

Fracture Toughness and Charpy CVN Data for A36 Steel with Wet Welding

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Abstract: This study presents K_{Ic} data obtained from K_{Ic} -CVN correlations from Charpy CVN values. For this study, T-welded connections were manufactured from ASTM A36 and E6013 electrodes in dry conditions. Then, a rectangular grinding at the weld toe was carried out and filled with wet welding. Charpy specimens were extracted to obtain CVN values. An exhaustive search through the literature of several authors was performed to collect experimental CVN data about wet welding being applied to A36 steel for comparison with CVN data obtained in this study. By using Charpy impact energy (CVN), K_{Ic} values could be predicted by K_{Ic} -CVN correlations. In addition, correlations were presented to obtain K_{Ic} values in the lower shelf, transition temperature zones and different zones for the energy-temperature curve of A36 steel. Of these correlations, Barsom's equation was adopted, because he applied the stress yield (σ_{ys}) of the material and it can be applied in all zones for the energy-temperature curve. The results revealed that CVN values are proportionate to K_{Ic} , this data decreases as water depth increases. This took place because several discontinuities, such as, porosity, slag inclusion, non-metallic inclusion, cracking and microstructures are present in the wet welding.

Keywords: Fracture Toughness (K_{Ic}); Charpy impact energy (CVN); Wet welding; A36 Steel; Porosity.

1. Introduction

It's widely recognized that Charpy CVN impact energy values can be converted to K_{Ic} using K_{Ic} -CVN correlations [1-7]. In this context, plane-strain fracture toughness (K_{Ic}) is an important material property in the prediction and prevention of fracture, and for damage tolerance assessment of brittle materials [8]. The K_{Ic} in the linear elastic fracture mechanics (LEFM) is the size of the stress intensity factor at the tip of the crack if the strain in the body is elastic. The ASTM E-399 standard [9] is used to obtain K_{Ic} values in plane-strain for the displacement mode of the opening cracking. However, it is not always possible to prepare such specimens when the analyzed material does not have the proper dimensions [10], even at room temperature, standard tests for K_{Ic} are difficult, time-consuming and costly [8].

Charpy impact energy CVN is utilized to indirectly estimate CVN data for K_{Ic} values. Then, to obtain the K_{Ic} values from the CVN data, it is necessary to select the behavior from CVN impact results, according to the interest zone, σ_{ys} of the material and the CVN energy values. Although this CVN impact test data does not represent the real fracture toughness data, this data can be used as a series of points to estimate toughness in an evaluation of fracture mechanics. Nevertheless, there are studies in the literature about estimating K_{Ic} values from correlations of CVN impact data [11-17]. K_{Ic} -CVN correlations available in published literature are applied to lower-shelf, transition zone and upper-shelf temperatures and are not applicable to wet welding. However, many of these correlations are based on the yield stress (σ_{ys}) of the material, Young's Module (E), and correlations applied to different zones in a Charpy transition temperature curve. Thus, these correlations could be applied to different metals and wet welding.

The purpose of this study was to estimate the fracture toughness (K_{Ic}) of CVN data from different authors who employed wet welding in A36 steel and compared with CVN values obtained in this study. A36 steel and E6013 electrodes were utilized to construct T-welded connections. Then, a rectangular grinding was carried out at the weld toe



with two depths, 6 mm and 10 mm. The rectangular grinding was filled with wet welding simulating seawater at depths of 50, 70 and 100 m. Standard Charpy specimens were extracted to obtain energy values. In addition, a search of Charpy CVN values using A36 steel and E6013 electrodes was performed with wet welding. Finally, K_{Ic} -CVN correlations were presented to measure the fracture toughness.

2. Experimental Procedure

2.1. Wet welding

T-welded connections of A36 steel were manufactured in dry conditions. The T-welded connections were welded employing E6013 electrodes following the procedure detailed by Terán et al. [17,18] and conforming to the welding procedure specification (WPS) AWS D.1.1/D1.1M code [19]. In Figure 1a, T-welded connections and Charpy specimens are graphically presented. The rectangular grinding was carried out with a manual grinder machine employing 4 mm of width. Two different grinding depths of 6 mm and 10 mm, corresponding to 30% and 50% of the plate thickness, respectively, were carried out at the weld toe of the T-welded connections. Then, underwater wet welding was employed with three different water depths of 50 m, 70 m and 100 m. The wet welding method was done according to [18,20]. A hyperbaric chamber was employed to simulate the water depths, see Figure 1b. Inside of the chamber, a gravity welding system (GWS) was used to fill the rectangular grinding with wet welding, as illustrated in Figure 1c. The E6013 electrodes were used with diameters of 2.4 mm and 3.2 mm and the length of each electrode was 350 mm. The electrodes were coated with vinylic varnish. Variables used in wet welding are presented in Table 1. The polarity was from a direct current. A T-welded connection, after applying wet welding, is presented in Figure 1d.

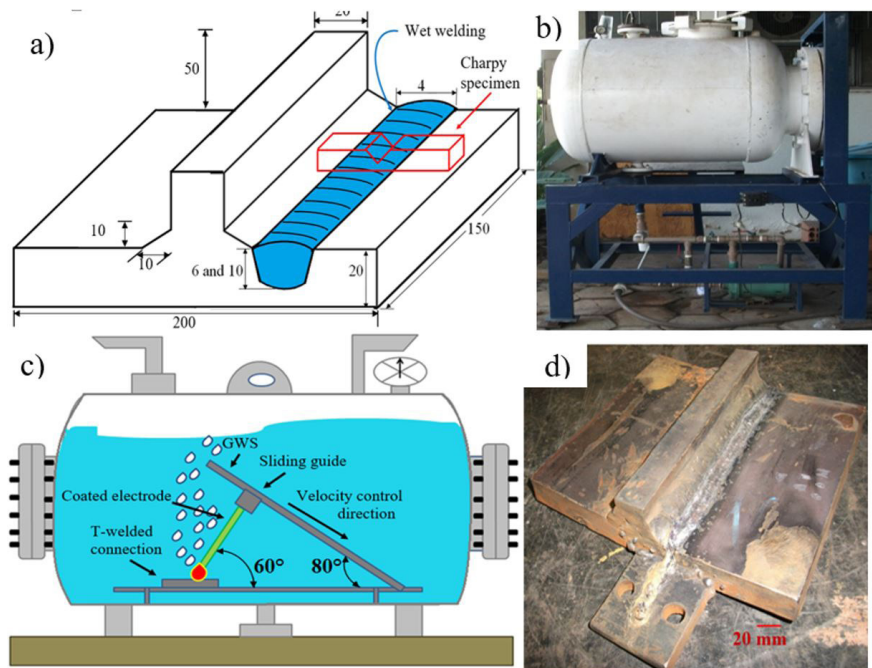


Figure 1. a) Charpy specimen extraction in T-welded connection, units in mm; b) Hyperbaric chamber; c) Gravity welding system (GWS) inside the chamber; and d) grinding filled with wet welding in T-welded connections.

Table 1. Variables used for testing the wet welding process [18].

Applied current (Amperes)	Electrode studying angle (Degree)	Electrode diameter (mm)	Water depth (m)
160	60	2.4 and 3.2	50 and 70
190	55	2.4 and 3.2	100

To obtain CVN values, standard specimens for Charpy V-notch (CVN) were extracted at the weld toe. Charpy tests were performed at a test temperature of 20 °C. A Charpy model 74 machine with a capacity of 0.0-274 ft-lb conforming to the recommendation of ASTM E23 [21] was employed. Charpy specimen's dimensions were 10 mm × 10 mm × 55 mm. For each grinding and water depth, three Charpy test were conducted.

2.2. K_{IC} -CVN correlations

A review has been conducted to obtain K_{IC} -CVN correlations. These were taken into the studying zone in the energy-temperature curve. Lower shelf and transition zones were considered because different authors conducted their impact test from -1 °C to 0 °C [22-29]. Also, there are studies for temperatures in the transition zone and upper zone [30]. In this study, Charpy values were conducted at room temperature, 20 °C. The equations from a literature review where the lower shelf region and transition temperature region were considered can be seen in Table 2. Then, K_{IC} -CVN correlations must be considered in the lower shelf, since different authors tested their Charpy tests at 0 °C. Robert-Newton [31], and INSTA [32] applied their correlations to the lower shelf region. While in the transition zone, the equations from Barsom-Rolfe [33], Marandet-Sanz [34], and Sailor-Corten [7] can be used. In the search of CVN by different authors, it has been necessary to employ the stress yield (σ_{ys}) of the material. Then, in order to make a Charpy energy values comparison and to obtain K_{IC} values, the Barsom-Rolfe's equation [31] was used. It is because it can be applied to all areas of the energy-temperature curve, lower shelf, transition temperature region and upper shelf for several steels. Another important consideration is that this equation considers the σ_{ys} of the material to convert K_{IC} -CVN values. Although those K_{IC} -CVN correlations are for base materials, they can be used in wet welding, since it is considered the brittle-ductile transition zone for steels.

Table 2. K_{IC} -CVN correlations for lower shelf regions and different zones found in published literature.

Transition temperature region		
Barsom and Rolfe [33]	$\frac{K_{IC}^2}{E} = 2(CVN)^{3/2}$	ksi \sqrt{in} , ksi, ft-lb
		40-250 ksi, 4-82 J
Marandet and Sanz [34]	$K_{IC} = 19(CVN)^{1/2}$	MPa \sqrt{m} , MPa, J
		303-820MPa 43-118ksi
Sailor and Corten [7]	$\frac{K_{IC}^2}{E} = 8(CVN)$	psi \sqrt{in} , CVN=ft-lbf E= psi
		268-923MPa 39-134 ksi
Lower shelf region		
Robert and Newton [31]	$K_{IC} = 8.47(CVN)^{0.63}$	MPa \sqrt{m} , J
INSTA [32]	$K_{IC} = 12\sqrt{CVN}$	MPa \sqrt{m} , J
Different zones		
Barsom and Rolfe [33]	$K_{IC} = R_{p0,2} \sqrt{\frac{5}{R_{p0,2}} \left(CVN - \frac{R_{p0,2}}{20} \right)}$	ksi \sqrt{in} , ksi, ft-lb

3. Results and Discussion

3.1. CVN data

Table 3 presents the Charpy CVN values and σ_{ys} of the material gathered by different authors who employed wet welding in ASTM A36 steel. It can be observed that Di Lorenzo [30] did not report the σ_{ys} values of the material. The overview of absorbed energy versus water depth is displayed in Figure 2. One objective in observing the absorbed

energy and temperature data is to know the ductile-brittle behavior from the Charpy transition temperature curve of the wet welding.

Di Lorenzo et al. [30] applied wet welding for A36 plates steel and E6013 electrodes to different meters of water columns (wcm) at 0 wcm, 20 wcm, 40 wcm and 60 wcm, which are 0 m, 19.36 m, 38.72 m and 58.08 m, of water depth, respectively. Figure 3 depicts Lorenzo's results where he used the tangent-hyperbolic method. It is noted that CVN values decrease as water depth increases. Although in this study only Charpy specimens were

Table 3. Absorbed Charpy energy and yield stress (σ_{ys}) values by different authors, (units in Joules and MPa) who employed wet welding on ASTM A36 steel.

	Depth [m]	J [Joules]	σ_{ys} [MPa]
West et al. [22]	2	41.5	539.5
	10	40.5	531
Szelagoswki et al. [23]	55	31	444
	61	17	415
Szelagoswki [24]	6	33	572
	55	22	524
	101	14	448
Grubbs and Reynolds [25]	6	46	531
	10	37	510
	50	42	434
	99	39	407
Rowe et al. [26]	21	21	489
	43	18	448
	61	16	407
	91	13	406
Perez-Guerrero et al. [27]	50	20	483
Di Lorenzo et al. [30]	19	39	--
	38	33	--
	58	25	--
Pessoa [28]	50	13	450
	100	12	425
Santo [29]	0.5	45.6	511
Teran et al. [17,18]	50*	19.0	430
	50*	19.7	372
	50*	25.1	409
	70*	19.0	363
	70*	19.7	365
	70*	15.6	318
	100*	10.0	301
	100*	12.5	323
	100*	10.0	285
	50 ⁺	16.0	410
	50 ⁺	18.0	392
	50 ⁺	18.5	382
	70 ⁺	16.0	376
	70 ⁺	12.5	327
70 ⁺	9.0	364	
100 ⁺	19.0	347	
100 ⁺	16.0	316	
100 ⁺	14.0	357	

*for 10 mm grinding depth. *for 6 mm grinding depth, & for Di Lorenzo, a temperature of 50 °C was chosen.

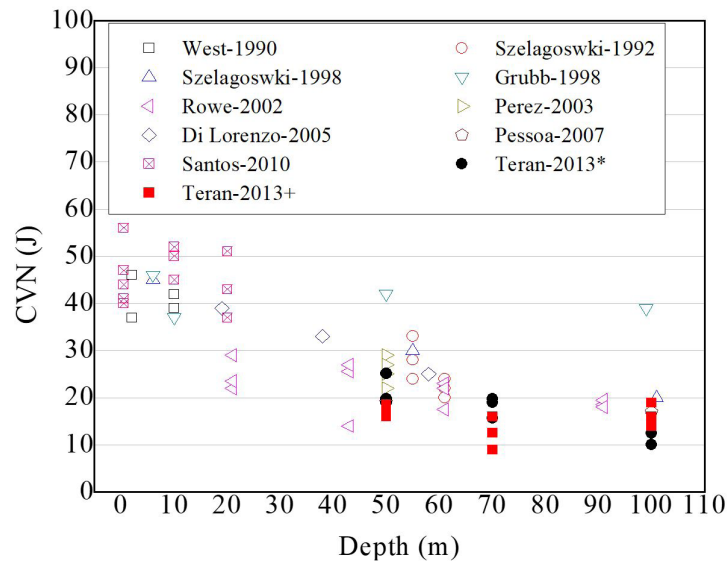


Figure 2. Comparison of CVN versus water depth for several authors.

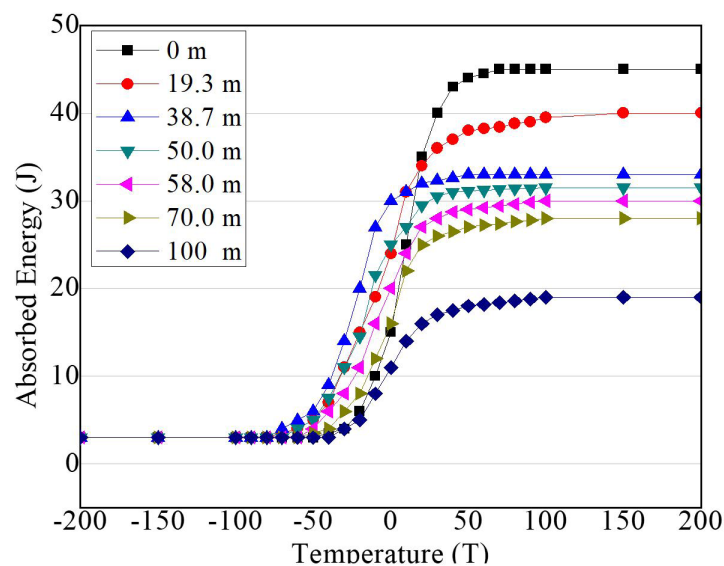


Figure 3. Charpy transition temperature curves for underwater wet welding at several depths, adapted from Di Lorenzo [30].

assessed at 20 °C, with the CVN data of other authors and Lorenzo's values, it is reasonable to predict the Charpy transition temperature curves for underwater wet welding at several depths.

It is well-recognized that energy absorption decreases as the water depth increases. During the wet welding process, different discontinuities can be present, such as, porosity, slag inclusion, non-metallic inclusion, or cracking. An image of a fracture surface for a Charpy specimen, as shown in Figure 4a, exhibits pores at the notch. This fracture surface was photographed after wet welding was employed. A great number of pores was observed in the weld beads (WB) on the fracture surfaces of the Charpy specimens. Figure 4b presents pores at the WB. This is due to the fast cooling of the weld, and it is difficult to eliminate these discontinuities [30]. The discontinuities reduce the energy absorbed, and, consequently, low CVN values can be anticipated as the water depth increases. Welding discontinuities, such as slag inclusions, lack of penetration, lack of fusion, cracking, undercutting and porosity, can be enlarged with the increase of depth of welding [35].

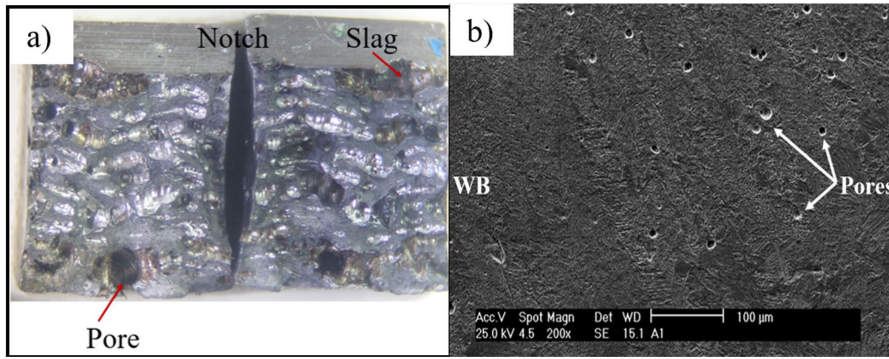


Figure 4. a) Charpy specimen tested with 10 mm of grinding depth and 50 m of water depth, and b) pores in the welding bead (WB).

3.2. K_{Ic} data of K_{Ic} -CVN correlations

K_{Ic} values estimated for this study are listed in Table 4 and Figure 5. In this table, K_{Ic} values are not reported by Di Lorenzo [30] because he did not report stress yield data and the Barsom-Rolfe equation [33] could not be used. This table proves that K_{Ic} values decreased as the depth increased. As presented above, this trend is because CVN values are proportionate to the K_{Ic} values and CVN decreases due to the porosity percentage, microstructure and slag produced in the wet weld beads. It is well-known that porosity is caused by gases (H_2 , CO and CO_2) trapped during weld melting. In these gases, the pores contain 96% of H_2 in volume, 0.4% of CO and 0.06% of CO_2 [36]. For example, Pessoa et al. [37] found porosity values of 1% and 8% for 50 and 100 m water depths because the slag reaches the top of the weld seams by itself, due to its low density. However, due to the hydrostatic pressure, it could not reach the top of the weld seam and was retained to develop porosity. Another explanation is that, when cracking is in the welding, these cracks connect the pores in the fracture planes, resulting in low K_{Ic} values [38,39]. The toughness and ductility are more affected than yield and strength limits [40]. The following microstructure phases were identified [17,18]: ferrite with aligned second phase (FSA), sideplate ferrite (FS) and grain-boundary ferrite (GBF). These microstructures are typical of low mechanical properties according to Charpy impact test values. Therefore, in order to have high Charpy impact data and K_{Ic} values, acicular ferrite in wet weld beads must be obtained. This can be achieved if specific elements are added to the electrodes, such as titanium and boron together with an adequate concentration of oxygen and manganese [36].

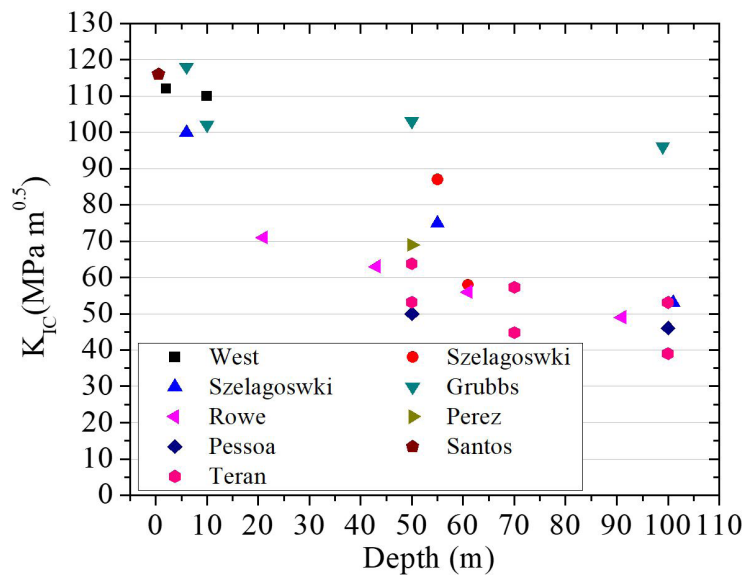


Figure 5. Comparison of K_{Ic} values as a function of water depth according to several authors.

Table 4. Mechanical properties of K_{IC} results, units in $MPa\sqrt{m}$.

	Depth, [m]	Barsom-Rolfe's equation, [33]
		K_{IC} [$MPa\sqrt{m}$]
West et al. [22]	2	112
	10	110
Szelagoswki et al. [23]	55	87
	61	59
Szelagoswki [24]	6	100
	55	75
	101	53
Grubbs and Reynolds [25]	6	118
	10	102
	50	103
	99	96
Rowe et al. [26]	21	71
	43	63
	61	56
	91	49
Perez-Guerrero et al. [27]	50	69
Pessoa [28]	50	50
	100	46
Santos [29]	0.5	116
Teran et al. [17,18]	50*	64.0
	50*	62.0
	50*	74.5
	70*	60.1
	70*	61.5
	70*	50.5
	100*	36.9
	100*	44.0
	100*	36.3
	50 ⁺	39.7
	50 ⁺	59.7
	50 ⁺	60.4
	70 ⁺	54.6
70 ⁺	44.2	
70 ⁺	35.6	
100 ⁺	59.0	
100 ⁺	51.2	
100 ⁺	49.1	

*for 10 mm grinding depth. ⁺for 6 mm grinding depth.

It can be seen, as expected, that K_{IC} values reported by Barsom [13] are close to the real K_{IC} measured values. It is necessary, however, to compare this data testing with the ASTM E399 [9]. Barsom uses the σ_{ys} of the material to estimate K_{IC} values. Using the yield stress of the material and Charpy values, it is possible to achieve more reliable results which are close to the ASTM E399 standard. In this sense, to estimate K_{IC} -CVN values, more K_{IC} -CVN correlations are required. Other parameters of the material in the K_{IC} -CVN correlations, such as, σ_{ys} , ultimate tensile stress (σ_{UTS}), and hardness (HRC), as well as the microstructures could be considered. For example, Salemi et al. [41] determined that it is necessary to establish the microstructures and to better determine their mechanical behavior. If the microstructure used in the K_{IC} -CVN correlations is not indicated, erroneous values

could be obtained. Qamar [8] developed models for K_{Ic} data based on hardness (HRC) and impact energy (CVN) data. Dexter [42] measured the J value for underwater wet welds for several types of steels, electrodes and water depths using J_{Ic} compact tension specimens. In these conditions, A36 steel, E6013 electrodes and water depths of 10, 35 and 60 m are chosen. Using the conversion of J values to K_{Ic} results for water depths of 10 m, 35 m and 60 m, K_{Ic} values are $7.2 \text{ MPa}\sqrt{m}$, $39.12 \text{ MPa}\sqrt{m}$, and $44.50 \text{ MPa}\sqrt{m}$, respectively. K_{Ic} data decreases as water depths increase, which agrees with the values presented above. On the other hand, it can be seen that there is no single fracture toughness value for steel even at a fixed temperature and loading rate as pointed out by several authors [43-45]. Thus, at room temperature, the fracture toughness values measured at high loading rates are lower than those measured at lower loading rates. Therefore, K_{Ic} can vary for the same material tested under similar laboratory conditions. As expected, the K_{Ic} values obtained in the present study can be considered effective and used in research to estimate the real fracture toughness. Therefore, this data can be taken as a starting point to estimate K_{Ic} values.

Hence, with traditional equations of K_{Ic} , it was possible to estimate K_{Ic} values and to assess traditional components. The treatment of fracture toughness data is used in the analysis of fracture mechanics and depends on the available data. This dependence makes the structural integrity evaluation difficult when only applying a simplified procedure. Data of fracture toughness cannot be available in all situations or cannot be obtained due to the lack of material or the impossibility of removing material from an actual structure. CVN testing is less demanding than K_{Ic} testing in terms of experimental complexity, speed, and cost [8]. We know that the correlation equations are based on base materials without welding. However, there are no other equations proposed to obtain K_{Ic} data, thus, these equations will serve to estimate K_{Ic} data and could be applied for several metals and wet welding. In these circumstances, the Charpy impact data can be all the available and reliable correlation between the Charpy impact energy and the fracture toughness that should be found [30].

The Charpy impact test will be used to indirectly estimate the fracture toughness of metals. It is necessary to make a K_{Ic} direct measurement and compare the K_{Ic} correlations values with standard specimens. However, because of the relative difficulty and expense of these tests, the Charpy test will probably continue to be used. The correlation between the Charpy CVN and K_{Ic} fracture toughness will be topics for future study due to the importance of K_{Ic} in the LEFM. In addition, this standard characterizes the critical value of the crack driving force at the initiation of crack extension, and this allegedly represents a material property that is transferable from a test specimen to a structure [45].

For the sake of illustration, Figure 6 shows normalized boxplot graphs of the distribution of the variables involved in the experiment of this study. In the Equation 1 [46], it was used to normalize the variables [46]:

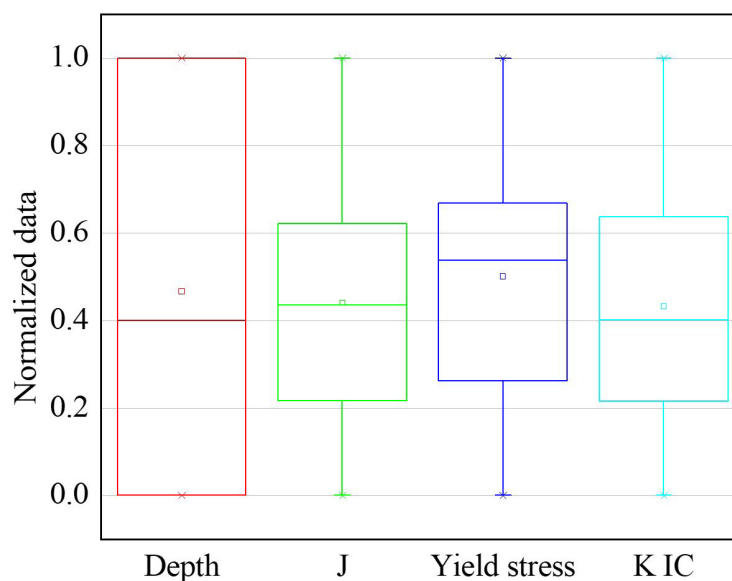


Figure 6. Normalized boxplot graph for the studied variables.

$$X^* = \frac{X - X_{\min}}{X_{\max} - X_{\min}} \quad (1)$$

In the equation above, X^* is the normalized value of the variable, X is the actual value, X_{\min} is the minimum value and X_{\max} is the maximum value. The boxplot graph is divided by minimum value, first quartile, median, third quartile and maximum. The central rectangle extends from the first to second quartile, representing the interquartile range (IQR). The rectangle and horizontal line inside the IQR display the mean and the median respectively. The summary of the experimental results is shown in Table 5 for each studied variable; this summary is useful for plotting Figure 6.

Table 5. Summary of the variables obtained in the experiment.

	Depth [m]	J [Joules]	σ_{ys} [MPa]	K_{Ic}
Minimum	50	9	285	35.6
Maximum	100	25.1	430	74.5
Mean	73.33	16.08	357.61	52.40
Standard Deviation	21.14	4.17	39.92	11.16

4. Conclusions

K_{Ic} values were estimated from K_{Ic} -CVN correlations from CVN impact data by authors who employed A36 steel, wet welding and E6013 electrodes. CVN impact energy and K_{Ic} values decrease as the sea depth increases. It is attributed to various discontinuities in wet welding, such as porosity, slag-inclusion, non-metallic inclusion and cracking. Although the equations of K_{Ic} -CVN correlations did not specify that they can apply to wet welding, the K_{Ic} values obtained by Barsom could be considered the closest values in order to compare them with the ASTM E369 standard specimens. This is because they use values from Charpy (CVN) in all zones of the absorbed energy curve, and the yield stress of the material. It is necessary to conduct or develop further correlation equations to introduce different mechanical properties, such as σ_{ys} , σ_{UTS} , hardness (HRC) and define microstructure. It is noted that there is no single fracture toughness value for steels, nor a fixed temperature, Charpy CVN data and water depths. A relatively accurate K_{Ic} -CVN correlation can be a useful tool in the MFLE. CVN testing is less demanding than K_{Ic} testing in terms of experimental complexity, speed, and cost. Then, traditional equations to estimate K_{Ic} data will be used. It is due to K_{Ic} representing a mechanical material property that it could be transferable from a test specimen to a structure.

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References

- [1] McNicol, R.C. Correlations of charpy test results for standard and nonstandard size specimens. Shaker Heights: Welding Research Council 385; 1965.
- [2] Phaal R, Macdonald KA, Brown PA. Correlations between fracture and charpy impact energy. Cambridge: Cooperative Research Programmed for Industrial Members Only; 1994. TWI Report 504. The Welding Institute.
- [3] Barsom JM, Rolfe ST. Fracture and fatigue control in structures. 3rd ed. New Jersey: Prentice Hall Englewood Cliffs; 1999.
- [4] Rolfe ST, Novak SR. Slow-bend K_{Ic} testing of medium high-toughness steel. Review of development in plane strain fracture toughness testing. ASTM Special Technical Publication. 1970;463:124-159.
- [5] Barsom JM, Rolfe ST. Correlations between K_{Ic} and Charpy V-notch test results in the transition-temperature range. Impact Testing of Metals. ASTM Special Technical Publication. 1970;466:281-302.
- [6] Roberts, R., Newton, C. Interpretive report on small scale test correlations with KIC data. New York: Welding Research Council; 1981. (WRC Bulletin; 265).
- [7] Sailors SH, Corten HT. Relations between material fracture toughness using fractures mechanics and transition temperature test. Fracture toughness. Proceeding of the 1971, National Symposium on Fracture Mechanics – Part II, STP 514, ASTM; 1972 August 31-September 2; Illinois, USA. Illinois: University of Illinois; 1972. p. 164-191.

- [8] Qamar SZ, Sheikh AK, Arif AFM, Pervez T. Regression-Based CVN- K_{IC} models for hot study tool steels. *Materials Science and Engineering A*. 2006;430(1-2):208-215. <http://dx.doi.org/10.1016/j.msea.2006.05.103>.
- [9] American Society for Testing and Materials. ASTM E-399: standard test method for linear-elastic plane-strain fracture toughness K_{IC} of metallic materials. West Conshohocken: ASTM; 2009.
- [10] Matusevich AE, Mancini RA, Giudici AJ. Determinación de la tenacidad a la fractura del material de un gasoducto. *Revista Latinoamericana de Metalurgia y Materiales*. 2010;32(2):253-260.
- [11] Wullaert RA. Fracture toughness predictions from Charpy V-notch data: what does the Charpy test really tell us? *Proceedings of the American Institute of Mining, Metallurgical and Petroleum Engineers*; 1978 February 27-28; Denver, Colorado, USA. Cleveland: American Society for Metals; 1978.
- [12] Roberts R, Newton C. Report on small-scale test correlations with K_{IC} data. *Welding Research Council Bulletin*. 1981;2(265):1-18.
- [13] Barsom JM. The development of AASHTO fracture toughness for bridge steel. *Engineering Fracture Mechanics*. 1975;7(3):605-618. [http://dx.doi.org/10.1016/0013-7944\(75\)90060-0](http://dx.doi.org/10.1016/0013-7944(75)90060-0).
- [14] Marandet B, Sanz G. STP 631: evaluation of the toughness of the medium-strength by using elastic fracture mechanics and correlations between KIC and Charpy V-notch, Flaw Growth and Fracture. West Conshohocken: ASTM; 1977. p. 72-95.
- [15] Norris DM, Reaugh JE, Server WL. A fracture-toughness correlations based on Charpy initiation energy. *Proceedings of the 13th Conference Fracture Mechanics STP 473*; 1980; Philadelphia, USA. Philadelphia: ASTM; 1981. p. 207-217.
- [16] Walling, K. New report methodology for selecting Charpy toughness criteria for thin high strength steels. Report Represented to Commission X, IIW 1994, Annual Assembly. Beijing: IIW; 1994. DOC. NO. X.1290.
- [17] Teran G, Capula-Colindres S, Angeles-Herrea A, Velazquez JC, Fernandez-Cueto MJ. Estimation of fracture toughness KIC from Charpy impact test data in T-welded connections repaired by grinding and wet welding. *Engineering Fracture Mechanics*. 2016;153:351-359. <http://dx.doi.org/10.1016/j.engfracmech.2015.12.010>.
- [18] Teran G, Cuamatzi-Melendez R, Albiter A, Maldonado C, Bracarense AQ. Characterization of the mechanical properties and structural integrity of T-welded connections repaired by grinding and wet welding. *Materials Science and Engineering A*. 2014;599:105-115. <http://dx.doi.org/10.1016/j.msea.2014.01.078>.
- [19] American Welding Society. AWS D1.1/D1.1M: structural welding code-steel. Miami: AWS; 2006.
- [20] Méndez GT, Cuamatzi-Meléndez R, Hernández AA. Combination of grinding and wet welding to repair localized cracking in t-welded connections. *Materials Science Forum*. 2014;793:51-58. <http://dx.doi.org/10.4028/www.scientific.net/MSF.793.51>.
- [21] American Society for Testing and Materials. ASTM E23-12: standard test methods for notched bar impact testing of metallic materials. West Conshohocken: ASTM; 2012.
- [22] West C, Mitchell G, Linberg E. Wet welding electrode evaluation for ship repair. *Welding Journal*. 1990;69(8):46-56.
- [23] Szelagowsky P, Pachniuk I, Stuhff H. Wet welding for platform repair. *Proceeding Second International Offshore and Polar Engineering Conference*; 1992 June 14-19; San Francisco, California, USA. San Francisco; 1992. p. 208-215.
- [24] Szelagowsky P. Wet welding as a serious repair procedure? *Journal of Offshore Mechanics and Arctic Engineering*. 1998;120(3):191-196. <http://dx.doi.org/10.1115/1.2829540>.
- [25] Grubbs CE, Reynolds TJ. State of the art underwater wet welding. Houston: World Oil Magazine; 1998. p. 79-83.
- [26] Rowe MD, Liu S, Reynolds TJ. The effect of ferro-alloy additions and depth on the quality of underwater wet welds. *Welding Journal*. 2001;81(8):1565-1665.
- [27] Perez-Guerrero F, Liu S, Smith, C., Rodriguez-Sanchez, E. Effect of nickel on toughness of underwater wet welds. *OMAE*. 2003;37261:1-6.
- [28] Pessoa ECP. Estudo da variação da porosidade ao longo do cordão em soldas subaquática molhadas [doctor thesis]. Belo Horizonte: Federal University of Minas Gerais; 2007.
- [29] Santos VR, Bracarense AQ, Pessoa ECP. Underwater welding consumables development. *International studieshop on the state of the art science and reliability of underwater and inspection technology*. In: *International workshop on the state of the art science and reliability of Underwater welding and inspection technology*. 2010 November 17-19; Houston, Texas, USA. Houston; 2010, p. 172-197.
- [30] Di Lorenzo RF, Soares WA, Bracarense AQ. Fracture toughness of ferritic steel underwater wet welding. *Proceedings of the 18th International Conference on Structure Mechanics in Reactor Technology (SMiRT 18)*; 2005 August 7-12; Beijing, China. Beijing: SMiRT 18; 2005. p. 1882-1895.
- [31] Roberts R, Newton CC. Report on small-scale test correlations with K_{IC} data. New York: Welding Research Council; 1984. (WRC Bulletin; 299)
- [32] INSTA. Technical report: assessment of structures containing discontinuities. Stockholm: Materials Standards Institution; 1991.
- [33] Barsom JM, Rolfe ST. ASTM STP 466 - impact testing of metals: correlations between KIC and Charpy V-notch test results in the transition-temperature range. West Conshohocken: ASTM; 1970. p. 281-302.
- [34] Marandet B, Sanz G. STP 631: evaluation of the toughness of the medium-strength by using elastic fracture mechanics and correlations between KIC and Charpy V-notch, Flaw Growth and Fracture. West Conshohocken: ASTM; 1977. p. 72-95.
- [35] Watson PD, Tsai CL, Wood B. Fitness of service design application for underwater wet welds. *Proceedings of the International studieshop on underwater welding of marine structures*; 1994 December 7-9; New Orleans, Louisiana, USA. Houston: American Bureau of Shipping, p. 201-236; 1994.
- [36] ASM Handbook®. Welding, Brazing and Soldering. 7th ed. Vol. 6. ASM International; 2005.
- [37] Pessoa ECP, Bracarense AQ, Zica EM, Liu S, Perez-Guerrero F. Porosity variation along multipass underwater wet welds and its influence on mechanical properties. *Journal of Materials Processing Technology*. 2006;179(1-3):239-243. <http://dx.doi.org/10.1016/j.jmatprotec.2006.03.071>.
- [38] Angeles-Herrera D, Albiter-Hernandez A, Cuamatzi-Melendez R, Gonzalez-Velazquez JL. Fracture toughness in the circumferential-longitudinal and circumferential-radial directions of longitudinal Weld API 5L X52 pipeline using standard C(T) and nonstandard curved SE(B) specimens. *International Journal of Fracture*. 2014;188(2):251-256. <http://dx.doi.org/10.1007/s10704-014-9949-1>.
- [39] Angeles-Herrera D, Gonzalez-Velazquez J, Morales-Ramírez AJ. Fracture-Toughness evaluation in submerged arc-welding seam welds in nonstandard curved SE(B) specimens in the short radial

- direction of API 5L Steel pipe. *Journal of Testing and Evaluation*. 2012;40(6):1-4. <http://dx.doi.org/10.1520/JTE103903>.
- [40] Danninger H, Jangg G, Weiss B, Stickler R. Microstructure and mechanical properties of sintered iron. Part II: experimental study. *International Journal of Powder Metallurgy*. 1993;25(4):170-173.
- [41] Salemi A. The effect of microstructure on estimation of the fracture toughness (K_{IC}) rotor steel using Charpy absorbed energy (CVN). *Journal of Advanced Materials and Processing*. 2013;1(3):11-17.
- [42] Dexter RJ. ASTM STP 1058: fracture toughness of underwater wet welds. *Fatigue and fracture testing of weldments*. In: McHenry HI, Potter JM, editors. American Society for Testing and Materials. Philadelphia: ASTM; 1990. p. 256-271.
- [43] Ripling EJ, Crosley PB. Crack arrest fracture toughness of a structural steel A36. *Welding Research*. 1982;(Suppl):65s-74s.
- [44] Sovak, J.F. The fracture resistance of 4-in thin A36 and A588 grade A electroslag weldments. *Welding Research*. 1981;(Suppl):269s-272s.
- [45] Federal Emergency Management Agency. State of the art report on base metals and fracture. Washington: FEMA; 2000. FEMA-355A.
- [46] Velazquez JC, Cruz-Ramirez JC, Valor A, Venegas V, Caleyó F, Hallen JM. Modeling localized corrosion of pipeline steels in oilfield produced water environments. *Engineering Failure Analysis*. 2017;79:216-231. <http://dx.doi.org/10.1016/j.engfailanal.2017.04.027>.