliquot of the mixture was removed and then centrifuged at 6,000 RPM for 5 minutes to separate the microfibers from the solution. After this process, the solution was analyzed by

Figure 1. Experimental apparatus used by the SBS.

Figure 2. Photocatalytic activity of TiO₂ fibers in the presence of UV light in the RhB dye degradation process.

spectra UV-Vis, obtaining its absorption band and possible intensity reduction again.

The photocatalytic degradation efficiencies of samples were determined using the Equation 5

$$
Degradation (\%) = \frac{C_0 - C}{C_0} \times 100
$$
 (5)

where C_0 is the concentration of RhB after 15 minutes of agitation without UV exposure, and C is the RhB concentration after UV irradiation for a specific time.

2.5. Procedure for analyzing water contamination and the biocidal efficacy of TiO² fibers

Water samples were collected from a semi-artesian well contaminated with E. coli and total coliforms. The aim was to evaluate whether TiO_2 fibers could eliminate these bacteria. The water samples were filtered through a sterile nitrocellulose membrane with a diameter of 47 mm and a porous size of 1.45 μm to analyze the biological contaminants. This filter is designed to retain and encourage the growth of both bacteria types, E. coli (blue dots) and total coliforms (red dots) (see Figure 3a). To compare the effectiveness of TiO₂ fibers, an equivalent amount of fibers, 0.01 g, was applied to the central region of the contaminated membranes (see Figure 3b). All contaminated membranes were then exposed to a 9 W ultraviolet lamp positioned 30 cm away for a brief 2-second irradiation period. After exposure, the membranes were incubated at 36 °C for 28 hours to allow any bacterial growth. Finally, the bacterial colony formation procedure was used to evaluate qualitatively both the treated and untreated membranes, revealing the impact of the $TiO₂$ fibers.

3. Results and Discussion

3.1. TiO₂ fibers morphology characterization

 $PEO/TiO₂$ fibers were produced using a blow spinning apparatus. The precise combination of injection rate, collector rotation, and working distance made producing precursor fibers without carrier gas possible. These parameters are injection rate of 53 μL, collector rotation of 60 rpm, and working distance of 35 cm. The injected PEO/TiO_2 solution presented the shape of a continuous thread that was manually taken from the tip of the nozzle to the collector, and then the rotating collector promoted the process of stretching the polymeric solution.

Figure 4 shows a SEM image of $PEO/TiO₂$ fibers and $TiO₂$ fibers resulting from the calcination of PEO/TIP at 600 °C. Fiber diameters were measured by analyzing SEM images and using ImageJ software. The distribution histograms and the calculated average diameters are shown inset in the figures. The diameter of the precursor fibers (Figure 4a) is larger than that of the calcined fibers (Figure 4b). This occurs due to the decomposition of the PEO polymer matrix. Heat treatment simultaneously removes organic components while promoting the growth of $TiO₂$ crystals due to oxidation of the Ti precursor.

A similar decrease in fiber diameter was also observed by Tan et al.⁴⁵, who prepared $TiO₂$ fibers using the SBS technique with Polyvinylpyrrolidone (PVP)/TiP. However, unlike the results shown in Figure 4, the fibers obtained after heat treatment were brittle and had reduced length compared to untreated fibers.

Figure 5 demonstrates the impact of varying temperatures on $TiO₂$ fibers during the heat treatment. Despite temperature changes, the fibers maintained their original form.

Figure 3. (a) Nitrocellulose membrane contaminated with both E. Coli (blue dots) and total coliforms (red dots) bacteria's and, (b) contaminated membrane with the presence of TiO_2 fibers.

Figure 4. SEM images of $PEO/TiO₂$ precursor fiber (a) and (b) after heat treatment at 600 °C.

The production method allowed the development of some hollow structures, as shown in Figures 4b and 5d. It is evident from a comparison of Figures 4 and 5 that the adherence of fibers during the sintering process may cause this morphological shape. The presence of these hollow structures presents significant advantages, as it significantly increases the accessible surface area of the system, making it ideal for photocatalytic applications.

Fibers processed with SBS showed an average diameter of 11 to 13 μ m after heat treatment. This draws attention to a crucial difference between electrospinning and SBS. Electrospinning best produces nanofiber architectures, while SBS generally results in fibers with substantially larger diameters. Daristotle et al.⁴⁶ who investigated the morphological and mechanical characteristics of fibers made with SBS, support this observation.

3.2. Structural analysis

The X-ray diffraction (XRD) pattern of $TiO₂$ fibers is depicted in Figure 6. The process of heat treatment has a significant effect on the formation of crystallographic

phases. The peaks of the anatase phase are observed in the thermally treated fibers at 600, 700, and 800°C (main peak is at $2\theta = 25.3^{\circ}$). In contrast, the rutile phase peaks are identified in the fibers treated at 900 and 1000 °C (main peak is at $2\theta = 27.3^{\circ}$. These findings are consistent with the results reported in the literature that demonstrate using heat treatment temperatures to produce the TiO₂ phase^{47,48}.

To completely transform the $TiO₂$ fibers from the anatase phase into rutile, heat treatment at 1000 °C was necessary in this study. Usually, anatase permanently transforms into rutile when exposed to temperatures above 600 °C in an atmosphere of air. However, the transition temperature from the anatase phase to the rutile phase can vary from 400-1200 °C depending on the specific $TiO₂$ synthesis technique49,50. It is essential to evaluate the transition kinetics of these phases. Factors that should be considered include the shape and size of particles, atmosphere, surface area, sample volume, heating rate, sample container type, contaminants, and measurement technique⁴⁹.

The coexistence of anatase and rutile phases is observed in fibers treated at 900 °C. The fraction of anatase and rutile

Figure 5. SEM images of TiO₂ fiber after heat treatment for 4h: (a) 600° C; (b) 700° C; (c) 800° C; (d) 900° C, and (e) 1000° C.

phases found in TiO_2 fibers was calculated using Equation 6. The rutile and anatase phases present percentages of 64.36% and 35.64%, respectively. Table 1 presents the samples' average size of the TiO_2 crystallite calculated by the Scherrer equation and the percentages of anatase and rutile phases $51,52$.

$$
w_R = \frac{I_R}{0.884I_A + I_R},\tag{6}
$$

where W_R is the percentage of the rutile phase present in the sample, and I_R and I_A represent the intensities of the diffraction peaks characteristic of the rutile and anatase phases.

3.3. Thermal characterization

The thermogravimetric analysis (TGA) technique was utilized to gather information about the thermal stability and breakdown of the polymeric matrix of the PEO and the impact of TiP on the degradation process of the polymeric matrix. The TGA curves for pure PEO and PEO/TiP fibers are displayed in Figure 7a. A single event of mass loss was observed in the polymeric matrix (PEO), which started at around 327 °C, and the intensity decreased at approximately 405 °C, decomposing about 92% of the sample. After this

Figure 6. TiO₂ fibers XRD heat treated at different temperatures for 4 h.

event, the process continued steadily until it reached the end temperature of 800 °C. The stability of the sample is mainly due to the test being conducted in an inert atmosphere (nitrogen flow) 53 .

The main byproducts of PEO mass loss are methyl alcohol, ethyl alcohol, alkene, formaldehyde, non-cyclic ethers, ethylene oxide, water, acetic aldehyde, CO_2 , and $CO^{54,55}$. In the case of the PEO/TiP composite, the integration of the precursor resulted in a decrease in the initial temperature of the polymer matrix breakdown process. There are two distinct events observed in this composite between the temperature range of 24 °C to 116 °C, with a mass loss of approximately 8%, which is attributed to ethanol and water evaporation⁵⁶. The second event occurs between the temperature range of 197 °C to 327 °C, which is attributed to TIP decomposition and PEO degradation.

Figure 7b shows the DSC curves of the PEO matrix and PEO/TiP composite. The melting temperature of the polymer matrix (PEO) corresponds to an endothermic peak at 73 °C, while the decomposition of $PEO⁵⁴$ produces an endothermic peak at 380 °C. The PEO/TiP composite DSC curve shows an endothermic peak between 24 °C and 116 °C. This weight loss is due to the evaporation of water and organic solvents⁵⁷. The exothermic peaks at 305 °C and 498 °C are due to the phase transition from amorphous to anatase $TiO₂⁵⁸$.

Figure 7. (a) TGA and (b) DSC curves for PEO and PEO/TiP fibers.

Table 1. Percentage of anatase and rutile phases of TiO_2 fibers submitted to different heat treatments.

Heat treatment $(^{\circ}C)$	Percentage of phases $(\%)$		Average crystallite size (nm)	
Temperature	Anatase	Rutile	Anatase	Rutile
600	100		25.26	$- - -$
700	100		30.63	---
800	100		36.73	---
900	35.64	64.36	26.24	29.18
1000	θ	100	---	46.86

Figure 8. (a) Photodegradation of RhB by TiO₂ fibers and (b) First order kinetic model for photodegradation of RhB by TiO₂ fibers.

0.0888	0.9763
0.0479	0.9982
0.0482	0.9988
0.0238	0.9922
0.0045	0.9134

Table 2. RhB degradation constant by $TiO₂$ fibers.

3.4. Photocatalytic performance

Figure 8 shows first-order photodegradation and kinetic graphs for TiO_2 fibers exposed to UV light (a-b). Table 2 presents the values of the degradation rate constant k (min⁻¹) for each sample. As shown in Figure 8a, after 40 minutes of reaction, the degradation of the RhB dye was 100%, 83%, 83%, 60%, and 18% for fibers heat treated at 600, 700, 800, 900, and 1000 °C, respectively. The anatase phase was the only phase present in fibers heat-treated between 600 °C and 800 °C; however, it was observed that the photocatalytic activity decreased when fibers were treated at temperatures above 600 °C. The observed decrease in photocatalytic activity is attributed to a reduction in the surface area of crystallites caused by an increase in microfiber treatment temperature. Larger average crystallite sizes correspond to a smaller surface area, resulting in a diminished number of active sites available for photocatalytic reactions (Table 1) $59-62$.

It was found that for samples heat-treated at 900 °C and 1000 °C, the increase in rutile phase reduced the photocatalytic efficiency (Fig. 8a-b). The XRD patterns of TiO₂ fibers (Table 1) show increased rutile phase formation after heat treatment at higher temperatures, confirming this observation. The anatase phase generally has more significant photocatalytic activity than the pure rutile phase⁶³. It is believed that the anatase phase has improved characteristics due to a decreased rate of electron-hole recombination and a stronger adsorption affinity for organic molecules⁶⁴. However, combining different phases has been

found to increase the efficiency of photocatalysis. Although some samples heated to 900 °C showed the coexistence of phases, it cannot be assumed that a heterojunction occurred in the photocatalyst, which is a fundamental factor for increased photocatalysis. Therefore, evaluating the photocatalytic activity based solely on phase composition is misleading.

3.5. Antimicrobial activity of PVDF/TiO² fibers

The process of photocatalytic inactivation of bacteria through $TiO₂$ fibers obtained at different treatment temperatures is illustrated in Figure 9. When contaminated water comes into contact with $TiO₂$ fibers on the surface of the membrane, ultraviolet light generates electron-hole pairs and radicals free (•OH). The •OH radical is a potent toxin capable of killing bacteria. The formation of the O_2 radical can also cause an attack, but the •OH radical is the most reactive because it can oxidize many types of organic compounds, including microbial cells. A thymine dimer forms in the bacterial DNA chromosome when the •OH radical comes into contact with the bacterial cell wall. This dimer forms knots between thymine and the DNA base, obstructing double helix formation and interrupting normal DNA replication. As a result, the cell's blocked growth eventually leads to its death. After two seconds of exposure to UV light, the $TiO₂$ fibers positioned in the central region of the nitrocellulose membranes significantly reduced the number of bacteria, E. coli and total coliforms, as demonstrated in Figure 9. According to the experiment, the membrane containing $TiO₂$ fibers in the anatase phase subjected to heat treatment at 600 °C proved to be the most effective in eliminating bacteria. The experiment results confirmed that the bactericidal efficiency of $TiO₂$ fibers decreases as the heat treatment temperature increases. This decrease in efficiency is due to the increase in the rutile phase, which is less effective in eliminating bacteria. It is worth mentioning that, as shown in Figure 9, the incidence time of the ultraviolet rays used in the development of the work is insufficient to kill bacteria in places without a photocatalyst. Therefore, the presence of $TiO₂$ fibers activated by UV radiation is responsible for the bactericidal effect.

Figure 9. Sterilization efficiency results obtained for TiO₂ fibers thermally treated at different temperatures (600, 700, 800, 900 and 1000 °C) and exposed to UV radiation for 2 seconds.

4. Conclusion

In conclusion, this study demonstrates a successful method for producing PEO/Tip precursor fibers using the SBS technique without the need for pressurized gas. The best result was obtained with the sample calcined at 600 °C, which presented an anatase phase with a smaller crystallite size. With this sample, 97% of the RhB dye (aqueous solution) was eliminated after 40 minutes of exposure to UV light. According to the study involving the bactericidal activity of TiO₂ fibers, the membrane containing TiO₂ fibers in the anatase phase and treated at 600 °C was also considered the most effective for killing bacteria such as E. coli and coliforms. However, the bactericidal efficiency and photocatalytic efficiency of TiO_2 fibers decreased as the heat treatment temperature increased due to the increase in the less effective rutile phase. The results obtained in this work indicate that TiO_2 fibers produced with SBS technology have antibacterial properties, making them potential candidates for water purification applications contaminated with pathogens dangerous to human health.

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6. References

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